

Ni Foam-Supported Ni Nanoclusters for Enhanced Electrocatalytic Oxygen Evolution Reaction

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ABSTRACT

Developing oxygen evolution reaction (OER) electrocatalysts is essential to accomplish viable CO₂ and water electrolysis. Herein, we report the fabrication and OER performance of Ni-foam (NF)-immobilized Ni₆ nanoclusters (NCs) (Ni₆/NF) prepared by a dip-coating process. The Ni₆/NF electrode exhibited a high current density of 500 mA/cm² for the OER at an overpotential as low as 0.39 V. Ni₆/NF exhibited high durability in an alkaline solution without corrosion. Electrokinetic studies revealed that OER can be easily initiated on Ni₆ NC with fast electron-transfer rates. Finally, we demonstrated stable CO₂-to-CO electroreduction using an NC-based zero-gap CO₂ electrolyzer operated at a current density of 100 mA/cm² and a full-cell potential of 2.0 V for 12 h.

Keywords : Oxygen evolution reaction, CO₂ reduction reaction, Nickel, Nanocluster, Electrode

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1. Introduction

Electrochemical CO₂ reduction reaction (CO₂RR) and hydrogen evolution reaction (HER) are recognized as promising strategies for storing renewable energy as chemical feedstock [1,2]. However, practical applications of CO₂RR and water electrolysis have suffered from the anodic oxygen evolution reaction (OER) [3-5], which requires a high overpotential because of the kinetically slow four-coupled electron and proton transfer pathways. Several OER electrocatalysts have been developed to reduce the overpotential. In particular, precious-metal-based Ir or Ru catalysts are predominantly used at present [6]. Therefore, inexpensive electrocatalysts with high activity are extensively studied to realize economically feasible energy-conversion electrolysis.

Recently, transition-metal-based OER catalysts such as metal oxides, metal sulfides, and layered double hydroxides have been reported [7-9]. Among

these, naturally abundant Ni catalysts have received considerable attention [10,11]. However, Ni catalysts generally undergo changes concerning the charge state and atomic structure during electrocatalysis [12,13]; therefore, correlating the catalytic activities with their structures is challenging. In addition, cutting-edge Ni-based OER catalysts typically require thermal treatment, significantly increasing the cost of production [14-16].

Recently, atomically precise metal nanoclusters (NCs) have been studied in various electrochemical systems [17-23]. They can be isolated as single species with molecular purity and show well-defined structures, as confirmed by X-ray crystallography [24]. Among them, Ni NCs are typically synthesized in the formula Ni_n(SR)_{2n}, where SR is a thiolate ligand (abbreviated as Ni_n) [25-27]. The Ni NCs are stable because of ligand protection and delocalized electrons on the cyclic ring structure [28,29]. In previous studies, Ni NCs have been exploited as OER electrocatalysts, typically immobilized on carbon substrates [30-33]. Kauffman *et al.* reported that Ni₆ NC exhibits higher activity than Ir or NiO. Using density functional theory calculations, they ascribed the high OER activity of the Ni₆ NC to the enhanced kinetics in the first electron transfer step [30]. Hus-

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sain *et al.* also reported the OER activities of Ni₄ and Ni₆ NCs, which exhibited higher OER activities than typical Ni-based electrodes [31]. Recently, Jin *et al.* reported the size-dependent OER activities of Ni₄, Ni₅, and Ni₆ NCs [32].

Although Ni NCs have been extensively studied as efficient electrocatalysts for water oxidation, their OER activity is far below the commercially viable level. Although carbon supports offer high electronic conductivity, they are prone to corrosion under electrochemical oxidation [34,35]. Although graphitized carbon species show relatively slow corrosion rates compared with typical carbon black [36], they also gradually corrode at high anodic potentials during long-term operations. Therefore, carbon-free OER catalysts with high durability are in high demand.

This study reports the preparation and OER activity of Ni-foam (NF)-supported Ni₆ NCs (Ni₆/NF) fabricated using the dip-coating method. The Ni₆ NC exhibited extraordinary OER activity on an NF substrate in an alkaline electrolyte. It showed high OER activity with a current density (j) of 500 mA/cm² at a low overpotential (η) of 0.39 V, outperforming the commercial IrO₂ electrocatalysts. Electrokinetic studies revealed that the high OER activity of Ni₆/NF originates from facile electron transfer on the non-corrosive substrate. In this study, an NC-based zero-gap CO₂ electrolyzer was constructed using a Au₂₅-immobilized gas diffusion electrode (Au₂₅/GDE) and Ni₆/NF electrodes as the cathode and anode, respectively, which exhibited highly active and stable CO₂RR and OER operations.

2. Experimental

2.1 NC Synthesis

The Ni₆(SEtPh)₁₂ NCs were synthesized following a previously reported method [37]. Briefly, NiCl₂·6H₂O (Merck, reagent grade, 0.42 mmol) and tetraoctylammonium bromide (Merck, 98%, 0.89 mmol) were dissolved in tetrahydrofuran (100 mL). After stirring for 30 min, 2-phenylethanethiol (HSEtPh) (Merck, 98%, 2.17 mmol) was added. NaBH₄ (Merck, 99%, 4.23 mmol) dissolved in cold water (14 mL) was added after 5 min, and the solution was stirred for an additional 24 h. The product was evaporated using a rotary evaporator. The organic phase was then separated and washed several times with methanol. The product was extracted using dichloromethane (CH₂

Cl₂) and crystallized by layering ethanol (10 mL) over the NC solution in CH₂Cl₂ (10 mL).

Au₂₅(SC₆H₁₃)₁₈ NC was prepared according to the previously reported procedure [38].

2.2 Characterizations of NCs

Ultraviolet-visible (UV-vis) absorption spectra of the NCs in CH₂Cl₂ were obtained using a Shimadzu UV-vis-NIR spectrophotometer (UV-3600). Electro-spray ionization (ESI) mass spectra were acquired using the negative ion mode in an ESI mass spectrometer (Agilent 6230 TOF LC/MS).

Single-crystal X-ray diffraction (SC-XRD) analysis was conducted on the crystals of Ni₆(SEtPh)₁₂ NCs grown at 25°C by layering hexanes over CH₂Cl₂ solution of Ni₆(SEtPh)₁₂ NCs (Fig. S1). The PAL BL2DSMDC program [39] was used for data collection. Cell refinement, reduction, and absorption correction were performed using HKL3000sm (Version 716.7) [40]. The crystal structure of Ni₆(SEtPh)₁₂ NCs was solved using the direct method with SHELX-XT (Ver. 2014/5) [41] and refined by full-matrix least-squares calculations with the SHELX-XL (Ver. 2016/4) [42] in the Olex2 [43] program package. The diffused electrons in large solvent-accessible void were treated using the SQUEEZE procedure (PLATON) [44,45].

After electrolysis experiments, the Ni₆ NCs and Nafion were extracted from the Ni₆/NF electrode using a mixture of CH₂Cl₂ and methanol. Pure Ni₆ NCs were further extracted with CH₂Cl₂ with approximately 50% of the recovery. The recovered NCs were analyzed using UV-vis absorption spectroscopy and mass spectrometry (Bruker, Autoflex Max).

2.3 Preparation and Characterization of Electrodes

The Ni₆ suspension for the 1 cm²-electrode was prepared by mixing Ni₆ NC (1.0 mg) and Nafion (10 μ L) (5 wt.%, Merck) in methanol (1.0 mL) and sonicating for 10 min. For large-scale fabrication, the suspension was scaled up proportionally to the electrode area. The Ni₆/NFs were prepared by dipping the NF (29-04275-01, Invisible Inc.) in the prepared suspension for 5 s and drying it for 1 h under a dynamic vacuum. We determined the loading amount of Ni₆ NC on the NF by measuring the weight change of the NF after the dipping process. Typical NC loading was determined to be 0.089 \pm 0.008 mg/cm². In addition, the loading amount was controllable (0.02, 0.05,

and 0.1 mg/cm²) by changing the Ni₆ concentration (0.2–1.0 mg/mL) in the dipping solution. Scanning electron microscopy (SEM), and energy-dispersive X-ray spectroscopy (EDS) analyses of the Ni₆/NF electrodes were performed using a JEOL scanning electron microscope (JEOL-7800F). For the EDS analysis of sulfur of the Ni₆/NF electrodes, the same weight of polyvinylidene fluoride (PVDF) binder was used instead of the sulfur-containing Nafion. The Ni₆-immobilized GDE (W1S1011, Ce-Tech) (Ni₆/GDE) was fabricated by spreading Ni₆ NC solution (0.5 mg of Ni₆ NC in 0.16 mL of acetone and 0.16 mL of CH₂Cl₂) on the microporous layer of 1 cm²-GDE. To compare the OER activities, commercial IrO₂-coated GDEs (IrO₂/GDEs) were purchased from Dioxide Materials.

The Au₂₅/GDE was prepared according to the procedure reported in a previous study [19].

2.4 Electrochemical Measurements

The half-cell measurement of the electrochemical OER was conducted using a three-electrode system in a one-compartment cell containing 20 mL of 1.0 M KOH solution with vigorous stirring to remove the generated O₂ bubbles and promote mass transport. The prepared anodes were used as the working electrodes; the typical anode area was 1 cm². A Pt foil cathode (1 cm²) and Ag/AgCl (1.0 M KCl) were employed as the counter and reference electrodes, respectively.

Linear sweep voltammetry (LSV), chronopotentiometry (CP), and electrochemical impedance spectroscopy (EIS) measurements were performed using a potentiostat (ZIVE BP2, WonATech). The NF anodes were pre-activated by conducting CP at 100 mA for 2 h before measuring their OER activity. LSV was conducted at a scan rate of 20 mV/s. The impedance spectra were recorded in a potentiostatic mode at various applied potentials in the frequency range of 100 kHz to 0.01 Hz with an amplitude of 10.0 mV.

The anode potentials recorded on the Ag/AgCl scale ($E_{\text{Ag/AgCl}}$) were converted to the reversible hydrogen electrode (RHE) scale (E_{RHE}) and η using the following equations:

$$E_{\text{RHE}} = E_{\text{Ag/AgCl}} + 0.21 + 0.059 \text{ pH} \quad (1)$$

$$\eta = E_{\text{RHE}} - 1.23 \text{ V} \quad (2)$$

The potential (E) was iR-compensated.

To measure the Faradaic efficiency of O₂ (FE_{O₂}), a laboratory-made, gas-tightened H-cell was used [20]. Residual air was purged with Ar gas, and the cell was tightly sealed. CP was performed at 50 mA for 1 h. After electrolysis, the headspace of the anode was analyzed by gas chromatography (GC) (GC 7890B, Agilent) equipped with a thermal conductivity detector and a flame ionization detector.

The full-cell measurement for CO₂RR-OER co-electrolysis was conducted in a zero-gap CO₂ electrolyzer [46], and the active areas of the cathode and anode were confined to 5 cm² by a flow field. CO₂ gas (200 sccm, humidified at 25°C) was fed into the cathode flow field, and fresh 1.0 M KOH was supplied to the anode line. In all CO₂RR experiments, CO and H₂ were produced as cathodic reaction products.

3. Results and Discussion

Ni₆(SEtPh)₁₂ NC was prepared according to a previously report (see the Experimental section for details) [37]. The Ni₆ NC was synthesized as a single species (Fig. 1a) with a yield of ~50% based on Ni. The SC-XRD analysis clearly revealed the atomic structure of the Ni₆(SEtPh)₁₂ NC, which is composed of six hexagonally arranged Ni atoms and twelve thiolate ligands located at intervals of the Ni ring (Fig. 1b). The atomic structure is consistent with that reported previously [37]. The Ni₆ NC showed exceptional stability owing to its protecting ligands and electron delocalization following the ring structure [28,29].

The synthesized Ni₆ NCs were further characterized using ESI mass spectrometry and UV-vis absorption spectroscopy. The single peak in the ESI mass spectrum (Fig. 1c) indicates that the obtained NC exists with molecular purity. The peak observed at m/z of 2034 Da was assigned to the NC-chloride adduct of [Ni₆(SEtPh)₁₂]Cl⁻ formed during ionization, by matching the simulated isotope pattern (Fig. 1c, inset). The UV-vis absorption spectrum of the Ni₆ NC (Fig. 1d) exhibits four characteristic absorption peaks at 339, 410, and 535 nm, indicating that the Ni₆ NC has a discrete energy level.

The Ni₆ NCs are very small (*ca.* 1.6 nm in diameter, including the ligand chain) and stable; therefore, they can be directly immobilized on NF electrodes

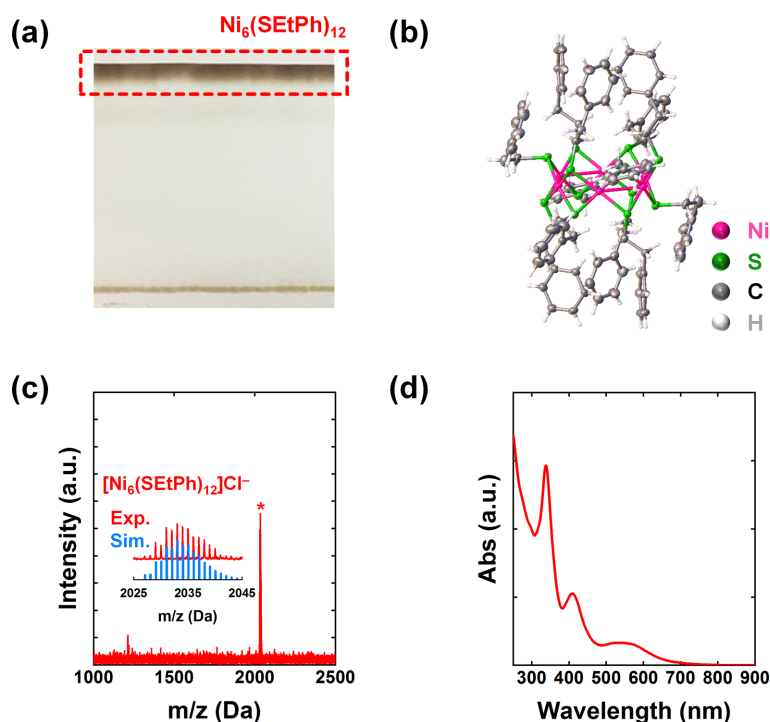


Fig. 1. (a) Digital photograph of the preparative thin layer chromatography separation of the Ni₆ NC products. (b) Crystal structure of the Ni₆(SETPh)₁₂ NC. (c) Negative-mode ESI mass and (d) UV-vis absorption spectrum of the Ni₆(SETPh)₁₂ NC. The inset in (c) compares the experimental data (red) and the simulated isotope patterns (blue).

even without support such as carbon black. The dip-coating process is considered an easily processable catalyst coating method [47,48]. Ni₆/NF was prepared by simply dipping the NF into a Ni₆ dipping solution (Fig. 2a). Methanol, in which the Ni₆ NC is poorly soluble, was chosen as the suspension medium to make Ni₆ NC weakly interact with the solvent and thereby anchor on NF. The use of a binder, either PVDF or Nafion, assisted the dispersion of the Ni₆ NCs (Fig. S2), resulting in a uniform suspension. Despite short dipping time, the loading amount was found to be consistent in the reproducible test (Fig. S3).

To confirm the distribution of Ni₆ NCs on the NF, we performed the SEM and corresponding EDS of the Ni₆/NF. The existence of the Ni₆ NC can be tracked by the sulfur atoms of the thiolate ligands in the EDS elemental mapping image. Therefore, we used PVDF as a binder instead of Nafion for the SEM-EDS measurements to avoid interference from sulfur of Nafion. SEM images show some aggregates of Ni₆ NCs on the NF substrate (Figs. 2b and S4a).

However, the SEM-EDS elemental mapping images of Ni₆/NF show that the Ni₆ NCs were uniformly distributed throughout the NF, as monitored by the sulfur atoms (Figs. 2b and S5).

The OER activity of Ni₆/NF was measured by LSV in a 1.0 M KOH electrolyte solution. Before measuring the OER activity, the electrodes were pretreated by conducting CP at an anodic current of 100 mA/cm² for 2 h. A significant increase in the oxidative current was observed after CP (Fig. S6a), suggesting that Ni₆/NF underwent some kind of activation process during CP. After CP electrolysis, Ni₆/NF displayed a clean surface without NC aggregates (Fig. S4b). Extracting NCs from Ni₆/NF after electrolysis, the electrode exhibited OER activity similar to that of bare NF (Fig. S6b), and the retrieved NCs showed nearly identical profiles to the intact NCs (Fig. S7). These results suggest that the activation can be attributed to the uniform spreading of Ni₆ NCs on the NF substrate during CP electrolysis.

As shown in Fig. 2c, Ni₆/NF showed a higher OER activity than bare NF. The OER current on Ni₆/NF

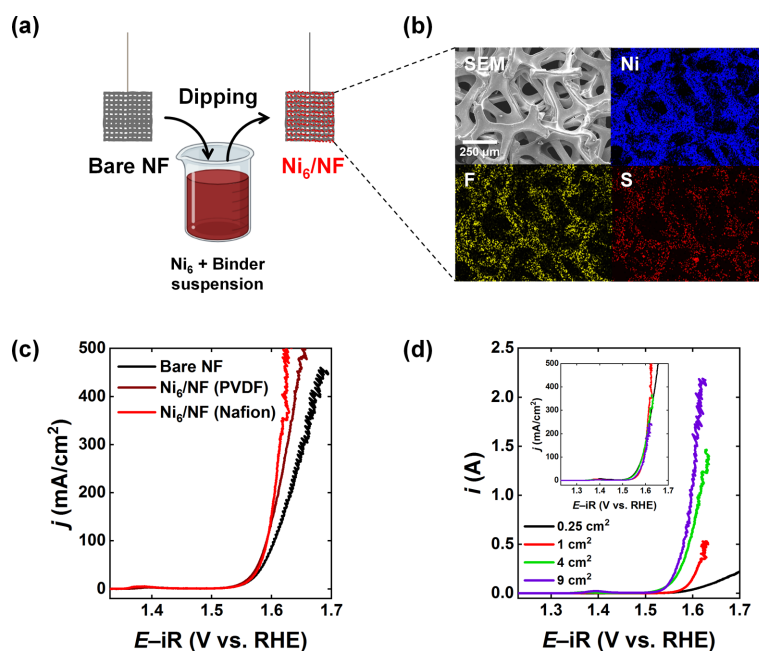


Fig. 2. (a) Scheme describing the dip-coating process of Ni₆ NCs onto NF. (b) SEM and SEM-EDS elemental mapping images of Ni₆/NF fabricated with PVDF binder. LSV curves recorded (c) on bare NF, Ni₆/NF (PVDF), and Ni₆/NF (Nafion) electrodes, and (d) on various sized Ni₆/NF (Nafion) electrodes.

started from an η of 0.30 V and reached 500 mA/cm² at a low η of 0.39 V. The OER activity of Ni₆/NF was slightly higher with the Nafion binder than with the PVDF binder because of the enhanced ion conductivity of the electrode [49]. Thus, OER experiments were conducted on Ni₆/NF with a Nafion binder. The OER activity of Ni₆/NF increased with increasing Ni₆ loading (Fig. S8a), showing the highest activity at a loading of 0.1 mg/cm². Both η values for achieving 10 and 100 mA/cm² were decreased as increasing the loading amount (Fig. S8b).

The high processability of the dipping method enables large-scale preparation of Ni₆/NF. Fig. 2d shows the OER activity of various sized Ni₆/NF. Evidently, the total oxidative current increases with increasing electrode area. The dip-processed 9 cm² of Ni₆/NF shows an OER current of 2.0 A at an η of 0.38 V. Normalized to the area, the *j*-*E* profile of the electrodes was nearly identical (Fig. 2d, inset), indicating that the unit activity is similar for different-sized electrodes. This suggests that large-area Ni₆/NF electrodes can be readily fabricated by scaling up the dipping process.

Though Ni NCs have been exploited as OER elec-

trocatalysts in previous reports [30-32], they were supported by carbon substrates. However, carbonaceous materials easily corrode under electro-oxidation conditions [34]. Therefore, Ni₆/NF is a promising electrode for sustainable OER. To compare the effect of the carbon substrate, a carbon-based GDE was used as the substrate for the Ni₆ and IrO₂ catalysts. IrO₂/GDE exhibited a lower η (0.27 V) for the onset of the catalytic current than Ni₆/NF; however, its OER activity was lower than that of Ni₆/NF over 100 mA/cm² (Fig. 3a). Surprisingly, Ni₆/GDE showed extremely low catalytic activity compared to Ni₆/NF and IrO₂/GDE (Fig. 3a). The OER activity obtained for Ni₆/GDE was in a similar range to those obtained in other studies [30-33].

To investigate the origin of the difference in OER activities, we quantitatively measured the FE_{O₂} on the electrodes (Fig. 3b) using GC analysis. IrO₂/GDE produced a FE_{O₂} of 98% despite using a carbon substrate. This is attributed to the high loading (4.8 mg) of the IrO₂ catalyst, which can prevent the electro-oxidation of GDE. However, the FE_{O₂} measured for Ni₆/GDE was significantly low (82%), indicating that the GDE was actively corroded during electrochemi-

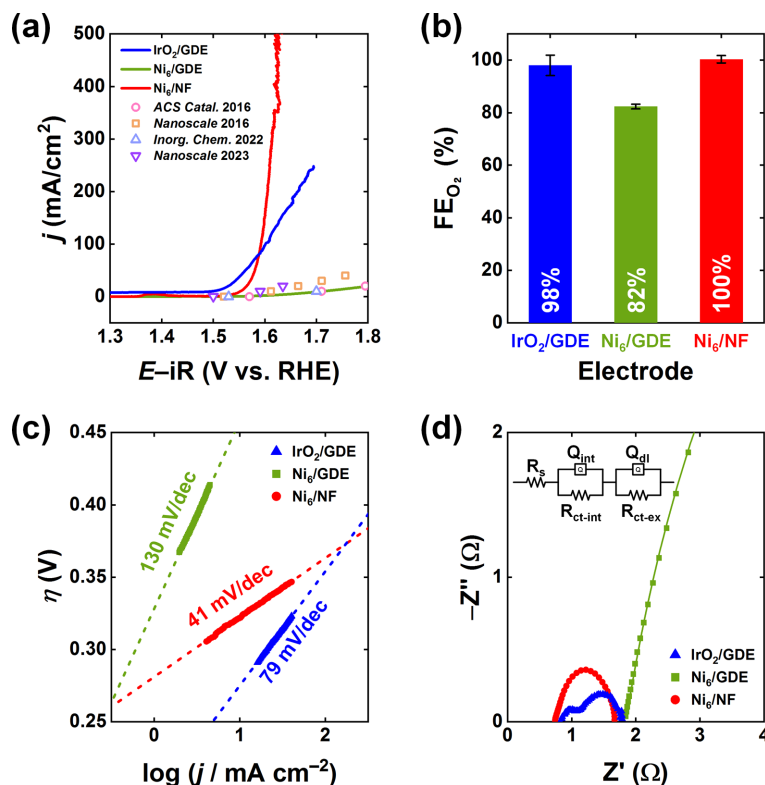
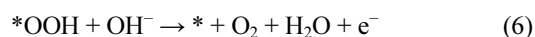
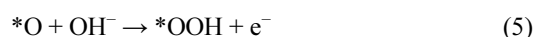
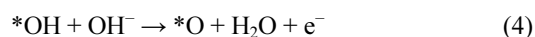


Fig. 3. (a) LSV curves, (b) FE for O₂ production at current density of 50 mA/cm², (c) Tafel, and (d) Nyquist plots at 1.64 V vs. RHE measured on Ni₆/NF (0.1 mg of catalyst), Ni₆/GDE (0.5 mg), and IrO₂/GDE (4.8 mg) in 1.0 M KOH solution. The current densities obtained on Ni₆ NCs in other reports are shown for comparison in (a). The equivalent electrical circuit [55] used to obtain the impedance parameters is shown in the inset of (d), where R_s, R_{ct-int}, and R_{ct-ex} denote the solution, internal and external charge transfer resistances, respectively, and Q_{int} and Q_{dl} are the constant phase element representing the internal and double layer capacitance.

cal OER (Fig. S9). In contrast, Ni₆/NF only performs OER with FE_{O₂} of 100%, indicating that all the produced electrons originate from water oxidation without any side reactions. This result suggests that corrosion-free electrodes can lead to highly active OER activities.

To further investigate the origin of the high OER activity of Ni₆/NF, electrokinetic studies were performed. First, Tafel analysis was performed by plotting log *j* vs. η in the kinetically controlled region. In Fig. 3c, IrO₂/GDE shows a Tafel slope of 79 mV/dec, which is consistent with the reported values [50-52]. The Tafel slope obtained for Ni₆/GDE is high (130 mV/dec) owing to carbon degradation. In contrast, the slope of Ni₆/NF was very low at 41 mV/dec, indicating that Ni₆/NF can initiate the OER with fast kinetics. In particular, the OER in a neutral-alkaline

solution involves four concerted proton-electron transfers described by the following elementary steps (here, * indicates the active site) [53]:



The theoretical Tafel slopes were 120 and 40 mV/dec when steps (3) and (4) were the rate-determining steps (RDSs), respectively [54]. Thus, the low Tafel slope of ~40 mV/dec observed for Ni₆ NC suggests that step (4) is the RDS of the OER on the Ni₆/NF

electrode. This implies that the first hydroxide adsorption is facilitated on the Ni₆/NF with fast electron transfer. This result is in good agreement with previously reported theoretical predictions for the OER on Ni₆ NC [30].

EIS was performed to compare the electron transfer kinetics of the electrodes. Fig. 3d shows the Nyquist plots obtained on IrO₂/GDE, Ni₆/GDE, and Ni₆/NF at 1.64 V in 1.0 M KOH. IrO₂/GDE shows two semicircles; the small semicircle in the high-frequency region represents the internal charge transfer of the electrode (R_{ct-int}), and the second semicircle in the low-frequency region is accompanied by external charge transfer between the electrode and reactants (R_{ct-ex}) [55]. R_{ct-int} was absent on the Ni₆/GDE and Ni₆/NF electrodes. The electron transfer rate constant (k_{ET}) [56] extracted from the sum of ($R_{ct-int}+R_{ct-ex}$) is shown in Fig. S10. Ni₆/NF exhibits the largest k_{ET} magnitude compared to IrO₂/GDE and Ni₆/GDE, indicating that electron transfer from the reactants to Ni₆/

NF is facile. These electrokinetic results strongly corroborate the extraordinary OER activity of Ni₆/NF.

To evaluate the feasibility of Ni₆/NF in CO₂ electrolysis, the Ni₆/NF anode was coupled with a Au₂₅/GDE cathode in a zero-gap CO₂ electrolyzer (see the Experimental Section for details). Ni₆/NF exhibits higher cell activity than that of the IrO₂/GDE (Fig. 4a). The cell potential obtained using the Ni₆/NF anode was approximately 0.4 V smaller than that of IrO₂/GDE at a current density of 100 mA/cm². We performed CP experiments and GC analyses to confirm the origin of the cathodic currents. Fig. 4b shows the partial current densities and selectivities for CO and H₂ measured using the zero-gap CO₂ electrolyzer. Regardless of the anodes, Au₂₅/GDE exhibits high selectivity for CO production; however, the cell potentials were reduced by more than 0.4 V with Ni₆/NF due to reduced OER overpotentials. Furthermore, a long-term stability test of Ni₆/NF was conducted at 100 mA/cm² (Fig. 4c). During the initial

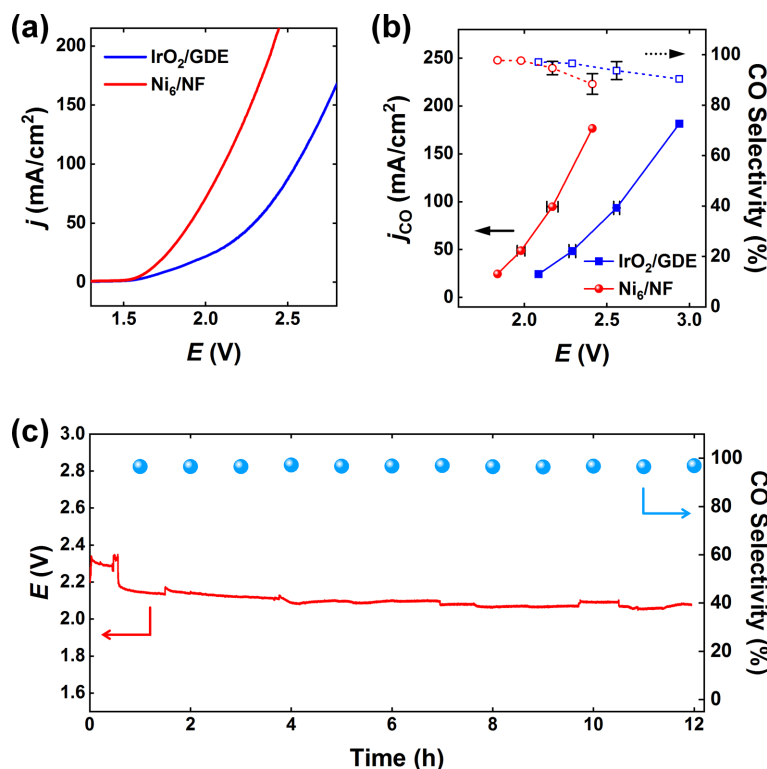


Fig. 4. (a) LSV curves, (b) j_{CO} , and CO selectivity measured on IrO₂/GDE and Ni₆/NF anodes in zero-gap CO₂ electrolyzer. Au₂₅/GDE (0.08 mg/cm²) was used as cathode. (c) Long-term CO₂ electrolysis of Au₂₅/GDE-Ni₆/NF equipped zero-gap CO₂ electrolyzer monitored by cell potential and CO selectivity at 100 mA/cm².

operation (~2 h), the cell potential gradually decreased owing to the activation of the electrodes. The electrolyzer then sustained its reactions at a cell potential of ~2.0 V for 12 h, maintaining a CO selectivity above 95%. The energy efficiency of CO₂-to-CO conversion at 100 mA/cm² is above 63%, which is higher than those reported for the most efficient electrocatalysts to the best of our knowledge [57-60].

4. Conclusions

We demonstrated that the dip-processed Ni₆/NF electrode exhibits exceptional OER activity in alkaline media. The Ni₆/NF electrode was fabricated by simply dipping NF into the NC suspension. The Ni₆/NF electrode showed higher OER activity than those of IrO₂/GDE and Ni₆/GDE without corrosion under oxidative conditions. Tafel analysis and EIS revealed that fast electron transfer on Ni₆/NF resulted in enhanced kinetics during the OER. The zero-gap CO₂ electrolyzer with the Ni₆/NF anode was stably operated at a current density of 100 mA/cm² for 12 h while maintaining a cell potential of ~2.0 V and CO selectivity above 95%. This study provides a useful methodology for the facile and uniform preparation of NC-catalyst-coated electrodes.

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Supporting Information

Supporting Information is available at <https://doi.org/10.33961/jecst.2023.00024>

References

- [1] M. S. Dresselhaus and I. L. Thomas, *Nature*, **2001**, *414*, 332-337.
- [2] P. De Luna, C. Hahn, D. Higgins, S. A. Jaffer, T. F. Jaramillo and E. H. Sargent, *Science*, **2019**, *364*, eaav3506.
- [3] W. H. Lee, H. N. Nong, C. H. Choi, K. H. Chae, Y. J. Hwang, B. K. Min, P. Strasser and H.-S. Oh, *Appl. Catal. B*, **2020**, *269*, 118820.
- [4] E. Fabbri and T. J. Schmidt, *ACS Catal.*, **2018**, *8*(10), 9765-9774.
- [5] A. K. Niaz and H.-T. Lim, *J. Electrochem. Sci. Technol.*, **2022**, *13*(3), 369-376.
- [6] Y. Lee, J. Suntivich, K. J. May, E. E. Perry and Y. Shao-Horn, *J. Phys. Chem. Lett.*, **2012**, *3*(3), 399-404.
- [7] T. Reier, Z. Pawolek, S. Cherevko, M. Bruns, T. Jones, D. Teschner, S. Selve, A. Bergmann, H. N. Nong, R. Schlögl, K. J. J. Mayrhofer and P. Strasser, *J. Am. Chem. Soc.*, **2015**, *137*(40), 13031-13040.
- [8] Z. Chen, M. Ju, M. Sun, L. Jin, R. Cai, Z. Wang, L. Dong, L. Peng, X. Long, B. Huang and S. Yang, *Angew. Chem. Int. Ed.*, **2021**, *60*(17), 9699-9705.
- [9] P. Chauhan and B. Lal, *J. Electrochem. Sci. Technol.*, **2022**, *13*(4), 497-503.
- [10] L.-F. Li, Y.-F. Li and Z.-P. Liu, *ACS Catal.* **2020**, *10*(4), 2581-2590.
- [11] T. Kou, S. Wang, J. L. Hauser, M. Chen, S. R. J. Oliver, Y. Ye, J. Guo and Y. Li, *ACS Energy Lett.*, **2019**, *4*(3), 622-628.
- [12] C. Cui, L. Gan, M. Heggen, S. Rudi and P. Strasser, *Nature Mater.*, **2013**, *12*, 765-771.
- [13] J. Huang, N. Hörmann, E. Oveisi, A. Loidjice, G. L. De Gregorio, O. Andreussi, N. Marzari and R. Buonsanti, *Nat. Commun.*, **2018**, *9*, 3117.
- [14] D. Wang, Q. Li, C. Han, Q. Lu, Z. Xing and X. Yang, *Nat. Commun.*, **2019**, *10*, 3899.
- [15] Z. Cai, L. Li, Y. Zhang, Z. Yang, J. Yang, Y. Guo and L. Guo, *Angew. Chem. Int. Ed.*, **2019**, *58*(13), 4189-4194.
- [16] F. Qin, Z. Zhao, M. K. Alam, Y. Ni, F. Robles-Hernandez, L. Yu, S. Chen, Z. Ren, Z. Wang and J. Bao, *ACS Energy Lett.*, **2018**, *3*(3), 546-554.
- [17] K. Kwak, U. P. Azad, W. Choi, K. Pyo, M. Jang and D. Lee, *ChemElectroChem*, **2016**, *3*(8), 1253-1260.
- [18] K. Kwak, W. Choi, Q. Tang, M. Kim, Y. Lee, D. Jiang and D. Lee, *Nat. Commun.*, **2017**, *8*, 14723.
- [19] H. Seong, V. Efremov, G. Park, H. Kim, J. S. Yoo and D. Lee, *Angew. Chem. Int. Ed.*, **2021**, *60*(26), 14563-14570.
- [20] W. Choi, H. Seong, V. Efremov, Y. Lee, S. Im, D.-H. Lim, J. S. Yoo and D. Lee, *J. Chem. Phys.*, **2021**, *155*, 014305.
- [21] M. Kim, Q. Tang, A. V. Narendra Kumar, K. Kwak, W. Choi, D. Jiang and D. Lee, *J. Phys. Chem. Lett.*, **2018**, *9*(5), 982-989.
- [22] V. D. Thanthirige, M. Kim, W. Choi, K. Kwak, D. Lee and G. Ramakrishna, *J. Phys. Chem. C*, **2016**, *120*(40), 23180-23188.
- [23] D. C. Lim, J. H. Jeong, K. Hong, S. Nho, J.-Y. Lee, Q. V. Hoang, S. K. Lee, K. Pyo, D. Lee and S. Cho, *Prog. Photovolt.*, **2018**, *26*(3), 188-195.
- [24] R. Jin, C. Zeng, M. Zhou and Y. Chen, *Chem. Rev.*, **2016**, *116*(18), 10346-10413.
- [25] P. Woodward, L. F. Dahl, E. W. Abel and B. C. Crosse, *J. Am. Chem. Soc.*, **1965**, *87*(22), 5251-5253.
- [26] C. Zhang, T. Matsumoto, M. Samoc, S. Petrie, S. Meng,

- T. Christopher Corkery, R. Stranger, J. Zhang, M. G. Humphrey and K. Tatsumi, *Angew. Chem. Int. Ed.*, **2010**, *49*(25), 4209-4212.
- [27] W. Zhang, J. Hong, J. Zheng, Z. Huang, J. Zhou and R. Xu, *J. Am. Chem. Soc.*, **2011**, *133*(51), 20680-20683.
- [28] A. Datta, N. S. John, G. U. Kulkarni and S. K. Pati, *J. Phys. Chem. A*, **2005**, *109*(51), 11647-11649.
- [29] A. Muñoz-Castro, *J. Phys. Chem. A*, **2011**, *115*(39), 10789-10794.
- [30] D. R. Kauffman, D. Alfonso, D. N. Tafen, J. Lekse, C. Wang, X. Deng, J. Lee, H. Jang, J. Lee, S. Kumar and C. Matranga, *ACS Catal.*, **2016**, *6*(2), 1225-1234.
- [31] K. S. Joya, L. Sinatra, L. G. AbdulHalim, C. P. Joshi, M. N. Hedhili, O. M. Bakr and I. Hussain, *Nanoscale*, **2016**, *8*, 9695-9703.
- [32] S. Srinivasan, Z. Liu, S. House and R. Jin, *Inorg. Chem.*, **2023**, *62*(5), 1875-1884.
- [33] S. Funaki, T. Kawawaki, T. Okada, K. Takemae, S. Hossain, Y. Niihori, T. Naito, M. Takagi, T. Shimazaki, S. Kikkawa, S. Yamazoe, M. Tachikawa and Y. Negishi, *Nanoscale*, **2023**, *15*, 5201-5208.
- [34] S. Möller, S. Barwe, J. Masa, D. Wintrich, S. Seisel, H. Baltruschat and W. Schuhmann, *Angew. Chem. Int. Ed.*, **2020**, *59*(4), 1585-1589.
- [35] Y. Yi, G. Weinberg, M. Prenzel, M. Greiner, S. Heumann, S. Becker and R. Schlögl, *Catal. Today*, **2017**, *295*, 32-40.
- [36] Y. Sato, N. Yamada, S. Kitano, D. Kowalski, Y. Aoki and H. Habazaki, *J. Mater. Chem. A*, **2022**, *10*, 8208-8217.
- [37] H. N. Kagalwala, E. Gottlieb, G. Li, T. Li, R. Jin and S. Bernhard, *Inorg. Chem.*, **2013**, *52*(15), 9094-9101.
- [38] B. Kim, H. Seong, J. T. Song, K. Kwak, H. Song, Y. C. Tan, G. Park, D. Lee and J. Oh, *ACS Energy Lett.*, **2020**, *5*(3), 749-757.
- [39] J. W. Shin, K. Eom and D. Moon, *J. Synchrotron Rad.*, **2016**, *23*(1), 369-373.
- [40] Z. Otwinowski and W. Minor, Processing of X-ray diffraction data collected in oscillation mode, in *Methods in Enzymology*, Academic Press, **1997**, *276*, 307-326.
- [41] G. M. Sheldrick, *Acta Cryst.*, **2015**, *A71*, 3-8.
- [42] G. M. Sheldrick, *Acta Cryst.*, **2015**, *C71*, 3-8.
- [43] O. V. Dolomanov, L. J. Bourhis, R. J. Gildea, J. A. K. Howard and H. Puschmann, *J. Appl. Crystallogr.*, **2009**, *42*(2), 339-341.
- [44] P. V. D. Sluis and A. L. Spek, *Acta Cryst.*, **1990**, *A46*, 194-201.
- [45] A. L. Spek, *Acta Cryst.*, **2015**, *C71*, 9-18.
- [46] H. Seong, M. Choi, S. Park, H. Kim, J. Kim, W. Kim, J. S. Yoo and D. Lee, *ACS Energy Lett.*, **2022**, *7*(12), 4177-4184.
- [47] J. Ge, H.-B. Yao, W. Hu, X.-F. Yu, Y.-X. Yan, L.-B. Mao, H.-H. Li, S.-S. Li and S.-H. Yu, *Nano Energy*, **2013**, *2*(4), 505-513.
- [48] X. Du, Z. Chen, Z. Li, H. Hao, Q. Zeng, C. Dong and B. Yang, *Adv. Energy Mater.*, **2014**, *4*(9), 1400135.
- [49] G.-F. Li, D. Yang and P.-Y. A. Chuang, *ACS Catal.*, **2018**, *8*(12), 11688-11698.
- [50] Y. Zhu, W. Zhou, Y. Zhong, Y. Bu, X. Chen, Q. Zhong, M. Liu and Z. Shao, *Adv. Energy Mater.*, **2017**, *7*(8), 1602122.
- [51] Z. Song, K. Wang, Q. Sun, L. Zhang, J. Li, D. Li, P.-W. Sze, Y. Liang, X. Sun, X.-Z. Fu and J.-L. Luo, *Adv. Sci.*, **2021**, *8*(14), 2100498.
- [52] W. Zhong, Z. Lin, S. Feng, D. Wang, S. Shen, Q. Zhang, L. Gu, Z. Wang and B. Fang, *Nanoscale*, **2019**, *11*, 4407-4413.
- [53] C. Hu, L. Zhang and J. Gong, *Energy Environ. Sci.*, **2019**, *12*, 2620-2645.
- [54] S. Fletcher, *J. Solid State Electrochem.*, **2009**, *13*, 537-549.
- [55] J. Jiang, F. Sun, S. Zhou, W. Hu, H. Zhang, J. Dong, Z. Jiang, J. Zhao, J. Li, W. Yan and M. Wang, *Nat. Commun.*, **2018**, *9*, 2885.
- [56] H. Chen, P. Kannan, L. Guo, H. Chen and D.-H. Kim, *J. Mater. Chem.*, **2011**, *21*, 18271-18278.
- [57] S. Ren, D. Joulié, D. Salvatore, K. Torbensen, M. Wang, M. Robert and C. P. Berlinguette, *Science*, **2019**, *365*(6451), 367-369.
- [58] W. H. Lee, Y.-J. Ko, Y. Choi, S. Y. Lee, C. H. Choi, Y. J. Hwang, B. K. Min, P. Strasser and H.-S. Oh, *Nano Energy*, **2020**, *76*, 105030.
- [59] W. Ren, X. Tan, C. Jia, A. Krammer, Q. Sun, J. Qu, S. C. Smith, A. Schueler, X. Hu and C. Zhao, *Angew. Chem. Int. Ed.*, **2022**, *61*(26), e202203335.
- [60] X. Y. Zhang, W. J. Li, J. Chen, X. F. Wu, Y. W. Liu, F. Mao, H. Y. Yuan, M. Zhu, S. Dai, H. F. Wang, P. Hu, C. Sun, P. F. Liu and H. G. Yang, *Angew. Chem. Int. Ed.*, **2022**, *61*(28), e202202298.