Electrochemical Characteristics of Synthesized Nb₂O₅-Li₃VO₄ Composites as Li Storage Materials

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The increasing demand for energy storage in mobile electronic devices and electric vehicles has emphasized the importance of electrochemical energy storage devices such as Li-ion batteries (LIBs) and supercapacitors. For reversible Li storage, alternative anode materials are actively being developed. In this study, we designed and fabricated an Nb₂O₅-Li₃VO₄ composite for use as an anode material in LIBs and hybrid supercapacitors. Nb₂O₅ powders were dissolved into a solution and the precursors were precipitated onto Li₃VO₄ through a simple, low-temperature hydrothermal reaction. The annealing process yielded an Nb₂O₅-Li₃VO₄ composite that was characterized by X-ray diffraction, electron microscopy, and X-ray photoelectron spectroscopy. Electrochemical tests revealed that the Nb₂O₅-Li₃VO₄ composite electrode demonstrated increased capacities of approximately 350 and 140 mAh g⁻¹ at 0.1 and 5 C, respectively, were maintained up to 1000 cycles. The reversible capacity and rate capability of the composite electrode were enhanced compared to those of pure Nb₂O₅-based electrodes. These results can be attributed to the microstructure design of the synthesized composite material.

Keywords: Niobium pentoxide, Li_3VO_4 , Lithium storage, Electrode, Electrochemical properties

1. Introduction

Recently, the importance of electrochemical energy storage devices is increasing with the rapid development of mobile electronic devices and electric vehicles. Among these devices, Li-ion batteries (LIBs) are predominantly used in the current market because of their high energy density, long cycle life, and reasonable cost [1-3]. Carbonbased materials such as graphite are commercially used as anode (negative electrode) materials in LIBs. Features such as moderate capacity (372 mAh g^{-1} for Li₆C phases), low reaction potential, and great cycle performance make graphite suitable for commercial LIBs [1,4]. However, current commercial applications of LIBs require them to have several other properties. Therefore, researchers have been actively pursuing alternative anode materials to replace graphite. In order to increase specific capacity, Li alloy-based materials such as Si and Sn have been widely investigated [5]. Furthermore, alternative intercalationbased materials such as Li₄Ti₅O₁₂, TiO₂, and Nb₂O₅ have been examined to enhance the rate capability (fast charging and high power) and safety features [6-8].

These materials exhibit relatively good rate performance; therefore, they can also be used in hybrid supercapacitors and Li-ion capacitors (LICs), wherein the aforementioned battery-type materials are employed for anodes and activated carbon materials are used for cathodes. Both LIB and LIC applications require intercalation-type materials to further improve the capacity and rate capability. Of these materials, niobium pentoxide (Nb₂O₅) materials have recently attracted significant attention as high-rate anodes because they are naturally abundant and can provide moderate capacity (~200 mAh g⁻¹) as well as excellent rate performance through a pseudocapacitive intercalation mechanism [9]. However, the current market demands increased reversible capacity while maintaining the rate performance.

In this study, we designed and synthesized an Nb₂O₅-Li₃VO₄ composite material to enhance the reversible capacity for electrochemical Li storage. Li₃VO₄ has garnered great interest as an intercalation-based anode material [10]. Its capacity of 397 mAh g⁻¹ (Li_{3+x}VO₄, x = 2) is relatively higher than those of graphite and Nb₂O₅. Another benefit of this material is the high mobility of Li

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ions; therefore, Nb₂O₅ was designed to be dispersed within the Li₃VO₄ matrix. The composite was synthesized by first dissolving the Nb₂O₅ powders into a solution and then precipitating the precursors onto Li₃VO₄ through a simple, low-temperature hydrothermal reaction. The resulting Nb₂O₅-Li₃VO₄ composite was characterized using several analytical tools. Electrochemical tests revealed that the composite electrode exhibits an increased capacity of approximately 350 mAh g⁻¹ at 0.1 C, which is higher than that of Nb₂O₅ materials. Moreover, the capacity of about 140 mAh g⁻¹ was maintained well without fading for up to 1000 cycles at a high rate of 5 C.

2. Experimental Methods

2.1 Material synthesis

 Nb_2O_5 powder was purchased from Sigma-Aldrich (99.9%, -325 mesh) and used as received. First, 1.0 g the Nb_2O_5 powder was dissolved into a mixed solution of ethyl alcohol (60 mL) and distilled water (60 mL). Subsequently, 8 mM NH₄VO₃ (Sigma-Aldrich, 99%) and 24 mM LiOH·H₂O (Samchun, 98%) were dissolved in the solution. The solutions were placed in a Teflon-lined steel autoclave, heated to 120 °C and maintained at that temperature for 12 h to induce hydrothermal reaction. After several washes with distilled water and ethyl alcohol, the precipitates were dried and annealed at 400 °C for 2 h in an electrical furnace under pure nitrogen atmosphere. Subsequently, the furnace was automatically cooled down to room temperature.

2.2 Materials characterization

Crystalline structures of the prepared materials were identified through X-ray diffraction (XRD) using the Rigaku D/MAX-2500 with Cu K α radiation. Field-emission scanning electron microscopy (FE-SEM, JEOL 7500) and highresolution transmission electron microscopy (HR-TEM, JEOL ARM-200F) with an energy dispersive spectroscopy (EDS) detector were employed to characterize the morphologies, crystallinity, and elemental distribution of the synthesized samples. X-ray photoelectron spectroscopy (XPS, Thermo Scientific K α) was adopted to determine the surface chemical states of the samples.

2.3 Electrochemical measurements

The working electrodes were prepared by coating a slurry

comprising the synthesized active material (80 wt%), a conducting agent (10 wt%, Super P, Sigma-Aldrich), and a binder (10 wt%, polyacrylic acid, Sigma-Aldrich) onto Cu foil current collectors and dried at 80 °C for 3 h. Then, the electrode was pressed by a roll-presser and vacuumdried for 12 h. Subsequently, the electrode was cut into a disc with a diameter of 10 mm and used as a working electrode. Li metal foil was employed as the counter/ reference electrode, and the electrolyte comprised 1.0 M LiPF₆ in ethylene carbonate and dimethyl carbonate (3:7 by volume ratio) along with 10 wt% fluoroethylene carbonate; porous polyethylene separators were also used. Coin-type 2032 half-cells were assembled in an Ar-filled glove box to characterize the electrochemical properties. Cycling experiments were galvanostatically performed using a battery tester (Basytec CTS-Lab) in the potential range of 0.01-3.0 V (vs. Li⁺/Li) at 25 °C . Rate capability was measured at various specific currents under the same conditions.

3. Results and Discussion

Fig. 1 displays the XRD patterns of the synthesized material with references. All observed diffraction peaks could be indexed to two phases of orthorhombic Li_3VO_4 (JCPDS No. 38-1247) and orthorhombic Nb_2O_5 (JCPDS No. 27-1402). This indicates that the Li_3VO_4 phase was synthesized without impurities through hydrothermal reaction and annealing. The composite is expected to



Fig. 1. XRD patterns of the synthesized material along with references



Fig. 2. (a,b) FE-SEM images and (c) the corresponding EDS elemental mapping results of the Nb_2O_5 - Li_3VO_4 composite

comprise starting Nb₂O₅ and synthesized Li₃VO₄. The morphology of the sample was examined using FE-SEM. Fig. 2a and b display the FE-SEM images of the Nb₂O₅-Li₃VO₄ composite. The spherically shaped particles measuring a few microns were observed to be agglomerated. The corresponding EDS elemental mapping results are shown in Fig. 2c. The signals for Nb and V were inversely detected, indicating that primary Nb₂O₅ and Li₃VO₄ particles are mixed in the composite.

Fig. 3 presents the TEM analysis results for the composite. Fig. 3a and b display the low- and highmagnification TEM images of the composite particles, which were enlarged to obtain several HR-TEM images. Two typical HR-TEM images along with fast Fourier transform (FFT) patterns (inset) are shown in Fig. 3c and d. Well-crystallized Nb₂O₅ domains were observed and the corresponding d-spacing of 3.14 Å could be attributed to the (180) reflection plane of orthorhombic Nb₂O₅ (Fig. 3c), which was confirmed in the FFT pattern [11]. Fig. 3d displays another HR-TEM image wherein a crystalline domain was observed with a different d-spacing of 4.13 Å, which could be matched with the (110) reflection of orthorhombic Li₃VO₄ [12]. The inset FFT pattern also confirmed the existence of the Li₃VO₄ phase. The TEM image shown in Fig. 3a was analyzed using EDS elemental mapping, the results of which are shown in Fig. 3e. The O element was observed in the entire particle composite, whereas Nb and V elements were inversely located. This observation corresponds well to the EDS



Fig. 3. (a, b) TEM images, (c, d) HR-TEM images (inset: FFT pattern), and (e) EDS elemental mapping results of the Nb_2O_5 -Li₃VO₄ composite

elemental mapping results in the FE-SEM image (Fig. 2c).

The chemical states of the composites were examined using XPS. The surface of the sample was cleaned using Ar-ion etching for 300 s. Fig. 4a displays the XPS V 2p core-level spectrum for the Nb₂O₅-Li₃VO₄ composite with peaks corresponding to the 2p1/2 and 2p3/2 orbitals of V. The peaks were deconvoluted into two major sub-profiles centered at 524.7 and 516.8 eV that represented V in its pentavalent state (V^{5+}) with its oxidation number corresponding to the Li₃VO₄ phase [13,14]. Small subprofiles at 523.0 and 516.8 eV represented the tetravalent state of $V(V^{4+})$. These results indicated that some oxygen vacancies may still exist in the material [14]. The Nb 3d core-level spectrum of the composite sample with deconvoluted sub-profiles is shown in Fig. 4b. The binding energy distribution was similar to that of V with subprofiles centered at 210.4 and 207.6 eV representing the Nb⁵⁺ oxidation state, as previously reported [11,15]. Very small sub-profiles observed at 209.0 and 206.1 eV are ascribed to Nb⁴⁺ state [15]. All these features originate



Fig. 4. XPS core-level spectra of the Nb_2O_5 -Li $_3VO_4$ composite: (a) V 2p, (b) Nb 3d, and (c) O 1s

from the orthorhombic Nb_2O_5 phase in the composite. The O 1s core-level spectrum is shown in Fig. 4c. The spectrum was deconvoluted into two sub-profiles centered at 531.5 and 529.8 eV, corresponding to the O in Nb_2O_5 and Li_3VO_4 , respectively [11,13]. All these XPS analysis results revealed that the Nb_2O_5 and Li_3VO_4 phases are mixed in the composite.

Fig. 5a displays the voltage profiles for the first two cycles of the Nb₂O₅-Li₃VO₄ composite electrode that were measured at a constant specific current of 50 mA g⁻¹. The discharge (Li insertion) and charge (Li extraction) capacities were 528 and 353 mAh g⁻¹, respectively, with an initial Coulombic efficiency of 66.8%. The sloped portion between 2.0 and 1.0 V vs. Li⁺/Li during discharge could be associated with the Li insertion into the Nb₂O₅ phase, as previously reported [16,17]. Furthermore, solid electrolyte interphase (SEI) layer formation could occur in the region. Below 1.0 V, the Li can be inserted into the Li₃VO₄, as previously reported [18,19]. During charging, Li would be sequentially extracted from Li₃VO₄ and Nb₂O₅.

Fig. 5b shows the cycle performance of the Nb₂O₅-Li₃VO₄ composite electrode measured at 50 mA g⁻¹. A reversible capacity of more than 300 mAh g⁻¹ could be maintained for up to 100 cycles. This robust cycling stability can be attributed to the microstructure of the composite. The Nb₂O₅ and Li₃VO₄ particles are mixed and well attached to each other in the composite, which can buffer the volume change of both materials during Li insertion and extraction cycling. Long-term cycle performance at a high rate of 5 C (1.5 A g^{-1} , 1 C = 300 mAh g^{-1}) is shown in Fig. 5c. In the first stage, the reversible capacity increased gradually with an increase in cycle number. This phenomenon can be related to the activation of the electrode materials because the cycling current was very high. After 100 cycles, the capacity reached approximately 140 mAh g⁻¹ and was maintained well for up to 1000 cycles. In our previous study [20], a carbon-coated Nb₂O₅ nanoparticle electrode exhibited a reversible capacity of approximately 100 mAh g⁻¹ after 1000 cycles at a rate of 5 C (1.0 A g^{-1} , 1 C = 200 mAh g^{-1} ¹ for Nb₂O₅). In comparison, the Nb₂O₅-Li₃VO₄ composite electrode fabricated here demonstrated a much higher capacity at a high specific current (1.5 A g^{-1}). This result could be attributed to fast Li ion transport caused by the high Li ion conduction in the Li_3VO_4 phase. Moreover, the Li₃VO₄ phase can store more Li ions through insertion reactions.



Fig. 5. Electrochemical properties of the Nb₂O₅-Li₃VO₄ composite electrode: (a) voltage profiles at 50 mAh g^{-1} , (b) cycle performance at 50 mAh g^{-1} , (c) long-term cycle performance at 1.5 A g^{-1} (5 C), and (d) rate performance

Fig. 5d presents the rate performance of the Nb₂O₅-Li₃VO₄ composite electrode at varying rates of 0.1–5 C. The electrode exhibited a reversible capacity of about 350 and 210 mAh g⁻¹ at 0.1 and 1 C, respectively. The capacity of the composite at 1 C was higher than those of pure or C-coated Nb₂O₅ materials [20], which reveals that the synthesized composite electrode has some potential for mid- and/or high-rate Li-storage applications.

4. Conclusions

An Nb₂O₅-Li₃VO₄ composite was prepared as an anode material for LIBs and hybrid supercapacitors. Nb₂O₅ powders were first dissolved into a solution and then the precursors were precipitated onto the Li₃VO₄ through simple low-temperature hydrothermal reaction. After annealing, the Nb₂O₅-Li₃VO₄ composite was successfully synthesized. The formation of Li₃VO₄ was confirmed by XRD and XPS. The morphology was observed using FE- SEM and TEM; EDS elemental mapping results revealed the distribution of each element. The electrochemical test revealed that the Nb₂O₅-Li₃VO₄ composite electrode had an increased capacity of approximately 350 mAh g⁻¹, compared to that of Nb₂O₅, as well as good cycle and rate performance that can be ascribed to the microstructure design of the synthesized composite material.

Acknowledgments

This work was supported by the National Research Foundation of Korea (NRF) Grant funded by the Korean Government (2015R1A5A7037615 and 2019R1F1A10 62835).

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