TRANSACTIONS ON ELECTRICAL AND ELECTRONIC MATERIALS Vol. 16, No. 2, pp. 70-73, April 25, 2015



pISSN: 1229-7607 eISSN: 2092-7592 DOI: http://dx.doi.org/10.4313/TEEM.2015.16.2.70 OAK Central: http://central.oak.go.kr

Study on Graphene Thin Films Grown on Single Crystal Sapphire Substrates Without a Catalytic Metal Using Pulsed Laser Deposition

Byoung Jin Na, Tae Hwa Kim, Cheon Lee[†], and Seok-Hyun Lee Department of Electrical Engineering, College of Information Technology and Engineering, Inha University, Incheon 402-751, Korea

Received February 26, 2015; Revised March 2, 2015; Accepted March 9, 2015

Many studies have used chemical vapor deposition (CVD) to grow graphene. However, CVD is inefficient in terms of production costs, and inefficient for mass production because a transfer process using a catalytic metal is needed. In this study, graphene thin films were grown on single crystal sapphire substrates without a catalytic metal, using pulsed laser deposition (PLD) to resolve these problems. In addition, the growth of graphene using PLD was confirmed to have a close relationship with the substrate temperature.

Keywords: Graphene, PLD, Thin film, Without catalytic metal, HOPG

1. INTRODUCTION

Since the discovery of new carbon allotropes in the second half of the 20th century, there has been considerable research into their production and applications. Zero-dimensional fullerene, which was discovered in 1985, carbon nanotubes, which were discovered in 1991, and two-dimensional graphene, which was the latest discovery, have been included in that research [1]. Graphene has a hexagonal lattice composed of carbon atoms with a one-atom thickness.

The Manchester group team and Geim first produced graphene by mechanical exfoliation using scotch tape in 2004 [2,3]. Graphene is very stable, both structurally and chemically, has a high electron mobility of 20,000 cm²/Vs and high thermal conductivity of 5,000 W/mK [4-6].

Graphene thin films can be obtained by using any one of four methods: mechanical exfoliation, chemical exfoliation, epitaxial growth, and chemical vapor deposition (CVD) [2,7-9]. The most

[†] Author to whom all correspondence should be addressed: E-mail: chnlee@inha.ac.kr

Copyright ©2015 KIEEME. All rights reserved.

This is an open-access article distributed under the terms of the Creative Commons Attribution Non-Commercial License (http://creativecommons.org/licenses/by-nc/3.0) which permits unrestricted noncommercial use, distribution, and reproduction in any medium, provided the original work is properly cited.

common method is CVD, which involves the use of a catalytic metal such as Ni, Pd, Co, Cu or Au, and high temperatures of 1,000 °C. In addition, a transfer process using a catalytic metal is required. The process is inefficient in terms of production costs, with many defects arising during the transfer process and further difficulties in mass production [10-12].

Pulsed laser deposition (PLD) is expected be an appropriate technique for thin film growth without any changes in the composition of the target, because ablation occurs momentarily when the target is irradiated with a laser beam. In addition, it is easy to produce crystalline carbon thin film materials, such as carbon nanostructures. One advantage of PLD is that transcription processes, which is a problem with CVD, can be eliminated, and a graphene thin film can be prepared without a catalytic metal. Graphene growth by PLD was confirmed to have a close relationship with temperature [13-15].

In this study, graphene thin films were grown on single crystal sapphire substrates without a catalytic metal, using pulsed laser deposition. The crystalline characteristics of grown graphene thin films were investigated.

2. EXPERIMENT

A Quantel Brillant B Q-switched 4th harmonic Nd:YAG laser (λ



Fig. 1. Schematic diagram of the PLD system.

Table 1. Pulsed laser deposition conditions

Laser Power Density	1.2 J/cm ²
Substrate Temperature	400, 500, and 600 °C
Deposition Time	90 seconds
Target Material	Highly Ordered Pyrolytic Graphite (HOPG)

= 266 nm) with a fluence maximum of 2 J/cm² and a repetition rate of 10 Hz was used to deposit the graphene thin films by PLD. The laser power density was 1.2 J/cm². The chamber for growth was initially evacuated to 10^{-5} Torr with an oil diffusion pump and a rotary pump.

The targets used to deposit thin films were highly ordered pyrolytic graphite (HOPG). The optical absorption rate of HOPG has a characteristic high optical absorption rate of the 4th harmonic (266 nm). The target was rotated at a constant speed of approximately 20 rpm so that the laser beam would irradiate different areas.

The substrates were located 40 mm away from the target at various temperatures between 400 and 600 °C. A 1 cm × 1 cm × 0.4 mm sapphire substrate was used. The surfaces of the sapphire substrates were cleaned ultrasonically in acetone, ethanol and deionized water, for 10 minutes at each step, and then dried using an N₂ gun. The cleaned sapphire substrates were then moved into a vacuum.

Figure 1 shows a schematic diagram of the PLD system used in this study. Table 1 lists the pulsed laser deposition conditions.

Raman spectroscopy with a green laser beam (514.2 nm) was used to examine the structure of graphene. The spectra were obtained in the range 1,000~3,000 cm⁻¹. The surface topography of the graphene thin films was observed by atomic force microscopy (AFM). The chemical-bond state of the graphene thin films was analyzed by X-ray photoelectron spectroscopy (XPS).

3. RESULTS AND DISCUSSION

Figure 2 shows the Raman spectrum of the graphene thin film deposited with a 1.2 J/cm² laser at a substrate temperature. The G band at 1,580 cm⁻¹ was assigned to a non-resonant scattering process. The characteristics of the in-plane sp² carbon hexagonal networks due to the optical phonon (E_{2g}) near the Brillouin zone center (Γ point) were examined [16]. The presence of a G band is characteristic of the formation of a hexagonal carbon lattice.

The D band at 1,350 cm⁻¹ and 2D band at 2,700 cm⁻¹ were assigned to double resonant scattering processes. This involves two real spaces between two continuous electronic excitations with different energies and momenta, and is caused by phonons



Fig. 2. Raman spectra of graphene thin films deposited with a 1.2 J/ cm^2 laser at: (A) 400 °C, (B) 500 °C, and (C) 600 °C.

near the K point of the Brillouin zone. The D band indicates the disorder of the sp² combined state. The 2D band will be present even in the absence of the D band. On the other hand, the 2D band to G band intensity ratio indicates the number of graphene layers [17].

Figures 2(A) and (B) show the 400 °C and 500 °C Raman spectra of the D peak and G peak spectra overlapping. This is a not crystallized form of graphene, which means that nano-graphite should be considered. Figure 2(C) shows that the 600 °C Raman spectra of the D peak and G peak were separated. The successful formation of graphene was considered. The I_D/I_G ratio of graphene formed at 600 °C was 0.55, which indicates the presence



Fig. 3(A). XPS spectrum of the graphene thin film deposited with a laser density of 1.2 J/cm² at a substrate temperature of 600 \degree C.



Fig. 3(B). XPS spectrum around the C1s.



Fig. 3(C). XPS spectrum around the O1s.



Fig. 4. AFM image of the graphene thin film deposited with a laser density of 1.2 J/cm² at a substrate temperature of 600 $^{\circ}$ C.

sample was fabricated.

Figure 3(A) shows an XPS spectrum of the graphene thin film deposited with a laser density of 1.2 J/cm² at a substrate temperature of 600 °C. The XPS spectra survey indicated the range of binding energies from 0 to 1,200 eV, showing that the main chemical components are O1s, N1s, Si2p, and C1s. The component of the graphene thin film was C and O. Figure 3(B) shows an XPS spectrum of C1s. The C1s peak was observed as the main peak at 284 eV, which combines the binding states of the C=C, C-OH, C=O, and O=C-OH bonds. Figure 3(C) shows an XPS spectrum of O1s. The O1s peak shows the bonding state between oxygen and carbon. The main O1s peak was observed at 530 eV, which indicates the presence of C-O and C=O bonds. Based on the peak in the XPS spectra, the bond between the oxygen and carbon was considered.

Figure 4 shows an AFM image of the graphene thin film deposited with a laser density of 1.2 J/cm² at a substrate temperature of 600 °C. The number of graphene layers was interpreted as FLG by Raman analysis, but the AFM image indicated three graphene layers.

4. CONCLUSIONS

To overcome the CVD growth method's disadvantage of having to use a catalytic metal, graphene thin films were deposited directly without a catalytic metal by PLD, using a Nd:YAG laser. The formation of FLG at a substrate temperature of 600 °C was confirmed by Raman spectroscopy. On the other hand, nanographite was formed at substrate temperatures lower than 600 °C. Nano-graphite formed in the graphene thin film with increasing substrate temperature. The formation of a graphene thin film was attributed to carbon particles being emitted from HOPG and rearranged by high substrate temperatures.

ACKNOWLEGMENT

of defects in graphene. The I_{2D}/I_G ratio of the graphene formed at 600 °C was 0.3, which indicates three layers of graphene. Based on the peak in the Raman spectra, a few-layer graphene (FLG)

This research was supported by Inha University Research Grant of 2014 (50249-01).

Trans. Electr. Electron. Mater. 16(2) 70 (2015): B. J. Na et al.

REFERENCES

- A. K. Geim and P. Kim, *Scientific American*, 298, 90 (2008). [DOI: http://dx.doi.org/10.1038/scientificamerican0408-90].
- [2] K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, Y. Zhang, S. V. Dubonos, I. V. Grigorieva, and A. A. Firsov, *Science*, **306**, 666 (2004). [DOI: http://dx.doi.org/10.1126/science.1102896].
- [3] K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, M. I. Katsnelson, I, V. Grigorieva, S. V. Dubonos, and A. A. Firsov, *Nature*, 438, 197 (2005). [DOI: http://dx.doi.org/10.1038/nature04233].
- [4] C. Faugeras, B. Faugeras, M. Orlita, M. Potemski, R. R. Nair, and A. K. Geim, ACS Nano, 4, 1889 (2010). [DOI: http://dx.doi. org/10.1021/nn9016229].
- [5] W. Cai, A. L. Moore, Y. Zhu, X. Li, S. Chen, L. Shi, and R. S. Ruoff, *Nano Lett.*, **10**, 1645 (2010). [DOI: http://dx.doi.org/10.1021/ nl9041966].
- [6] C. Lee, X. Wei, J. X. Kysar, and J. Hone, *Science*, **321**, 385 (2008).
 [DOI: http://dx.doi.org/10.1126/science.1157996].
- [7] S. Stankovich, D. A. Dikin, G.H.B. Dommett, K. M. Kohlhaas, E. J. Zimney, E. A. Stach, R. D. Piner, S. T. Nguyen, and R. S. Ruoff, *Nature*, 442, 282 (2006). [DOI: http://dx.doi.org/10.1038/na-ture04969].
- [8] P. Sutter, *Nature Mater.*, 8, 171 (2009). [DOI: http://dx.doi. org/10.1038/nmat2392].

- [9] K. S. Kim, Y. Zhao, H. Jang, S. Y. Lee, J. M. Kim, K. S. Kim, J. H. Ahn, P. Kim, J. Y. Choi, and B. H. Hong, *Nature*, 457, 706 (2009). [DOI: http://dx.doi.org/10.1038/nature07719].
- [10] T. Oznuluer, E. Pince, E. O. Polat, O. Balci, O. Salihoglu, and C. Kocabas, *Appl. Phys. Lett.*, **98**, 183101 (2011). [DOI: http:// dx.doi.org/10.1063/1.3584006].
- [11] E. S. Kim, H. J. Shin, S. M. Yoon, S. J. Chae, J. J. Bae, F. Gunes, J. Y. Choi, and H. Y. Lee, *Appl. Phys. Lett.*, **99**, 223102 (2011). [DOI: http://dx.doi.org/10.1063/1.3665616].
- [12] D. Wan, T. Lin, H. Bi, F. Huang, X. Xie, I. W. Chen, and M. Jiang, *Adv. Funct. Mater.*, **22**, 1033 (2012). [DOI: http://dx.doi. org/10.1002/adfm.201102560].
- S. R. S. Kumar and H. N. Alshareef, *Appl. Phys. Lett.*, **102**, 012110 (2013). [DOI: http://dx.doi.org/10.1063/1.4773987].
- [14] F. Y. Chunang, C. Y. Sun, T. T. Chen, and I. N. Lin, *Appl. Phys. Lett.*, **69**, 3504 (1996). [DOI: http://dx.doi. org/10.1063/1.117226].
- [15] K. Wang, G. Tai, K. H. Wong, S. P. Lau, and W. Guo, *AIP ADV*, 1, 022141 (2011). [DOI: http://dx.doi.org/10.1063/1.3602855].
- [16] C. Faugeras, A. Nerriere, M. Potemski, A. Mahmood, E. Dujardin, C. Berger, and W. A. de Heer, *Appl. Phys. Lett.*, **92**, 011914 (2008). [DOI: http://dx.doi.org/10.1063/1.2828975].
- [17] A. Obraztsov, E. Obraztsova, A. Zolotukhin, and A. Tyurnina, *J. Exp. Theor. Phys.*, **106**, 569 (2008). [DOI: http://dx.doi. org/10.1134/S1063776108030163].