

이산화탄소 포집 및 탄산염 전환 통해 제조된 FeCO₃/rGO 복합체: 리튬이온 전지향 고성능 음극재

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FeCO₃/rGO Composites Prepared by CO₂ Capture and Carbonate Conversion: Anode Material in Lithium-Ion Batteries with Enhanced Performance

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요약: 탄소중립을 달성하기 위해 이산화탄소를 포집, 활용, 저장하는 CCUS (carbon capture, utilization, and storage) 기술이 주목받고 있다. 본 연구에서는 광물 탄산화 공정을 통해 이산화탄소를 탄산염으로 고정하고, 이를 전이금속 탄산염 기반 리튬이온배터리 (LIB) 음극재로 적용하였다. CO₂를 탄산염으로 고정후, 이를 이용해 FeCO₃를 제작하고, rGO와 PVP와 복합화하여 음극활물질에 적용하였다. rGO는 전기전도도를 높이고 입자의 응집을 방지해 부피 팽창을 완화했으며, PVP는 계면 활성제로서 입자 표면을 안정화하여 구조적 안정성을 강화하였다. FeCO₃-PVP-rGO 복합체 기반한 음극재에 대한 전기화학 테스트를 진행한 결과, FeCO₃/rGO 복합체는 1,620 mA/g의 전류 밀도에서 50 사이클 이후에도 400 mAh/g의 용량을 유지하였다. 본 연구는 CO₂를 고부가가치 배터리 소재로 전환하여 차세대 에너지 저장 기술에 기여할 가능성을 시사한다.

Abstract: Carbon capture, utilization, and storage (CCUS) technology is gaining attention as a key strategy for achieving carbon neutrality. In this study, CO₂ was permanently fixed as carbonate through mineral carbonation and applied as an anode material for lithium-ion batteries (LIBs). Such a fixed carbonate by CO₂ was used for the preparation of FeCO₃ composited with rGO and PVP to be applied as active material for negative electrode in LIBs. Specifically, the rGO plays an important role to increase electrical conductivity and prevent particle aggregation while PVP could stabilize the particle surface and strengthened structural stability as a surfactant. Electrochemical tests showed that the LIB anode material based on FeCO₃/rGO composite maintained a capacity of 400 mAh/g after 50 cycles at a current density of 1,620 mA/g. This study suggests the converting CO₂ into valuable battery materials can contribute to future energy storage technologies.

Keywords: lithium ion battery, CO₂ capture, FeCO₃, anode, rGO

1. Introduction

The drastic climate change and environmental pollution have become globally significant issue[1-3]. Carbon dioxide (CO₂) emissions from industrial processes have been identified as one of the major causes

of global warming[4-5]. Therefore, it is crucial to develop technologies that can capture and convert CO₂ into useful substances. carbon capture and utilization (CCU) and carbon capture, utilization, and storage (CCUS) technologies have gained attention for storing CO₂ in a stabilized form by converting it into useful

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materials. Specifically, mineral carbonation technology is an effective method to convert CO₂ into chemically stable storage form of carbonates with high environmental safety[6-9].

Graphite, the conventional anode material for lithium-ion batteries (LIBs), has a limited energy density due to its low theoretical capacity of 372 mAh/g, facing to be changed with alternative anode material with high capacity including metal oxide, Si and lithium [10-20]. Among them, transition metal carbonates have been highlighted as the potential alternative anode to significantly improve LIB performance due to their high reversible capacity. However, the existing transition metal carbonate anodes failed to be commercialized due to its low electrical conductivity and volume expansion, hindering their practical application towards LIB anode with high electrochemical performance[21-23].

In this regards, iron carbonate (FeCO₃) is an appropriate transition metal carbonate for LIB anode. Specifically, FeCO₃ is low-cost and non-toxic material which can be prepared by CO₂ conversion. However, the practical application of iron carbonate towards LIB anode is limited by its poor electrochemical performance due to low electrical conductivity and volume expansion of FeCO₃[24-26].

In this study, we report a novel CCUS technology including mineral carbonation of CO₂ to be utilized as LIB anode material through simple and facile process. Briefly, we first prepare FeCO₃ by serial process of CO₂ capture and sodium carbonate conversion followed by metathesis. For the enhancement of conductivity and volume expansion of FeCO₃, a composite of reduced graphene oxide (rGO) and polyvinylpyrrolidone (PVP) and FeCO₃ was prepared through hydrothermal process and high temperature sintering. The material and electrochemical properties of the synthesized nanocomposites were analyzed to evaluate the applicability of nanocomposite towards LIB anode.

2. Experimental

2.1. Materials

Sodium hydroxide (NaOH, 99.5%), sodium chloride (NaCl, 99.0%), L-ascorbic acid (C₆H₈O₆, AA, 99.0%), ethylene glycol (EG, 99.5%), hydrochloric acid (HCl, 35.0%), sulfuric acid (H₂SO₄, 95.0%), potassium permanganate (KMnO₄ 99.3%), hydrogen peroxide (H₂O₂, 30.0%) were purchased from Duksan Chemical. Iron(III) chloride hexahydrate (FeCl₃ · 6H₂O, 99.0%), graphite (99%), phosphoric acid (H₃PO₄, 85.0%), ethyl ether (99.0%) were purchased from Daejung chemical. All materials were used as-received for synthesis without further purification.

2.2. Synthesis of sodium carbonate through mineral carbonation

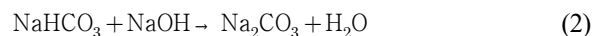
The sodium carbonate was prepared through carbonation and low-temperature crystallization with 2M NaOH under a constant temperature (35°C). The initial pH of the NaOH solution was approximately 14 while CO₂ (95 vol.%, 200 cc/min) was then injected into the NaOH solution until the pH of solution reached to 11. After the CO₂ injection, the reactor temperature was lowered to 10°C for the facilitated CO₂ conversion into bicarbonate and carbonate ions. The injection of CO₂ was continued for the crystallization at 2°C until the pH of the mixture reached to 7. After the crystallization, the white precipitate was formed in the solution and was separated through filtration.

The overall chemical reaction for Na₂CO₃ formation can be described as follows:

Initial reaction with NaOH to form sodium bicarbonate:



Further reaction of sodium bicarbonate with NaOH to form sodium carbonate:



In this experiment, 10.2 g of CO₂ was injected into the NaOH solution, leading to the formation of 8.1 g of Na₂CO₃. Considering the molar weights of CO₂ and Na₂CO₃, and assuming a 1:1 molar conversion ratio, the theoretical Na₂CO₃ yield was calculated. The actual yield achieved, 8.1 g, corresponds to a percentage yield of approximately 32.9%. These data provide a quantitative measure of the CO₂ conversion efficiency under these experimental conditions and indicate that around 1.26 g of CO₂ is required to produce 1 g of Na₂CO₃.

2.3. Synthesis of FeCO₃/rGO composite

For the preparation of FeCO₃/rGO composite, the previously synthesized sodium carbonate was used as a carbonate precursor to be mixed with graphene oxide. Specifically, graphene oxide (GO) dispersed in the distilled water was mixed with the solution of FeCl₃ in an ethylene glycol (EG)-water mixture (1:1 volume ratio). Then, ascorbic acid (AA) and polyvinylpyrrolidone (PVP) were subsequently added to the mixed solution of GO and FeCl₃ for the preparation of precursor for the subsequent hydrothermal synthesis. The as-prepared precursor was transferred to a teflon-lined autoclave to be synthesized as FeCO₃/GO composite at 180°C for 12 hours. In the end, FeCO₃/rGO composite can be prepared by high temperature reduction by sintering at 500°C for 4 hours under a nitrogen (N₂) atmosphere.

2.4. Material characterization

The crystal structure and chemical composition of the sodium compound, FeCO₃, and FeCO₃/rGO composite were characterized with physical and chemical analytical instruments. The crystal structure of the material was analyzed by X-ray diffractometer (XRD, Rigaku) at a 2θ range of 20~60° using CuKα (= 0.154059 nm) as radiation source. Surface chemical functionality (chemical bonds and molecular vibrations) for materials were analyzed mounted on ZnSe crystal at a wavenumber range of 750~2500 nm with Fourier Transform Infrared Spectroscopy (FT-IR, JASCO). The morphological information of the materials were obtained with

Scanning Electron Microscopy (FE-SEM, Hitachi) using a secondary electron detector (SE) under an acceleration voltage of 5 kV and a working distance of 8 mm.

2.5. Electrochemical characterization

The electrochemical properties of the material was characterized by analyzing their symmetric cell performance and electrochemical impedance spectroscopy (EIS). The cell electrodes was prepared with pasted slurry mixtures of the active material, Super P, and PVDF at 7:2:1 on the Cu foil. The as-prepared cell electrode was then cut into discs and assembled to coin type electrochemical cells containing lithium foil, separator (celgard 2400), and electrolytes (1 mol/L LiPF₆ in ethylene carbonate (EC)/dimethyl carbonate (DMC)). Galvanostatic cycling tests were conducted using a battery tester (WBCS3000, Wonatech) system over the voltage range of 0.01~3.0 V. Electrochemical impedance spectroscopy (EIS) was performed in the frequency range 100 kHz to 0.1 Hz with a potentiostat.

3. Results and Discussion

3.1. Material characterization

Fig. 1. displays the crystal structure of FeCO₃ and FeCO₃/rGO composite. All the peaks of the samples can be indexed into rhombohedral FeCO₃ (R3c space group, PDF Card No.: 5000036). Different from the characteristic peaks of FeCO₃, a weak and broad band of FeCO₃/rGO suggests the presence of rGO due to the decreased crystallinity and a reduced grain size of FeCO₃[28]. As-calculated grain sizes for FeCO₃ and FeCO₃/rGO are estimated to be 7.82 and 2.31 nm, respectively, attributing to the suppressed aggregation of FeCO₃ nanoparticles uniformly distributed on the rGO nanosheets[29].

Fig. 2 shows the FT-IR spectra of Na₂CO₃ prepared by CO₂ capturing and FeCO₃ synthesized using Na₂CO₃ as a carbonate precursor. As shown in the FT-IR spectra, strong absorption peaks at 1,390 cm⁻¹ and 880 cm⁻¹ observed in Na₂CO₃, indicates the

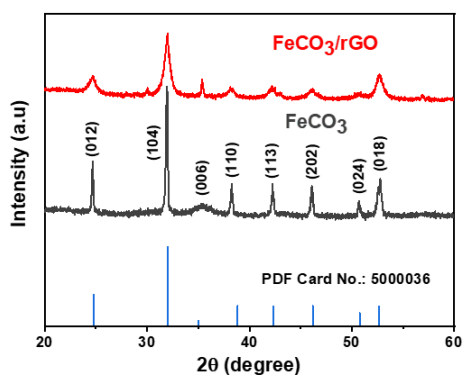


Fig. 1. Crystal structure analyses: XRD patterns of FeCO_3 and FeCO_3/rGO composite.

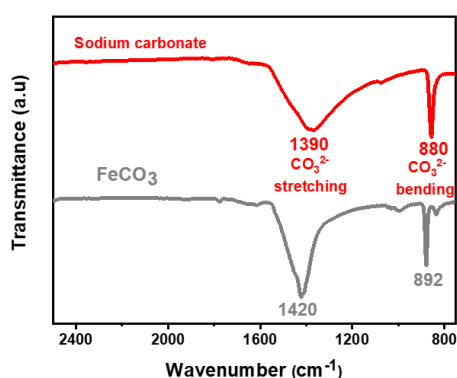


Fig. 2. Surface chemical functionality analyses: FT-IR spectra of sodium carbonate (Na_2CO_3) and iron carbonate (FeCO_3) synthesized via mineral carbonation.

stretching and bending vibrations of CO_3^{2-} , indicating the successful carbonate ions formation of Na_2CO_3 [13]. Similarly, strong absorption peaks at 1420 cm^{-1} and 892 cm^{-1} observed in FeCO_3 can be assigned to CO_3^{2-} , indicating the carbonate ions formed in FeCO_3 , similarly with that of Na_2CO_3 [19]. As demonstrated in the chemical functionality of materials, the successful CO_2 capturing have been performed through the effective storage as carbonate form and subsequently converted to FeCO_3 [30].

As displayed in the Fig. 3, the morphological structure of as-synthesized FeCO_3 and FeCO_3/rGO composite were analyzed with scanning electron microscopy (SEM). As shown in Fig. 3a, as-synthesized FeCO_3 microparticles by the solvothermal process are visible to have an irregular shape with the size of ca. 5~7

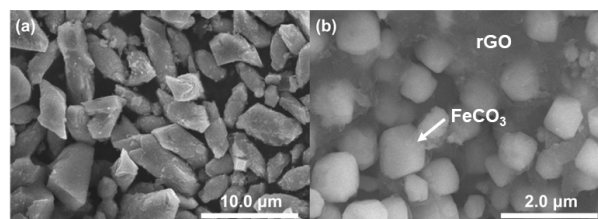


Fig. 3. Morphological analyses: SEM images of (a) FeCO_3 microparticles and (b) FeCO_3/rGO nanocomposites.

μm . As for the morphology for the rGO/FeCO_3 composite (Fig. 3b), as-formed FeCO_3 nanoparticles are observed as particles homogeneous distributed on rGO nanosheet with a mitigated size of 500~800 nm [31]. Such the different size and morphology of FeCO_3/rGO composite from that of FeCO_3 is attributed to the controlled growth of FeCO_3 nanoparticles in the rGO based composite matrix [15].

3.2. Electrochemical analysis

Fig. 4 compares the electrochemical properties for FeCO_3 and FeCO_3/rGO composite as the anode for lithium ion battery, As displayed in voltage profiles (Fig.s 4(a) and 4(b)) for charge/discharge at various cycles (1st, 2nd, 10th, and 50th cycles) at 1.62 A/g and 0.01 to 3 V, the large irreversible capacity losses are observed for FeCO_3 ($1,325\text{ mAh/g} \rightarrow 747\text{ mAh/g}$) and the FeCO_3/rGO composite ($1,490\text{ mAh/g} \rightarrow 908\text{ mAh/g}$) at the initial and second cycles due to the solid electrolyte interface (SEI) film formation and the degradation by the irreversible electrolyte/electrode reactions [11-18]. As-formed SEI and irreversible reactions reduce the available capacity for subsequent cycles by consumption of lithium ions [32]. Specifically, the declined plateaus are observed in the first discharge for the FeCO_3 (from 0.71 V to 0.43 V) and for the FeCO_3/rGO composite (from 0.78 V to 0.53 V) attributed to the iron reduction (Fe^{2+} to metallic Fe^0), and carbon reduction (C^{4+} to C^0) [32]. As shown in the voltage profiles, the plateau curve above 0.5 V indicates the SEI layer formation in the composite [33]. Note that, the diminished plateau with stabilized voltage profile after 10 cycles suggests the reversible elec-

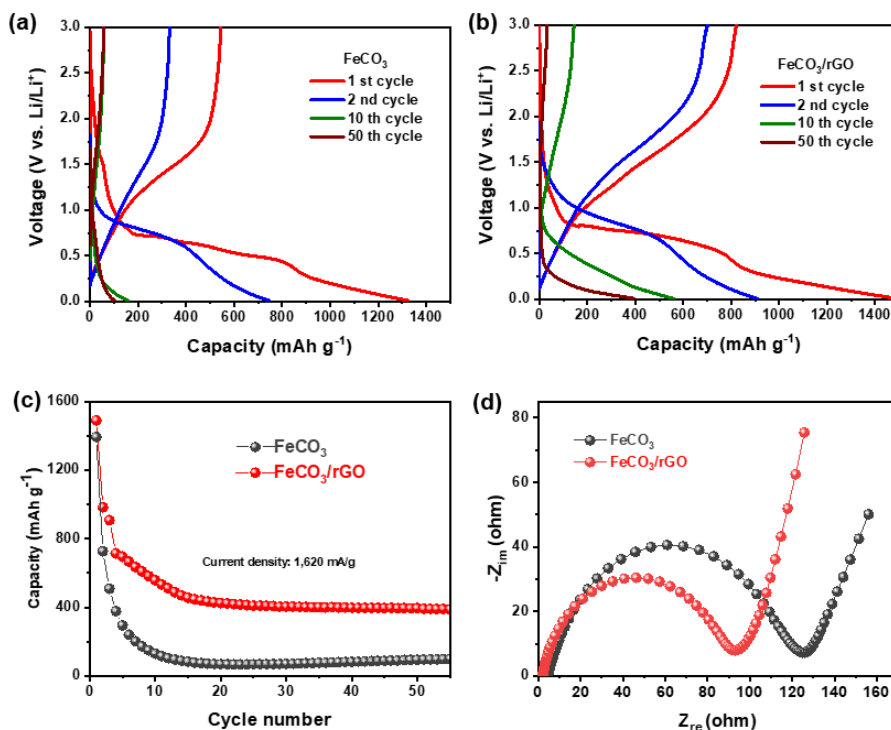


Fig. 4. Discharge/charge voltage profiles: (a) FeCO₃ (b) FeCO₃/rGO composite; (c) Cycling performance of FeCO₃ and FeCO₃/rGO composite; (d) EIS curves of FeCO₃ and FeCO₃/rGO composite.

trochemical reactions leading to the improved long-term cycle stability[34-35]. Furthermore, although the charge/discharge profiles show gradually decreased capacity at the elongated cycles (10th and 50th cycles) for both FeCO₃ and FeCO₃/rGO composites, FeCO₃/rGO composite displays much improved electrochemical performance compared to that of FeCO₃. Specifically, FeCO₃/rGO composite shows 557 mAh/g in the 10th cycle while the pristine FeCO₃ exhibits a capacity reduction to 159 mAh/g at the identical condition, indicating the more stabilized SEI layer and the reduced irreversible reactions of FeCO₃/rGO composite without continuous consumption of lithium ions [15-19]. Note that, FeCO₃/rGO composite shows the retained capacity (396 mAh/g) at elongated cycle (50th cycle) superior to that of FeCO₃ (103 mAh/g) at identical condition. Such the better capacity retention (cycle stability) of FeCO₃/rGO composite than that of pristine FeCO₃ indicates the mitigated structure degradation of FeCO₃ in the rGO composite due to structural stability and en-

hanced electronic conductivity by rGO nanosheet[17].

Fig. 4(c) compares the electrochemical performance (cycling stability and capacity) of FeCO₃ and FeCO₃/rGO composite. The FeCO₃/rGO nanocomposite exhibited the retained capacity (ca. 395 mAh/g) after 50 cycles, much superior to that of pristine FeCO₃ (95 mAh/g) at identical condition. Such the significantly improved performance (cycle stability and capacity retention) of the FeCO₃/rGO composite suggests effectively alleviated strain induced from FeCO₃ volume change and enhanced conductivity of composite by introduction of rGO in to composite.

As displayed in Fig. 4(d), the resistance of the samples were compared with electrochemical impedance spectroscopy (EIS) analyses. The semicircle observed in high-frequency corresponds to the charge-transfer resistance (R_{ct}). As displayed in the EIS results, FeCO₃/rGO composite exhibit much lower resistance (R_{ct} , 93.09 Ω) than that of bare FeCO₃ (121.62 Ω), indicating facilitated charge transfer and improved ion

Table 1. Comparison of Anode Materials

Samples	Capacity	Cycle	Ref.
FeCO ₃ -PVP-rGO	395 mAh/g	50	This work
FeCO ₃ /rGO	260 mAh/g	80	[31]
CoCO ₃	266 mAh/g	100	[36]
Al ₂ O ₃ coated graphite	335 mAh/g	100	[37]
GO-Mn ₃ O ₄	174 mAh/g	40	[38]
Structured graphite	233 mAh/g	100	[39]
N ₂ +H ₂ plasma modified graphite	185 mAh/g	500	[40]
AlF ₃ coated graphite	150 mAh/g	20	[41]
CNT fabric	305 mAh/g	30	[42]
Hard Carbon	290 mAh/g	100	[43]

mobility in the composite electrode[35].

4. Conclusion

In this study, we demonstrated a composite of FeCO₃ and rGO by carbonate based CO₂ capture/conversion and subsequent metathesis into FeCO₃ and rGO compositing to be applied as an anode material for LIBs.

As-prepared composite of FeCO₃ and rGO exhibited a superior electrochemical performance (retained cycle stability and capacity (400 mAh/g) after 50 cycles at 1.62 A/g) as LIB anode to that of pristine FeCO₃ based LIB anode (95 mAh/g) under the same test condition by overcoming the limitations of conventional transition metal carbonate anodes with the enhanced conductivity and alleviated strain of composite via introduction of rGO nanosheets. In addition, the FeCO₃/rGO composite exhibited a lower charge transfer resistance (R_{ct}) of 93.1 Ω than that for bare FeCO₃ (121.6 Ω), suggesting enhanced electron mobility and ionic conductivity of our composite.

As-demonstrated in this report, our study highlights the potential of wasted CO₂ utilization to address environmental issues (carbon neutrality) by CO₂ conversion into high-valued LIB materials[34].

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