

## Adsorption of Mercury(II) Chloride and Carbon Dioxide on Graphene/Calcium Oxide (0 0 1)

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**Abstract** In this work, recent progress on graphene/metal oxide composites as advanced materials for HgCl<sub>2</sub> and CO<sub>2</sub> capture was investigated. Density Functional Theory calculations were used to understand the effects of temperature on the adsorption ability of HgCl<sub>2</sub> and water vapor on CO<sub>2</sub> adsorption on CaO (001) with reinforced carbon-based nanostructures using B3LYP functional. Understanding the mechanism by which mercury and CO<sub>2</sub> adsorb on graphene/CaO (g-CaO) is crucial to the design and fabrication of effective capture technologies. The results obtained from the optimized geometries and frequencies of the proposed cluster site structures predicted that with respect to molecular binding the system possesses unusually large HgCl<sub>2</sub> (0.1-0.4 HgCl<sub>2</sub> g/g sorbent) and CO<sub>2</sub> (0.2-0.6 CO<sub>2</sub> g/g sorbent) uptake capacities. The HgCl<sub>2</sub> and CO<sub>2</sub> were found to be stable on the surface as a result of the topology and a strong interaction with the g-CaO system; these results strongly suggest the potential of CaO-doped carbon materials for HgCl<sub>2</sub> and CO<sub>2</sub> capture applications, the functional gives reliable answers compared to available experimental data.

**Key words** adsorption, computer modeling and simulation, desorption, nanostructures.

### 1. Introduction

Metal oxides are commonly used in industrial applications as catalysts or as support materials. Many properties of metal oxide surfaces have been studied during the past decade.<sup>1-3</sup> CaO is a metal oxide, commonly known as quicklime or burnt lime, is a widely used chemical compound. It is a white, caustic, alkaline, crystalline solid at room temperature investigated by Broqvist, et al.<sup>4</sup> Moreover; CaO can be an effective destructive adsorbent for small compounds such as fluoro-, chloro-, and bromocarbons, sulfur- and organophosphorus compounds.<sup>5</sup> Coal-fired power plants are the greatest anthropogenic source of mercury emissions.<sup>3</sup> In this work, Density Functional Theory(DFT) with cluster methods were predominantly used to simulate the CaO (001) surface. The DFT scheme has been proven to be fairly accurate.<sup>6-10</sup> The number of published papers employing CaO cluster models is quite limited, but some publications have appeared.<sup>11,12</sup> Recent work on a similar cluster

looked at NO<sub>x</sub> and other compounds being adsorbed on a relatively small CaO cluster.<sup>13</sup> Several researchers used a fixed cluster approach.<sup>11-14</sup> There are some calculations on CaO surfaces, which are for much smaller adsorbents where more elaborate methods, can be applied.<sup>11-17</sup> Nanomaterials are of great scientific interest due to their size-related unique properties and a wide variety of potential applications. However, a common phenomenon of metal oxides during preparation is uncontrolled agglomeration and growth of large particles. The aggregation nanomaterial results in the growth of nanoparticles often to sizes in excess of the nanoparticle size definition of less than a 100 nm, ultimately leading to sedimentation. The aggregated particles differ significantly from nanomaterials for which size-dependent properties are observed. This growth can be prevented by the presence of surfactants and other surface coatings. Therefore, synthesis of metal oxide nanostructures with controlled size, uniform morphology, good crystallinity and high dispersion remains a major challenge.

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Mercury is a toxic metal that accumulates in the food chain and is considered a hazardous air pollutant. Mercury has severe health effects and may cause central nervous system damage, pulmonary and renal failure, severe respiratory damage, blindness and chromosome damage. The Hg emissions from coal-fired power plants are of major environmental concern. Mercury in coal is vaporized into its gaseous elemental form throughout the coal combustion process. Elemental Hg can be oxidized in subsequent reactions with other gaseous components and solid materials in coal-fired flue gases. While oxidized Hg in coal-fired flue gases is readily controlled by its adsorption onto fly ash and/or its dissolution into existing solution-based sulfur dioxide scrubbers, elemental Hg is not controlled. The extent of elemental Hg formed during coal combustion is difficult to predict since it is dependent on the type of coal burned, combustion conditions, and existing control technologies installed. Therefore, it is important to understand heterogeneous Hg reaction mechanisms to predict the speciation of Hg emissions from coal-fired power plants to design and effectively determine the best applicable control technologies; specifically Mercuric chloride.  $\text{HgCl}_2$  is highly toxic, both acutely and as a cumulative poison. CaO can act as adsorbent of Hg-containing species which will appear in subsequent sections. Specifically,  $\text{HgCl}_2$  adsorption on the CaO (001) surface was studied by performing DFT calculations using B3LYP functionals. However, it should be pointed out that the accuracy of DFT in treating the Hg transition metal system does have limitations. The Hg system possesses spin multiplet structures that are energetically close to each other.<sup>9,10)</sup>

The most prominent greenhouse gas is carbon dioxide. Coal-fired power plants and fossil fuel combustion is also the major source of carbon dioxide emissions, there is difficulty in controlling carbon dioxide in flue gases because of the large quantities of carbon-content in fossil fuels. In order to derive novel adsorbents for removing carbon dioxide, it is important to understand the adsorption mechanisms of  $\text{CO}_2$ . Graphene is a thin layer of pure carbon; it is a single, tightly packed layer of carbon atoms that are bonded together in a hexagonal honeycomb lattice. Recent advances suggest that reduced graphene serves as a good 2-Dimensional (2D) support to nucleate metal oxide nanoparticles on the edges and surface.<sup>18)</sup> It has been demonstrated that graphene can serve as a perfect 2D support for anchoring metal or metal oxide nanoparticles.<sup>18)</sup>

The mechanism of mercury and  $\text{CO}_2$  adsorption on CaO is not exactly known and its understanding is crucial to the design and fabrication of effective capture technologies for mercury and  $\text{CO}_2$ . In this work, theoretical calculations have been performed to investigate the

adsorption of Hg and  $\text{CO}_2$  on the combined advantages of both graphene and metal oxides solid surfaces(g-CaO) for improving the  $\text{HgCl}_2$  capture. The objective of this research is to identify potential materials as multi-pollutant sorbents in power plants. This is carried out using high-level DFT electronic structure calculations to understand heterogeneous chemical pathways of Hg and  $\text{CO}_2$  as active materials. The effects of temperature on the adsorption ability of  $\text{HgCl}_2$  on CaO (001) with 2-D graphene surface support were calculated using DFT. Finally, the auxiliary aspect of this research is to investigate theoretically how carbon dioxide from flue gases of coal-fired power plants is adsorbed on an inorganic g-CaO surface in the presence of water vapor. The research uses a fundamental science-based approach to understand the environmental problems caused by coal-fired energy production and provides solutions to the power generation industry for emissions reductions.

## 2. Methodology

All calculations were performed using DFT with a B3LYP functional<sup>19)</sup> with computational cost lowered by the use of highly localized grid representations wherein each electronic wave function is expanded in a localized atom-centered basis set with each basis function defined analytically. The DFT approach can be used to calculate structural properties that are typically within 0.05 Å and 1-2° of experiment. The overall adsorption and reaction energies are typically within 5-7 kcal/mol of experimental values, and spectroscopic analyses are within a few percent of experimental data.<sup>6-8)</sup> For cluster geometries, spin-unrestricted calculations were carried out with a 6-31G\* valence double-zeta polarized basis set available. Charge densities were analyzed by the Mulliken method.<sup>20)</sup> For open-shell molecular radicals, the unrestricted formalism was used. The present level of calculation, DFT (UB3LYP)/6-31G\*, is known to produce reasonable results<sup>21)</sup> for bond lengths, bond angles, and bond energies for a wide range of molecules. The Hg relativistic corrections can be calculated,<sup>22)</sup> for the relativistic electron density near the nucleus that may be simulated using a relativistic effective core potential. It offers an adequate description of molecules containing Hg wherein the relativistic behavior of the inner electrons cannot be ignored because the standard Schrödinger equation is inadequate for these compounds. The computations were carried out using the ab initio quantum chemistry package, General Atomic and Molecular Electronic Structure System(GAMESS) code, available from Mark Gordons<sup>22)</sup> Quantum Theory group at Ames laboratory/Iowa State University. The dispersion interaction between two quantum mechanical molecules comes from the general

formula for the second-order term in intermolecular perturbation theory. A formula for the dispersion interaction between two molecules based on this functional form model was previously derived and implemented in the GAMESS package.

The Gibbs free energy for the equilibrium constant is calculated using:

$$\Delta G_{\text{ads}} \approx \Delta E_{\text{ads}} + \Delta E_0 + T(\Delta S_{\text{vib}} + \Delta S_{\text{trans,rot}}) - kT \ln(P/P_0) \quad (1)$$

where the  $\Delta G_{\text{ads}}$  is the change of Gibbs free energy for the adsorption process, the  $\Delta E_{\text{ads}}$  of mercury-containing species on the modeled structures, investigated with generalized-gradient approximation GGA/B3LYP methods were calculated through the following equation:

$$\Delta E_{\text{ads}} = E_{\text{HgCl}_2+\text{surface}} - E_{\text{HgCl}_2} - E_{\text{surface}} \quad (2)$$

where  $\Delta E_{\text{ads}}$  is the adsorption energy,  $E_{\text{HgCl}_2+\text{surface}}$  represents the energy of the adsorbate  $\text{HgCl}_2$ -surface,  $E_{\text{HgCl}_2}$  is the energy of the gas phase mercury-containing species, and  $E_{\text{surface}}$  is the energy of the isolated CaO or g-CaO surface model, in kcal/mol. The  $\Delta E_0$  is the zero-point energy change during the adsorption process, and  $\Delta S_{\text{vib}}$ ,  $\Delta S_{\text{trans,rot}}$  are the vibrational and translational, rotational entropy changes of adsorption in kcal/mol K.  $T$  is the temperature in K.  $k$  is Boltzmann's constant and the pressure terms cancel out because the pressure is constant in this adsorption system.<sup>16,17</sup> The equilibrium constants ( $K_{\text{eq}}$ ) is given as

$$K_{\text{eq}} = \exp(-\Delta G_{\text{ads}} / RT) \quad (3)$$

$R$  is the ideal gas constant in kcal/mol K. The clean perfect surface modeled by a  $5 \times 5 \times 2$  CaO (001) structure with and without graphene as 2D support was allowed to relax.

Electronic structure descriptors have been computed to analyze the geometrical and electronic changes such as the  $\Delta G$ . The above equations were used to obtain the Gibbs free energy and the equilibrium constant for different adsorption scenarios.<sup>27-33</sup> Measuring the Gibbs free energy can assess the stability of  $\text{CO}_2$  in g-CaO with water. A number of quantum mechanical continuum solvation models were developed for this purpose.<sup>23,24</sup> These originated from the Onsager continuum model,<sup>24</sup> and were formulated by Tomasi et al.<sup>23,25,26,27</sup> as the polarizable continuum model (PCM) that is a commonly used method in computational chemistry to model solvation effects. If it were necessary to consider each solvent molecule as a separate molecule, the computational cost of modeling a solvent-mediated chemical reaction would grow prohibitively high. Modeling the

solvent as a polarizable continuum, and not as individual molecules, makes ab initio computation feasible, surrounding the solute molecules outside of a molecular cavity.

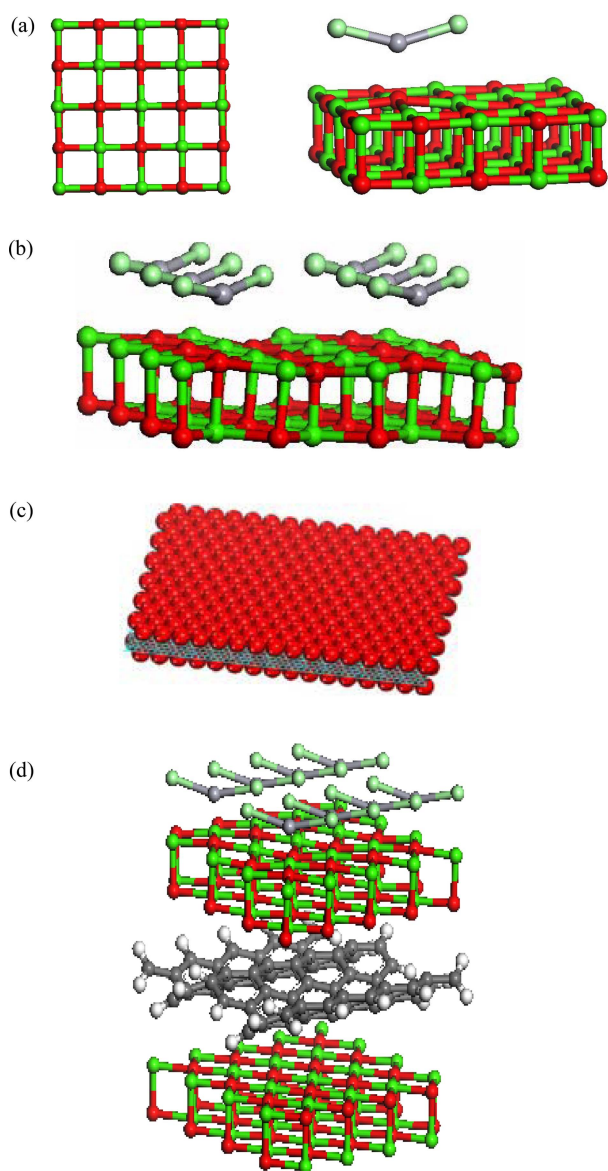
GaussSum and wxMacMolPlt parses the output files of GAMESS calculations to extract useful information. GaussSum<sup>28</sup> tracks the progress of the SCF convergence and geometry optimization, plots the density of states spectrum and the partial density of states, in the case of groups of atoms. The software is written by Noel O'Boyle and is available free under the GNU Public License. wxMacMolPlt<sup>29</sup> is a cross-platform gui for preparing, submitting and visualizing input and output for the GAMESS quantum chemistry package its features include a graphical molecule builder, GAMESS input generation, animation of output and visualization of molecules, normal modes, orbitals and other properties.

DFT performed with GGA and the local-density approximation (LDA) of the electronic exchange correlation energy has become the ab initio method of choice for modeling and characterization for mercury and carbon capture materials. Standard DFT-LDA and DFT-GGA methods are known to describe with precision and affordability a wide spectrum of interactions in bulk crystals and surfaces; however, computational studies on the accuracy of DFT in reproducing Hg and  $\text{CO}_2$ -sorbent forces are surprisingly scarce in the literature. In view of the ubiquity of DFT methods, it is therefore crucial to start filling this knowledge gap while putting a special emphasis on the underlying physics. Computational benchmark studies on Hg and  $\text{CO}_2$ -sorbent interactions, however, are technically intricate and conceptually difficult since most Hg and carbon capture materials have structural motifs that are large in size. In particular, genuinely accurate but computationally very intensive quantum chemistry approaches like MP2 and CCSD(T) can deal efficiently only with small systems composed of up to a few tens of atoms, whereas DFT can be used for much larger systems. DFT and MP2 methods qualitatively provide similar results, with hybrid DFT and MP2 in almost quantitative agreement.<sup>30</sup>

### 3. Results and Discussions

#### 3.1 Electronic properties of $\text{HgCl}_2$ adsorption on CaO with and without graphene support

$\text{HgCl}_2$  exists not as a salt composed of discrete ions, but rather is composed of linear triatomic molecules. In the crystal, each mercury atom is bonded to two close chloride ligands with Hg-Cl distance of 2.38 Å; six more chlorides are more distant at 3.38 Å, as shown in Fig. 1. The distances between mercury and oxygen on the  $5 \times 5 \times 2$  CaO surface is 2.344 Å for Hg(+2) at the present



**Fig. 1.** Optimized geometry of (a) the finite  $5 \times 5 \times 2$  CaO (001) at low concentration, (b) HgCl<sub>2</sub> adsorbed on the CaO (001) wherein each chlorine atom gravitated to calcium sites at high concentration (c) Sandwich-like model: graphene serves as a template for the creation of a metal oxide/graphene/metal oxide sandwich-like structure. (d) HgCl<sub>2</sub> captured on a CaO and graphene composite at high concentration. Green color depicts Calcium atoms; light green is Chlorine, red is Oxygen, and gray is Mercury. The CO<sub>2</sub> system can be modeled simply by replacing the HgCl<sub>2</sub> with CO<sub>2</sub> molecules.

calculation levels; the chlorine atom has a bond length of 3.258 Å to the surface with calcium. In both Fig. 1(b) and Fig. 1(d) the movement of the oxygen out of the flat surface makes it more favorable for strong interactions with mercury so that the Hg is much closer to the surface at 2.344 Å for HgCl<sub>2</sub> on CaO and 2.441 Å for HgCl<sub>2</sub> on g-CaO. Graphene an allotrope of carbon in the structure of a plane of sp<sup>2</sup>-bonded atoms with a bond length of

0.142 nanometers sandwiched in between the CaO.

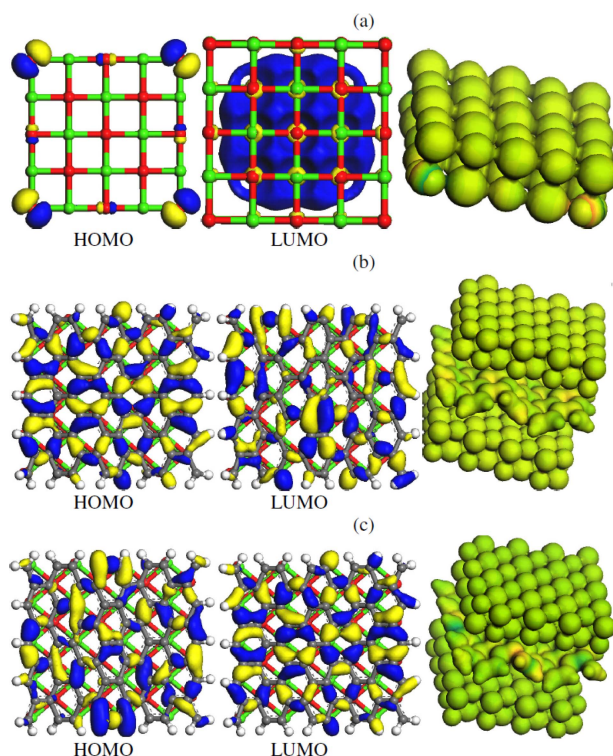
The adsorption energies of HgCl<sub>2</sub> on the modeled structure is  $-30.28$  kcal/mol at low concentration (*l*CaO);  $-24.24$  kcal/mol at high concentration (*h*CaO) and  $-22.80$  kcal/mol at high concentration with graphene support (*hg*-CaO) obtained with B3LYP methods through equation (2). The relaxation effect plays an important role in the HgCl<sub>2</sub> adsorption to move the adsorption energy into the chemisorption range. The oxidized mercury has a strong adsorption energy value on the  $5 \times 5 \times 2$  slab. The result suggests that CaO will be a good adsorbent for the reactive oxidized mercury species. The adsorption energies of HgCl<sub>2</sub> on the CaO surfaces fall within the chemisorption range. The chlorine atoms released from HCl or Cl<sub>2</sub> in the coal combustion flue gas system greatly enhanced the adsorption capability according to these predicted results.<sup>31)</sup>

Although absolute values of Mulliken and the Hirshfeld populations have little physical meaning, their relative values can be useful. For example, positive and negative values of bond population mean that the atoms are bonded or antibonded, respectively. A large positive value indicates that the bond is largely covalent, whereas there is no interaction between the two atoms if the bond population is close to zero. The atomic charges for a fixed clean periodic CaO slab and CaO slab reacted with HgCl<sub>2</sub> were used to obtain the final Mulliken and Hirshfeld charges. The Mulliken charge of the mercury atom in HgCl<sub>2</sub> on the surface is changed by  $-0.069$ ,  $-0.089$ ,  $-0.077$  and the chlorine is negatively charged to  $-0.145$ ,  $-0.075$ ,  $-0.070$  for *l*CaO, *h*CaO and *hg*-CaO configurations respectively. On the CaO surface, the final atomic charges of the nearby calcium atoms were almost the same after CaO-Hg adsorption while all charges on nearby oxygen atoms were changed positively after CaO-mercury species were formed. The Hirshfeld charges of the nearby calcium atoms were changed by  $-0.098$ ,  $-0.098$ ,  $-0.097$  and  $-0.097$  for the *hg*-CaO as an example. This is consistent with the Mulliken charge analysis as the same calcium atoms of the surface interact with mercury adsorbates.

The aim is to maximize the practical use of the combined advantages of both graphene and metal oxides as active materials for improving the HgCl<sub>2</sub> storage. The Calculation of the binding energies of g-CaO structures was elucidated with the following formula:

$$\Delta E_{g-CaO} = E_{g-CaO} - E_{CaO} - E_g \quad (4)$$

where  $\Delta E_{g-CaO}$  is the binding energy of CaO layer with the graphene layer together,  $E_{g-CaO}$  represents the energy of the CaO-graphene composite,  $E_{CaO}$  and  $E_g$  are the energy of the CaO and graphene layers in kcal/mol. In a



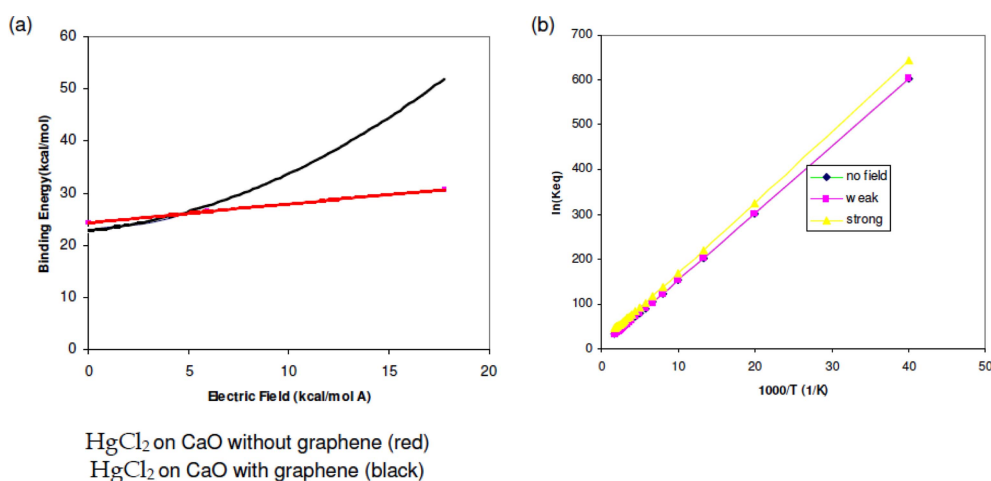
**Fig. 2.** HOMO-LUMO surface plots of (a) bare CaO with out an electric field (b) CaO+graphene composite without an electric field (c) CaO+graphene composite in the presence of an electric field. Green color depicts Calcium atoms and red is Oxygen.

composite, graphene provides chemical functionality and compatibility to allow easy processing of metal oxides in the composite. The metal oxide component mainly provides high capacity depending on its structure, size and crystallinity. The resultant composite has a  $-5.04$  kcal/mol binding energy that is not merely the sum of the individual components, but rather a new material with new functionalities and properties. From the view point of structure, on one hand, CaO anchored or dispersed on graphene can suppress the agglomeration of graphene, which can be accessed by considering now the graphene-graphene interlayer binding energy,  $\Delta E_{g/g}$  of the bilayer graphene system. The  $\Delta E_{g/g}$  was found by calculating the total energy of a single graphene layer, and subtracting that from the bilayer calculation, the result is that the  $\Delta E_{g-CaO}$  is significantly less than  $\Delta E_{g/g}$  which has a value of  $-0.42$  kcal/mol, supporting the claim that agglomeration with restacking is diminished in the presence of CaO. Also graphene increases the available surface area, leading to high chemical activity. On the other hand, the binding energy of CaO to itself in the presence of graphene is greater than the binding energy of CaO with itself without graphene by 18 kcal/mol which indicates that graphene can act as a support of metal oxides because it can induce the nucleation growth along with

the formation of metal oxide nano/microstructures with uniform dispersion and perhaps controlled morphology also on its surface with high chemical functionality. The final metal oxide-anchored graphene and the graphene-supported metal oxide can form a perfect integrated structure with a developed system. The synergistic effects often occurs on the graphene/metal oxide composites because of size effects and interfacial interactions. Graphene act as a 2D support for uniformly anchoring or dispersing metal oxides with well-defined sizes, shapes and crystallinity; (2) metal oxides suppressing the restacking of graphene; (3) graphene can induce the nucleation growth of metal oxide nano/microstructures and suppress its agglomeration.

Fig. 2, above, shows HOMO(the highest occupied molecular orbital) molecular orbital for the fully optimized  $5 \times 5$  CaO cluster model. For Fig. 2(a) the bare CaO with polarized HOMO at the boundary ends may lead to attraction with adjacent CaO within the vicinity that eventually may result to agglomeration; the edge oxygen atoms have a high energy state and  $\pi^*$  molecular orbitals which are formed from  $4s(\text{Ca})$  and  $2p(\text{O})$  atomic orbitals. In Fig. 2(b) the g-CaO sheet without an externally applied electric field is presented, with delocalized HOMO and a localized LUMO(the lowest unoccupied molecular orbital) at the center acts an electrophile to attract electrons from CaO layers in effect the CaO layer is charged less negatively. Polarized HOMO at the boundary ends of the CaO have vanished that can prevent CaO agglomeration. Fig. 2(c) depicts an externally applied field in the g-CaO sheet with polarized HOMO at the boundary ends and delocalized LUMO acts a nucleophile to donate electrons to CaO layers that in effect the CaO layer is more negatively charged. The charged surface is manifested in a darker green color compared to Fig. 2(b). that may lead to an enhanced interaction with  $\text{HgCl}_2$  and  $\text{CO}_2$  as discussed in the next. All adsorptions are electronically harmless processes and venial impacts on the energy gap were observed.

The effect of temperature on the equilibrium constants for the adsorption of mercury containing species on the CaO (001) surface was investigated with B3LYP calculations. Fig. 3(a) shows that the binding energy increases as the magnitude of the externally applied electric field increases although, it can be noticed that the CaO+graphene composite shown in black increases rapidly compared to the unsupported one; an advantage of graphene/metal oxide composites as advanced materials in  $\text{HgCl}_2$  capture as seen in the HOMO-LUMO surface plots. Due to the enhanced features of  $\text{HgCl}_2$  on a CaO (001) with graphene, the predicted equilibrium constant for adsorption of as a function of temperature at high concentration was calculated with and without the in-



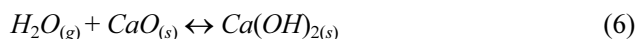
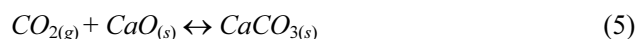
**Fig. 3.** The (a) binding energy of HgCl<sub>2</sub> on CaO with and without graphene reinforcement versus an externally applied electric field. The (b) predicted equilibrium constant for adsorption of HgCl<sub>2</sub> on a CaO (001) with graphene as a function of temperature at high concentration calculated with the B3LYP functional with and without the influence of an externally applied electric field. Weak field is 5.92 kcal/mol Å and strong field is 17.77 kcal/mol Å

fluence of an externally applied electric field. Fig. 3(b) shows effect of temperature on the equilibrium constants for the adsorption of HgCl<sub>2</sub> in the range of 250–600 K and was generated by using the atomistic thermodynamic calculations described earlier. Temperature has a strong effect on the equilibrium constant for HgCl<sub>2</sub> adsorption and the equilibrium constant has a steep change for HgCl<sub>2</sub> adsorption. The equilibrium constants decrease as the temperature increases with or without the application of an external electric field and the adsorption of HgCl<sub>2</sub> on CaO (001) surface is more favored at low temperatures. In the presence of a weak field of 5.92 kcal/mol Å it can be noticed that there is no significant change in the equilibrium constant however, at 17.77 kcal/mol Å a noticeable increase can be noted already. This is in good agreement with trends for exothermic systems.<sup>32–47</sup> However, the small equilibrium constants of HgCl<sub>2</sub> at high temperatures means that oxidized forms are not effectively adsorbed under these conditions.

One recovers the used adsorbents despite of the high adsorption energies at high heating rates, the desorption is kinetically favored. Although other effects connected with the physical form of the specie deposited must be considered. The desorption can be dependent on the dielectric and surface properties, electric field strength. The response of the specie to changes in surface properties upon rapid heating might result in favorable modifications of the surface charging effects, and other factors can also assist the desorption under rapid heating conditions. In general, the use of high electric-field gradients and heating can readily cause desorption from the surface.

### 3.2 Electronic properties of CO<sub>2</sub> adsorption on g-CaO support with and without the presence of water vapor

Carbon dioxide is chosen as the reference gas to calculate global warming potentials for other greenhouse gases because of its large contribution. Simple carbonation (Eq. 5) and hydration (Eq. 6) reactions were used namely:



The change of enthalpy for calcium carbonate CaCO<sub>3</sub> is –42.49 kcal/mol and it is a strongly exothermic reaction. The calculated enthalpy change was –55.87 kcal/mol, which yields a 24 % error compared to the experimental value. This confirmed one molecule of calcium carbonate could not lead to reliable adsorption results because it exists in a crystal structure. Therefore, the surface adsorption method was applied and a 5 × 5 × 2 CaO surface.

The distance between carbon of the carbon dioxide molecule and oxygen on the g-CaO surface is 1.37 Å after optimization. The bond lengths between carbon and the oxygen atoms of CO<sub>2</sub> molecule have changed from 1.51 Å to 1.25 Å when the double bonds are broken. In the calcite (CaCO<sub>3</sub>) structure, the default bond lengths between carbon and oxygen are all equally 1.28 Å. As the g-CaO surface is relaxed, the optimized structure shows similarity to the calcite geometry. When the CO<sub>2</sub>

adsorbs on the g-CaO surface in the presence of water vapor in the PCM model, it has the same geometry as the CO<sub>2</sub>-only adsorption model. This indicates that water vapor in the PCM formalism does not affect the optimized CO<sub>2</sub> adsorption geometry in forming the calcite structure. The adsorption energies were calculated using optimized structures obtained from the previous calculations in section 3.1 by replacing the HgCl<sub>2</sub> with CO<sub>2</sub> molecules where they are defined by

$$\Delta E_{ads} = E_{CO_2+g-CaO} - E_{CO_2} - E_{g-CaO} \quad (8)$$

$$\Delta E_{ads}^{PCM} = E_{CO_2+g-CaO}^{PCM} - E_{CO_2}^{PCM} - E_{g-CaO}^{PCM} \quad (9)$$

The adsorption energies of CO<sub>2</sub> with and without water on the 5 × 5 × 2 structure were -23.82 and -40.31 kcal/mol, respectively. The adsorption energy for strong CO<sub>2</sub> is in the chemical adsorption range in agreement with literature.<sup>38-39)</sup> The water vapor enhances the adsorption ability of CO<sub>2</sub> on the relaxed g-CaO surface as it increases the adsorption energy by ~ -16 kcal/mol. The water vapor leads to more favorable CO<sub>2</sub> adsorption on the relaxed CaO/graphene here. The strong hydration effect is in agreement with experimental results. Lime-based sorbents are used for fuel- and flue-gas capture, thereby representing an economic and effective way to reduce CO<sub>2</sub> emissions. Their use involves cyclic carbonation/calcination, which results in a significant conversion reduction with increasing number of cycles. To reactivate spent CaO, vapor phase hydration is typically performed. Steam was attributed to improving carbonation performance on graphene/CaO surfaces. The adsorptions are electronically harmless as with the HgCl<sub>2</sub> adsorption on CaO with and without the graphene support.

The ΔG of carbonation reaction on the model CaO surface is -32.47 kcal/mol at 298.15 K, with only a 5 % error compared to experiment. The ΔG is also the chemical potential that is minimized when a system reaches equilibrium at constant pressure and temperature. Its derivative with respect to the reaction coordinate of the system vanishes at the equilibrium point. As such, it is a convenient criterion of spontaneity for processes with constant pressure and temperature. The results of CO<sub>2</sub> adsorption were compared with the Gibbs energy for CO<sub>2</sub> adsorption with water vapor, the hydration effect contributed to much more favorable adsorption at -39.17 kcal/mol. The high adsorbability of CO<sub>2</sub> with a water molecule using the PCM model on the relaxed CaO surface is valuable for real flue gas systems. Most CO<sub>2</sub> exists in the presence of water vapor. However, CO<sub>2</sub> adsorption without a water molecule on the relaxed CaO surface is more favorable at high temperatures. A water molecule in the liquid form, which is the condition at

lower temperatures of the flue gases, would lead to more favorable CO<sub>2</sub> adsorption on calcium oxide surface. Regardless, the ability for adsorbing CO<sub>2</sub> in the presence, or absence, of a water molecule on the CaO/graphene slab is chemisorptions at all temperature ranges. Desorption is suggested to be achieved also using high electric-field gradients.

#### 4. Conclusions

The effect of temperature on the equilibrium constant for the adsorption of HgCl<sub>2</sub> species on 5 × 5 × 2 g-CaO surface was investigated by using the GGA/B3LYP calculations. Results show that graphene is a support for uniform dispersion of CaO; the metal oxide suppresses the re-stacking of graphene and graphene suppresses the volume change and agglomeration of CaO as supported by HOMO-LUMO surface plots. The compound, HgCl<sub>2</sub> was found to be stable on the surface; it does not desorb or return to the reactant species. The equilibrium constants for mercury adsorption on CaO surfaces with and without the presence of an externally applied electric field are affected by temperature. From the trends, HgCl<sub>2</sub> adsorption on g-CaO surfaces is favorable at low flue gas temperatures. Carbon dioxide adsorption on the g-CaO slab surface led to formation of a calcium carbonate structure through the carbonation reaction. There was enhanced CO<sub>2</sub> adsorption ability in the presence of water vapor. Results show that water molecules mediated through PCM model leads to enhanced CO<sub>2</sub> adsorption on the relaxed g-CaO surface through the hydration effect. Understanding the mechanism associated with Hg and CO<sub>2</sub> adsorption on g-CaO will help to optimize the conditions to limit emissions from coal-fired power plant processes.

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