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# Plasma Textured Glass Surface Morphologies for Amorphous Silicon Thin Film Solar Cells-A review

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The surface morphology of the front transparent conductive oxide (TCO) films plays a vital role in amorphous silicon thin film solar cells (a-Si TFSCs) due to their high transparency, conductivity and excellent light scattering properties. Recently, plasma textured glass surface morphologies received much attention for light trapping in a-Si TFSCs. We report various plasma textured glass surface morphologies for the high efficiency of a-Si TFSCs. Plasma textured glass surface morphologies for the high efficiency of a-Si TFSCs. Plasma textured glass surface morphologies for the high efficiency of a-Si TFSCs. Plasma textured glass surface features and are proposed for future high efficiency of a-Si TFSCs.

Keywords: Light trapping, ICP-RIE, Multi-textured glass, Rms roughness, Haze ratio, a-Si thin film solar cell

## **1. INTRODUCTION**

Amorphous silicon thin film solar cells (a-Si TFSCs) are considered promising candidates for future low-cost and large-area photovoltaic devices. Light trapping is considered an essential mechanism that allows the reduction of reflection losses with an increase in the optical path length of incident light in photovoltaic devices. The front transparent conductive oxide (TCO) films must exhibit good transparency, low resistivity and excellent light scattering properties for high efficiency a-Si TFSCs. Various textured TCO films like SnO<sub>2</sub>:F, ZnO:Al, and ZnO:B are commonly used as substrates to increase the light path within the absorber

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This is an open-access article distributed under the terms of the Creative Commons Attribution Non-Commercial License (http://creativecommons.org/licenses/by-nc/3.0) which permits unrestricted noncommercial use, distribution, and reproduction in any medium, provided the original work is properly cited. layer for the enhancement of light trapping in the solar cell [1-15]. However, these textured TCOs substrates are insufficient to gain the light scattering effect needed for longer wavelength region due to their smaller (~600 nm) textured size features. Recently, few research groups proposed larger feature size TCO substrates. Arbitrary patterned glass substrates and self-assembled metal particle applying to etching mask [54].

An alternative approach to improve the light-trapping in a-Si TFSCs is to use of TCO films on the textured glass substrates [1-10]. Various methods to create textured glass surface morphologies have been investigated in recent years like sand blasting, powder blasting, wet chemical etching with hydrofluoric acid (HF) based solutions, aluminum induced texturization (AIT) and plasma etching through a nanostructured metal mask [11-16]. Thus, the textured glass surface substrates with nano- and micro size features can be employed to scatter light in lower as well as in longer wavelength region. Recently, textured glass surface morphologies with high rms roughness and haze ratio were proposed for high efficiency a-Si TFSCs due to their superior electrical and optical characteristics as compared to textured TCO films [15,16-25].

The inductive coupled plasma-reactive ion etching (ICP-RIE) process has received much attention, as the glass surface can be textured with high rms roughness and haze ratio. Hongsingthong et al. [6-8] reported the influence of RIE textured glass substrates with high rms roughness and haze ratio of ZnO films for the improved performance in a-Si TFSCs, due to their nano- and microsize textured surface morphologies. Isabella et al. [4] reported modulated surface morphologies with various geometrical features for an enhanced scattering mechanism.

These nano- and micro-size textured surface morphologies can scatter light in the visible as well as NIR wavelength regions [4,7-10]. Recently, Hussain et al. [5,39,54] reported the influence of periodic  $SF_6$ /Ar plasma textured glass surface morphologies for the light trapping in a-Si TFSCs.

We review the relevant literature reports on plasma textured glass surface morphologies for a-Si TFSCs and various popular light scattering surface morphologies for the TCO films are discussed. A short historical survey of textured glass surface morphologies for amorphous silicon solar cells is included in this review. Several RIE textured glass surface morphologies with various experimental parameters are also discussed.

## 2. RESULTS AND DISCUSSION

#### 2.1 Textured TCO surface morphologies

Rough TCO/silicon interfaces are generally used in a-Si TFSCs for an efficient light trapping. Effective light scattering not only enhances the current density of a-Si TFSCs but also saves cost by making thinner solar cells [26-40]. To obtain textured TCO surface morphologies, various strategies have been applied in the solar cell industry. Figure 1 shows various light trapping surface morphologies for the TCO films. Figure 1(a) and 1(b) represents textured TCO surface morphologies of SnO<sub>2</sub>:F and ZnO:B films being employed by the Ashai-U research group [46,56-61]. The ZnO:B films were grown by low pressure CVD (LP-CVD) from precursors of diethyl-zinc and H<sub>2</sub>O; the conductivity was easily controlled by using di-borane as the dopant source [58,59]. The SnO<sub>2</sub>:F were deposited by (APCVD) from the precursors of SnCl4 and H<sub>2</sub>O [60-62]. Figure 1(c) and 1(d) show pyramid and cauliflower ZnO surface morphologies used by Tokyo technology. The pyramid type ZnO films were deposited on the Corning glass substrate by MOCVD technique by using D<sub>2</sub>O and H<sub>2</sub>O mixtures as oxidants for di-ethylzinc (DEZ). A mixture of D<sub>2</sub>O/ H<sub>2</sub>O was employed to control the surface morphology of the ZnO films [1,6,7]. The cauliflower ZnO films were deposited on RIE textured glass substrate using a MOCVD process with a mixture of D<sub>2</sub>O/H<sub>2</sub>O for the control of surface morphology [6]. B2H6 (1% H<sub>2</sub> diluted) was used as the dopant gas. Photovoltaics laboratory (PV-Lab) is well known for the various light trapping techniques employed in a-Si TFSCs. The ZnO:B films used by PV-Lab are deposited by low pressure chemical vapor deposition (LP-CVD) process for the light scattering in a-Si TFSCs [59,63].

#### 2.2 Plasma textured glass surface morphologies

Plasma etching (physical sputtering) was introduced to integrated circuit (IC) manufacturing in the 1970s to reduce liquid waste disposal in manufacturing and achieve high selectivities as compared to wet chemistry. The ability to anisotropically etch aluminum, silicon, and silicon dioxide in plasmas became the innovation that allowed the features in IC technology to continuously shrink over the next 40 years. Today, plasma etching plays a major role in allowing complex circuit patterns







ZnO:B (Ashai-U)

Cauli-flower ZnO (Tokyo Tech)

Fig. 1. Various light trapping surface morphologies for the transparent conductive oxides (TCOs) films. Fig. 1 (a), (b) are reproduced with permission from [46], F. -J. Haug et al., Light management in thin fllm silicon solar cells, Energy Environ. Sci.,8 (2015) 824. Permission granted. License number 3679260160905 Copyright© Royal Society of Chemistry. Fig. 1 (c) is reproduced with permission from [1], M. Konagai, Present Status and Future Prospects of Silicon Thin-Film Solar Cells, Jpn. J. Appl. Phys. 50 (2011) 030001. Permission granted Copyright © The Japan Society of Applied Physics. Fig. 1 (d) is reproduced with permission from [6], A. Hongsingthong et al., ZnO Films with Very High Haze Value for Use as Front Transparent Conductive Oxide Films in Thin-Film Silicon Solar Cells, Appl. Phys. Exp.,3 (2010) 051102. Permission granted.Copyright© The Japan society of Applied Physics.

printed in a photolithographically defined polymers to be transferred to the glass, silicon, metals and silicon dioxide that make up the integrated circuits at the heart of these devices [10,41-44,64].

Silicate glasses are widely used in everyday life due to their unique properties. Silicate glasses are generally resistant to most gases and liquids and at room temperature they are only readily dissolved by hydrofluoric acid or other HF containing aqueous solutions. These glasses are used for variety of applications with controlled dissolution in HF-based etchants [44-47,65].

Commonly used plasma parameters for the ICP-RIE system are flow rate of etching gas, etching time, RF power, biased power and the pressure. Here, we include few experimental results of RIE textured glass surface morphologies [21-24]. Various etching chemistries are generally used for glass etching. Commonly used etching chemistries contain CF<sub>4</sub> [6,7,11,38] and SF<sub>6</sub> [5,39,55] gases with the addition of either O<sub>2</sub> or Ar for the glass etching. Few research groups also used CHF<sub>3</sub> etching gas to texture glass. Konagai et al. mostly employed CF<sub>4</sub> gas for the random texturing of glass surface morphologies. Various metal masks like Al, SiN3, and PR are used for etching purposes according to the applications [48-50].

Table 1 illustrates the short historical survey about the textured glass surface morphologies used for solar cells. Most of research groups used Corning 7059 or Corning XG glass as a substrate due to their chemical composition.  $CF_4$  and  $SF_6/Ar$ are the commonly used etching gases for the texturing process. High rms roughness, transmittance and haze ratio are preferred characteristics for the a-Si:H or µc-Si:H solar cells. Most of the researchers used randomly textured RIE glass surface morphologies for the TCO deposition. The hot debate between random or periodic textured surface morphologies being superior for light trapping and high value of current density is under investigation

Table 1. Various plasma textured glass surface morphologies for solar cell applications.

Glass type	Etching Process	Etching Gas	RMS roughness	Transmittance	Haze	Application	Ref.
Corning 7059	RIE	$CF_4$	310 nm	>80%	~90%	p-i-n (µc-Si:H)	6
Corning 7059	RIE	$CF_4$	310 nm	>80%	~90%	p-i-n (a-Si:H)	38
Soda-lime	RIE	$CF_4$	nil	~90%	~85%	a-Si:H/µc-Si:H	11
Soda-lime	RIE	$CF_4$	360 nm	>80%	92%	p-i-n (µc-Si:H)	7
Corning 7059	RIE	$CF_4/O_2$	518 nm	~85%	95%	p-i-n (µc-Si:H)	37
Corning 2000	ICP-RIE	SF <sub>6</sub> /Ar	325 nm	~90%	65%	p-i-n (a-Si:H)	5
Corning 2000	ICP-RIE	SF <sub>6</sub> /Ar	461 nm	~88%	78%	p-i-n (a-Si:H)	39
Corning XG	RIE	$CF_4/Ar$	nil	nil	nil	p-i-n (a-Si:H)	40
Corning XG	IBE	Ar	~270 nm	nil	nil	p-i-n (µc-Si:H)	41
Corning XG	AIT	HNO <sub>3</sub> , HF	755 nm	~90%	95%	n-i-p (µc-Si:H)	42
White	RIE	$CF_4/O_2$	353 nm	>85%	80%	p-i-n (a-Si:H)	54



Fig. 2. SEM images of textured glass substrates for various etching times. The rms roughness of each textured glass is also shown. Reproduced with permission from [6], A. Hongsingthong et al., ZnO Films with Very High Haze Value for Use as Front Transparent Conductive Oxide Films in Thin-Film Silicon Solar Cells, Appl. Phys. Exp., 3 (2010) 051102. Permission granted. Copyright © The Japan society of Applied Physics.

[5-10,39,55]. Few research reports of periodic textured ICP-RIE glass surface morphologies for solar cell applications are recently available [5,39,55].

Figure 2 represent textured glass surface morphologies with various glass-substrate treatment times. The rms roughness of each surface morphology was obtained from 3-D AFM images (scan area:  $10 \ \mu m \times 10 \ \mu m$ ). As the glass substrate treatment times increased from 0 min to 40 min, the surface of the textured glass surface morphology changed from flat to largely textured. The rms roughness of the glass substrates increased from 1.27 nm to 175 nm as the treatment time increased from 0 min to 10 min. With the further increase in treatment time from 10 min to 40 min, the rms roughness of glass surface morphology increased from 175 nm to 310 nm, respectively [6,24,25].

The SEM images and rms roughness of the surface of sodalime glasses with various etching conditions are shown in Fig. 3. As the power density was increased from  $0 \text{ W/cm}^2$  to 1.5 W/cm<sup>2</sup>, the glass surface changed from a smooth flat surface to one with small craters and lateral feature size of 200 nm to 700 nm. The textured glass surface became smoother when the power density was  $1.9 \text{ W/cm}^2$  with rms value of 3 nm. This may be due to the difference in the energetic ion bombardments [7,51] that affect the etching kineticd and allow smooth surfaces. The rms roughness of textured glass slightly increased as the power density increased to  $1.5 \text{ W/cm}^2$ , and showed a quick drop for power



Fig. 3. SEM micrographs of soda-lime glasses with various etching conditions: a) 0 W/cm<sup>2</sup>; b) 1.5 W/cm<sup>2</sup>; c) 10 Pa; and d) 13 Pa. Reproduced with permission from [7], A. Hongsingthong et al., Development of textured ZnO-coated low-cost glass substrate with very high haze ratio for silicon-based thin film solar cells, Thin Solid Films, 537, (2013) 291. Permission granted. Copyright © 2013 Elsevier.

densities greater than 1.5 W/cm<sup>2</sup>. Figure 3(c) and 3(d) represent textured soda-lime glass surface morphology with varying pressures. The textured soda-lime glass surface morphology changed from a flat surface to one with large craters and lateral feature sizes between 4,000 nm and 6,000 nm, as well as having rms roughness increase from 0.6 nm to 360 nm with the increase of gas pressure from 0 Pa to 13 Pa. The different surface textures at various gas pressures may be caused by the variation of gas radical concentrations like ion-radical and neutral etchant species in the dry plasma [7,50,51].

Figure 4 shows the textured glass surface morphologies for various RIE etching times. The surface of white glass was etched gradually with the etching time up to 60 min. However, large crater sizes suddenly appeared on the glass surface. Various craters sizes ranging from 1  $\mu$ m - 10  $\mu$ m) were identified on the textured glass surface. The glass etching rate with CHF<sub>3</sub>-CF<sub>4</sub> mixture gas can be decreased with increasing nonvolatile elements contents in glass surface morphology as reported by Leech [54,66]. Thus, the area with low and high etching rates appeared due to the segregation of silica and nonvolatile elements that exist on the glass surface [11,52,53]. As the etching time increased from 60 min to 120 min, crater feature size became constant at an approximate size of 4  $\mu$ m. With the increase of etching time of about 150 min, the glass surface was dominated by smaller craters. The silica



Fig. 4. SEM micrographs of RIE-etched white glass substrate under various glass etching times: (a) no etching; (b) 60 min; (c) 120 min; and (d) 150 min. Reproduced with permission from [54], H. Wada et al., Improved light trapping effect for thin-fllm silicon solar cells fabricated on double-textured white glass substrate, Can. J. Phys., 92, (2014) 920. Permission granted. License number 3681950779704. Copyright © 2014 NRC Research Press.



Fig. 5. SEM images (top and cross sectional view) of the ICP-RIE textured corning glass with various pattern sizes. Reproduced with permission from [5], S. Q. Hussain et al., Light trapping scheme of ICP-RIE glass texturing by  $SF_6/Ar$  plasma for high haze ratio, Vacuum, 94, (2013) 87. Permission granted. License number 3679461402631. Copyright © 2013 Elsevier.

and nonvolatile element compositions at the inner part of glass stabilized. The chemical etching mechanism of  $CF_4/O_2$  for RIE process is given as,

$$CF_4 \rightarrow CF_2 + 2F$$
 (1)

$$SiO_2 + O_2 \rightarrow SiF_4 + 2O$$
 (2)

$$SiO_2 + 2CF_2 \rightarrow SiF_4 + 2CO$$
 (3)

Where the SiF<sub>4</sub>, O, and CO are volatile compounds and were evacuated from the chamber immediately [54].

Figure 5 shows the SF<sub>6</sub>/Ar plasma textured glass surface mor-

phologies with various UV photo-lithography pattern dimensions. The glass with pattern dimensions of  $(6\times 6)$  and  $(12\times 5)$  $\mu$ m<sup>2</sup> were etched to (3×9) and (8×9)  $\mu$ m<sup>2</sup> after 60 min of ICP-RIE process as shown by Fig. 5(a) and 5(c). The Al under-cut was primarily related to chemical etching and micro-trenching that increases with the passage of time. The Ar was mixed with SF<sub>6</sub> gas to remove under-cut and micro trench formation by the fast removal of by-products. Figure 5(b) and 5(d) shows the crosssectional views of the SF<sub>6</sub>/Ar plasma textured glass surface morphologies. All the textured glass surface morphologies showed almost uniform etching rate due to the fixed ICP-RIE process conditions. The minor variation in the etching rate from 22.66 nm/min to 27.33 nm/min was related to the different surface structure of glass. The maximum etching depth of ~ 1.73 µm was recorded by the textured glass with the pattern dimension of  $(12\times5)$  µm<sup>2</sup>. The higher concentration of metal elements like F, B, Al and Na in the glass surface can reduce the etching rate of textured glass by ICP-RIE process. These metal elements produced the non-volatile etch products after reacting with halogen gases. The Ar gas can be used to remove the non-volatile etch products by physical etching [5,39,55,66,67]. The etching mechanism of SF<sub>6</sub>/Ar is readily available in following research reports related to textured glass [5,39,55,68].

### 3. CONCLUSIONS

Front TCO films have the important role of enhancing the current density and hence improve the performance of a-Si TFSCs. An alternate approach for improved light trapping surface morphology is to texture the glass substrate instead of the TCO films. We present various plasma textured glass surface morphologies for light trapping in a-Si TFSCs. RIE is an important process to texture the glass surface with precise accuracy. RIE textured glass surface morphologies contain the micro- and nano-size features that can enhance the light scattering in visible as well as in NIR region. ICP-RIE textured glass surface morphology with high rms roughness, haze ratio are also proposed for the future high efficiency a-Si TFSCs.

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103

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