

Superconducting transitions of $\text{LuBa}_2\text{Cu}_3\text{O}_{7-z}$ heated under various atmospheres

M. S. Lee and H.K. Lee*

Kangwon National University, Chuncheon, Korea

(Received 16 December 2013; revised or reviewed 11 February 2014; accepted 12 February 2014)

Abstract

The influence of quenching temperature, heating atmosphere and annealing time on superconducting characteristics has been studied for $\text{LuBa}_2\text{Cu}_3\text{O}_{7-z}$ compound which has been recently synthesized in a nearly phase-pure form. Resistivity measurements for the as-prepared sample heated at 300 °C in oxygen and subsequently quenched into liquid nitrogen revealed that there is no significant change in T_c . On the other hand, T_c of the sample slightly increased when the sample was heated at 300 °C either in air or in N_2 atmosphere. It was also found that T_c of the sample decreased when the annealing temperature in N_2 atmosphere increased above 400 °C. The experimental results indicated that the as-prepared sample is under overdoped state. The enhanced superconducting transition observed in the samples after heating at 300 °C in air or N_2 atmosphere was discussed in conjunction with a slight removal of oxygen and ordering of oxygen atoms in the as-prepared sample.

Keywords: $\text{LuBa}_2\text{Cu}_3\text{O}_{7-z}$, Transition temperature, Overdoping, Heat treatment

1. INTRODUCTION

Soon after the discovery of high T_c superconductivity in $\text{YBa}_2\text{Cu}_3\text{O}_{7-z}$, it was found that Y in $\text{YBa}_2\text{Cu}_3\text{O}_{7-z}$ (Y-123) can be replaced by most of the rare earth (R) elements to form superconductors of the same crystal structure with T_c near 90 K except for R = Pr, Ce, Tb and Lu [1, 2]. The $\text{PrBa}_2\text{Cu}_3\text{O}_{7-z}$ cuprate is known to be non-superconducting while the orthorhombic compounds of $\text{CeBa}_2\text{Cu}_3\text{O}_{7-z}$ and $\text{TbBa}_2\text{Cu}_3\text{O}_{7-z}$ does not form by solid-state reaction methods [3]. On the other hand, it has been shown that $\text{LuBa}_2\text{Cu}_3\text{O}_{7-z}$ prepared by standard solid-state reaction contains a second phase that accounts for at least half the intensity in the powder X-ray diffraction patterns [1, 2].

Hodorowicz et al. [4] reported an extensive investigation on the phase diagram of Lu-Ba-Cu-O system and concluded that the Lu-123 phase cannot be synthesized by direct combination of the reactants at temperatures between 900 and 1000 °C both in air and in pure O_2 . However, it has been recently discovered that nearly phase-pure Lu-123 compound can be formed within a narrow range of oxygen partial pressure in the specific temperature range [5, 6]. In this work, we have investigated the effects of heat-treatment on superconducting properties of the newly synthesized phase-pure Lu-123 samples to obtain further information for optimization of superconducting properties of the sample.

2. EXPERIMENTALS

Polycrystalline samples with nominal composition of $\text{LuBa}_2\text{Cu}_3\text{O}_{7-z}$ were synthesized by a solid-state reaction technique described elsewhere [5]. The temperature dependence of the electric resistivity was measured by a standard 4-probe technique with silver paste as contacts within a temperature range of 10 K to room temperature. The room-temperature thermoelectric power was measured relative to copper electrodes by using a dc differential method and further calculated by correcting for the thermoelectric power of copper. Thermo-gravimetric analysis was carried out using a TA Inst. SDT-2960. The weight of the sample for analysis was about 20 mg and the heating rate was 10 °C/min in air atmosphere.

2. RESULTS AND DISCUSSION

Fig.1 shows the temperature dependence of electrical resistivity for $\text{LuBa}_2\text{Cu}_3\text{O}_{7-z}$ annealed at 300 °C for 10 h and 500 °C for 20 h in oxygen and subsequently quenched into liquid nitrogen. The as-prepared sample exhibits the onset transition temperature, $T_c(\text{onset}) = 91.3$ K and zero-resistivity temperature, $T_c(\text{zero}) = 86.1$ K and it can be seen that $T_c(\text{zero})$ of the sample hardly changed (within 1 K) after the annealing in oxygen atmosphere. On the other hand, when the sample was heated at 300 °C in air atmosphere, $T_c(\text{zero})$ of the sample increased with the increase in the annealing times as shown in Fig.2. The

* Corresponding author: hklee221@kangwon.ac.kr

$T_c(\text{zero})$ values of the samples after annealing at 300 °C for 3 h and 14.2 h are 88.7 K and 91.0 K, respectively.

Fig.3 reveals the temperature dependence of electrical resistivity for $\text{LuBa}_2\text{Cu}_3\text{O}_{7-z}$ samples annealed at various temperatures in nitrogen and subsequently quenched into liquid nitrogen. This figure indicates that $T_c(\text{zero})$ at first increases slightly after annealing at 300 °C and then decreases with increasing annealing temperature above 400 °C.

For p-type cuprate superconductors, T_c is known to vary universally and empirically as an inverted parabolic function [7] of the hole concentration (p) in the superconducting planes with maximum T_c at $p = 0.16$. Both

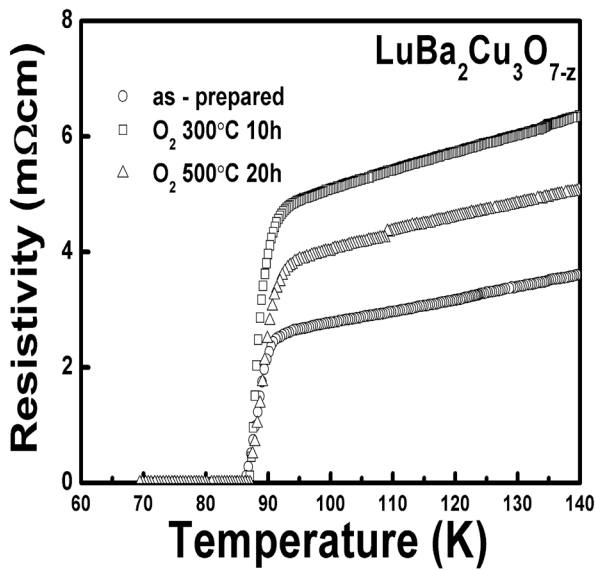


Fig. 1. Temperature dependence of electrical resistivity for $\text{LuBa}_2\text{Cu}_3\text{O}_{7-z}$ samples annealed at 300 °C for 10 h and 500 °C for 20 h in oxygen and subsequently quenched into liquid nitrogen.

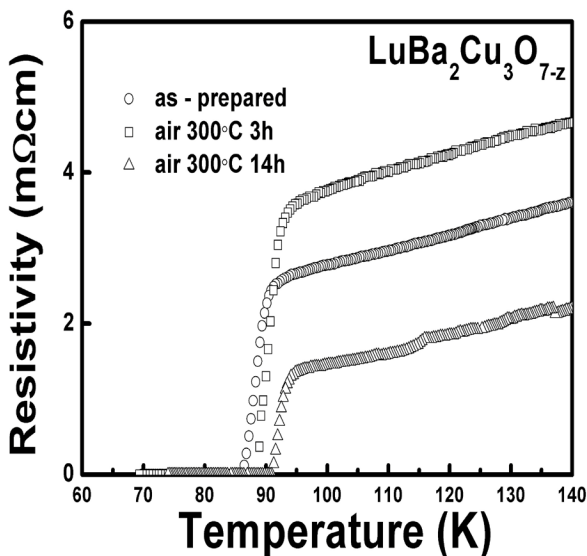


Fig. 2. Temperature dependence of electrical resistivity for $\text{LuBa}_2\text{Cu}_3\text{O}_{7-z}$ samples annealed at 300 °C for 3 h and 14.3 h in air and subsequently quenched into liquid nitrogen.

the change in oxygen content and the substitution of aliovalent elements in R-123 system bring about a change in hole concentration of the CuO_2 planes. Therefore, the variation of the critical temperature after thermal treatment under N_2 atmosphere may be associated with the change in oxygen content. To find more information on the change in oxygen content, a thermogravimetric analysis (TGA) was carried out on the as-prepared sample.

From Fig. 4, one can notice that the weight loss of the as-prepared sample takes place at temperature above 200 °C. The weight loss in the temperature between 200-400 °C is 0.24%. But the weight loss above about 400 °C becomes prominent as shown in Fig.4. Therefore, the decrease in T_c after annealing above 400 °C can be attributed to a decrease in oxygen content of the sample. A

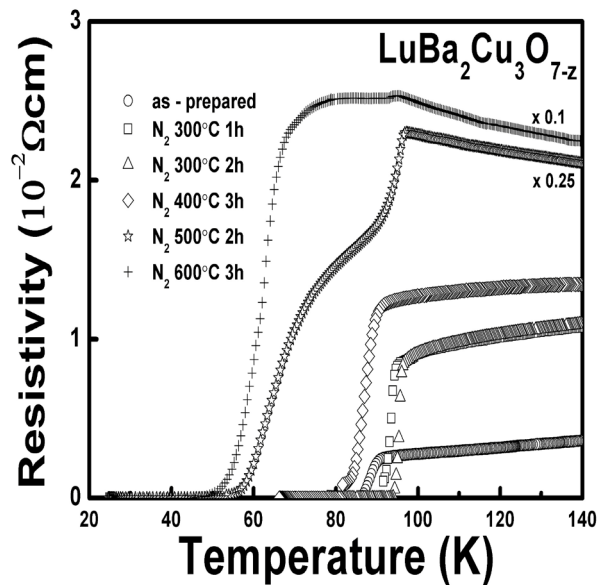


Fig. 3. Temperature dependence of electrical resistivity for $\text{LuBa}_2\text{Cu}_3\text{O}_{7-z}$ samples annealed at various temperature in nitrogen and subsequently quenched into liquid nitrogen.

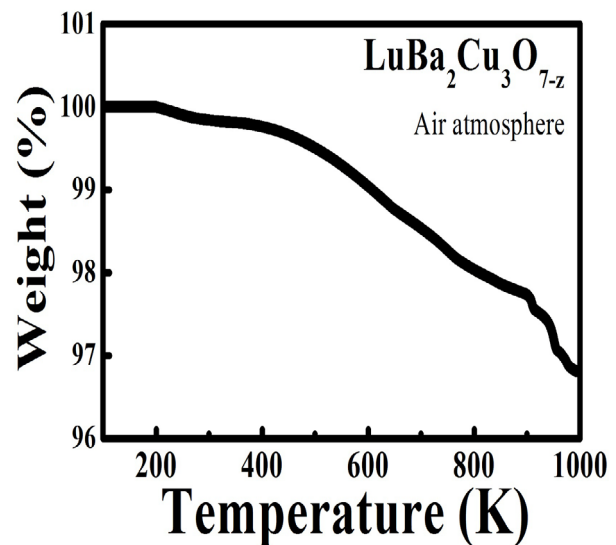


Fig. 4. Thermogravimetric curve for as-prepared $\text{LuBa}_2\text{Cu}_3\text{O}_{7-z}$. The heating rate was 10 °C/min in air.

decrease in oxygen content in cuprate superconductor is known to result in a decrease in hole concentration in the CuO_2 planes. In order to address the change in hole concentration of the sample, room temperature thermoelectric power (TEP) measurements were performed because TEP is known to vary in a systematic manner according to the carrier concentration [8]. It has large positive values in the underdoped range and negative values in the overdoped range at room-temperature, with optimal doping being 1-2 $\mu\text{V}/\text{K}$. The room-temperature TEP for the as-prepared sample was -3.0 $\mu\text{V}/\text{K}$ and the room-temperature TEPs for the samples after N_2 annealing at 300 °C for 1 h, 300 °C for 2 h, 400 °C for 3 h, 500 °C for 2 h and 600 °C for 3 h were 6.6, 6.1, 33.3, 42.9, and 61.4 $\mu\text{V}/\text{K}$, respectively. This result suggests that the as-prepared sample is overdoped state and N_2 annealing results in a decrease in hole concentration of the as-prepared sample. Therefore, the improvement of T_c after annealing at 300 °C in N_2 atmosphere can be explained by the decrease in hole concentration from the overdoped state and thereby reaching nearly optimum hole concentration. A similar change in T_c and hole concentration was also observed in overdoped $(\text{Lu}_{0.8}\text{Ca}_{0.2})\text{Ba}_2\text{Cu}_3\text{O}_z$ system [9, 10] after heat treatment under different temperatures and oxygen partial pressures including N_2 atmosphere.

On the other hand, the room-temperature TEPs for the samples after annealing at 300 °C for 3 h and 14.2 h in air were -3.1 and -3.0 $\mu\text{V}/\text{K}$, respectively. This result suggests a negligible change in hole concentration of the sample after the heat treatments in air. Therefore, in contrast to the case of the samples annealed at 300 °C in nitrogen atmosphere, the improvement of $T_c(\text{zero})$ for the samples annealed at 300 °C annealed in air cannot be attributed to the change in oxygen concentration. Related to this, Jorgensen *et al.* [11] reported that superconducting transition temperature in Y-123 system can be increased due to the local ordering of oxygen atoms with no significant changes in the average occupancies of oxygen sites. Therefore, the increase in $T_c(\text{zero})$ without significant change in hole concentration after annealing in air at low temperature of 300 °C is considered to be related with the local ordering of oxygen atoms.

2. CONCLUSIONS

The effects of heat treatment on superconducting properties of newly synthesized $\text{LuBa}_2\text{Cu}_3\text{O}_{7-z}$ have been investigated by characterizing the sample after annealing under various atmospheres, temperatures and times. It was observed that annealing under oxygen atmosphere below 500 °C hardly changed the superconducting transition temperature, while an annealing at 300 °C under air or N_2 atmospheres led to a slight increase in $T_c(\text{zero})$. We also observed that T_c of the Lu-123 sample decreased through the removal of oxygen when the sample is subjected to high temperature annealing above 400 °C in N_2 atmosphere. The increase in T_c observed for the sample annealed at 300 °C in N_2 atmosphere could be attributed to a slight deficiency

in oxygen. However, the improvement of $T_c(\text{zero})$ for the samples annealed at 300 °C annealed in air could not be attributed to the change in hole concentration. From these results, it was concluded that the as-prepared sample is overdoped state.

ACKNOWLEDGMENT

The authors thanks to the central laboratory of Kangwon National University for the help in XRD and TGA measurements. This work was partly supported by Korea Research Foundation Grant (KRF-2012R1A1A2042519).

REFERENCES

- [1] P. H. Hor, R. L. Meng, Y. Q. Wang, L. Gao, Z. J. Huang, J. Bechtold, K. Forster, and C. W. Chu, "Superconductivity above 90 K in the square-planar compound system $\text{ABa}_2\text{Cu}_3\text{O}_{6+x}$ with $A = \text{Y, La, Nd, Sm, Eu, Gd, Ho, Er, and Lu}$," *Phys. Rev. Lett.*, vol. 58, pp.1891-1894, 1987.
- [2] J. M. Tarascon, W. R. Mckinnon, L. H. Greene, G. W. Hull, and E. M. Vogel, "Oxygen and rare-earth doping of the 90 K superconducting perovskite $\text{YBa}_2\text{Cu}_3\text{O}_{7-x}$," *Phys. Rev. B*, vol. 36, pp.226-234,1987.
- [3] K. N. Yang, B. W. Lee, and M. B. Maple, "Compounds in Mixed Phase $\text{CeBa}_2\text{Cu}_3\text{O}_y$ and $\text{TbBa}_2\text{Cu}_3\text{O}_y$ Systems," *Appl. Phys. A*, vol. 46, pp.229-232, 1988.
- [4] E. Hodorowicz, S. A. Hodorowicz and H. A. Eick, "The Yb_2O_3 - and Lu_2O_3 - $\text{BaO}(\text{BaCO}_3)$ - CuO systems: compounds and phase compatibilities in air at 940 – 980 °C," *J. Alloys Compound.*, vol. 181, pp.445-456, 1992.
- [5] H. K. Lee, "Effect of Oxygen Partial Pressure on Phase Formation and Stability of $\text{LuBa}_2\text{Cu}_3\text{O}_{7-x}$ Compounds," *J Supercond. Nov. Magn.*, vol. 25, pp. 2519-2522, 2012.
- [6] M. S. Lee, H. K. Lee, and S. I. Seok, "Synthesis and Characterization of $\text{LuBa}_2\text{Cu}_3\text{O}_z$ Superconductors by Using of a Polymeric Precursor", *New Physics: Sae Mulli*, vol. 63, No. 10, pp.1176-1179, 2013.
- [7] J. L. Tallon and G. V. M. Williams, Comment on "Structural and superconducting properties of $\text{R}_{1-x}\text{Ca}_x\text{Ba}_2\text{Cu}_3\text{O}_{7-\delta}$ with $0.50 \geq x \geq 0.00$," *Phys. Rev. B*, vol. 61, pp. 9820-9823, 2000.
- [8] J. L. Tallon, C. Bernhard, H. Shaked, R. L. Hitterman, and J. D. Jorgensen, "Generic superconducting phase behavior in high- T_c cuprates: T_c variation with hole concentration in $\text{YBa}_2\text{Cu}_3\text{O}_{7-\delta}$," *Phys. Rev. B*, vol. 51, pp.12911-12914, 1995.
- [9] H. K. Lee and Y. H. Kim, "Effect of Ca Substitution on the Superconducting Properties of $\text{LuBa}_2\text{Cu}_3\text{O}_z$," *J. Korean Phys. Soc.*, vol. 48, pp. 1151-1154, 2006.
- [10] H. K. Lee, "Effect of Co doping on the superconducting properties of overdoped $(\text{Lu}_{0.8}\text{Ca}_{0.2})\text{Ba}_2\text{Cu}_3\text{O}_z$," *Advances Sci. Tech.*, vol. 47, pp 108-112, 2006.
- [11] J. D. Jorgensen, S. PEI, P. Lightfoot, H. SHI, A. P. Paulikas, and B. W. Veal, "Time-Dependent Structural Phenomena at Room Temperature in Quenched $\text{YBa}_2\text{Cu}_3\text{O}_{6.41}$," *Physica C*, vol. 167, pp. 571-578, 1990.