

# Effect of Additives on the Refractive Index of $B_2O_3$ - $SiO_2$ - $Al_2O_3$ Glasses for Photolithographic Process in Electronic Micro Devices

Juyeon Won\*, Seongjin Hwang, Jungki Lee and Hyungsun Kim<sup>†</sup>

School of Materials Engineering, Inha University, 253 Younyun-dong, Nam-gu, Incheon 402-751, Korea

\*Now she works at SSCP Co., Seongnam, Korea

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**Abstract** In fabricating plasma display panels, the photolithographic process is used to form patterns of barrier ribs with high accuracy and high aspect ratio. It is important in the photolithographic process to control the refractive index of the photosensitive paste. The composition of this paste for photolithography is based on the  $B_2O_3$ - $SiO_2$ - $Al_2O_3$  glass system, including additives of alkali oxides and rare earth oxides. In this work, we investigated the density, structure and refractive index of glasses based on the  $B_2O_3$ - $SiO_2$ - $Al_2O_3$  system with the addition of  $Li_2O$ ,  $K_2O$ ,  $Na_2O$ ,  $CaO$ ,  $SrO$ , and  $MgO$ . The refractive index of the glasses containing  $K_2O$ ,  $Na_2O$  and  $CaO$  was similar to that of the [BO3] fraction while that of the  $SrO$ ,  $MgO$  and  $Li_2O$  containing glasses were not correlated with the coordination fraction. The coordination number of the boron atoms was measured by MAS NMR. The refractive index increased with a decrease of molar volume due to the increase in the number of non-bridging oxygen atoms and the polarizability. The lowest refractive index (1.485) in this study was that of the  $B_2O_3$ - $SiO_2$ - $Al_2O_3$ - $K_2O$  glass system due to the larger ionic radius of  $K^+$ . Based on our results, it has been determined that the refractive index of the  $B_2O_3$ - $SiO_2$ - $Al_2O_3$  system should be controlled by the addition of alkali oxides and alkali earth oxides for proper formation of the photosensitive paste.

**Key words** borosilicate glass, nuclear magnetic resonance, refractive index.

## 1. Introduction

In the display industry, many studies have recently been conducted to achieve full-high definition (HD) with plasma display panels (PDPs). Since the barrier ribs prevent crosstalk among the phosphors (RGB) and maintain the discharge space in PDPs. It is necessary for the barrier ribs to have a pattern with a high aspect ratio.<sup>2-3)</sup> The photolithographic process can be used to form patterns of barrier ribs with high accuracy and a high aspect ratio.

For the photolithographic process, it is necessary for the glass frits and organic components to have a similar refractive index. The refractive index of the glass frit should be in the range of 1.4~1.7 because the refractive index of the organic components is commonly in the range of 1.4~1.7. Therefore, it is important to make the glass frit have a lower refractive index, because most glasses have a high refractive index of more than 1.7.<sup>3)</sup> Moreover in photolithographic process the refractive index of the photosensitive paste is a significant factor for achieving high accuracy and a high aspect ratio because in order to reduce the degree of reflection and scattering during the exposure process.

The general composition used for the photolithographic process is based on the  $B_2O_3$ - $SiO_2$ - $Al_2O_3$  glass system, including additives such as alkali oxides ( $Li_2O$ ,  $Na_2O$ ,  $K_2O$ ) and alkali earth oxides ( $SrO$ ,  $MgO$ ,  $CaO$ ).<sup>3)</sup> The refractive index and thermal properties of the glass system are changed by the addition of the alkali oxides and alkali earth oxides. These additives produce non-bridging oxygen in the glass network, causing its density to change. In addition to the change of the structural cross-link density, the refractive index, dielectric and thermal properties of the glass are correlated with the ionic radius and polarizability of the cations. There have been several studies of the  $B_2O_3$ - $SiO_2$ - $Al_2O_3$  glass system.<sup>4-7)</sup> As recently reported, the properties of the glasses in the  $Na_2O$ - $Al_2O_3$ - $B_2O_3$ - $SiO_2$  system are also strongly affected by the concentrations of the main glass components.<sup>4)</sup> It has been shown that the chemical durability and activation energy are increased and the ionic conductivity decreased by the addition of  $MgO$ ,  $CaO$ ,  $BaO$  or  $ZnO$  to the  $Na_2O$ - $Al_2O_3$ - $B_2O_3$ - $SiO_2$  glass system.<sup>5)</sup> This is due to the strengthening of the glass network by the  $R^{2+}$  cations.<sup>6)</sup> Also, with increasing  $BaO$  content in the  $B_2O_3$ - $Al_2O_3$ - $SiO_2$  glass system, [BO3] gradually changes to [BO4], and [AlO5] and [AlO6] change to [AlO4].<sup>7)</sup> However, the relation between the network and structure of the glasses and their refractive index remains unclear, because previous studies were focused on the effect of the com-

<sup>†</sup>Corresponding author  
E-Mail : kimhs@inha.ac.kr (H. Kim)

position on the individual properties.

In this work, we investigated the effect of the composition of the  $B_2O_3$ - $Al_2O_3$ - $SiO_2$  glass system containing alkali and alkali earth oxides on its network and structure. We measured the density, and refractive index of this glass system, as well as the coordination number of boron atoms by MAS NMR, in order to show that the refractive index of the glass is correlated with the composition of the in glass, especially the amount of additives.

## 2. Experimental Procedure

In order to prepare the various batches of glass, high purity powders of  $B_2O_3$ ,  $Al_2O_3$ ,  $SiO_2$ ,  $Li_2O$ ,  $Na_2O$ ,  $K_2O$ ,  $MgO$ ,  $CaO$  and  $SrO$  (99.9%, Aldrich, USA) were mixed. We prepared six series of glass samples with various amounts of alkali and alkali earth oxides, which are referred to herein as BS, BK, BN, BC, BM and BL, respectively. The compositions of the studied glasses are presented in Table 1. The raw materials were melted in a platinum crucible at 1200-1500°C. After heating them for 30 min in an electric furnace, the glass melts were quenched into a stainless roller to make glass cullets. To make bulk samples, the cullets were remelted in an electric furnace and then put into a square graphite mold and annealed for 1h at 465-660°C. The density of the samples was measured using the Archimedes method at room temperature. In order to take into account the effect of the sample size and surface condition, the density was measured 5 times to obtain a mean value.

Using a multi-wavelength Abbe refractometer (DR-M2, ATAGO, Tokyo, Japan), the refractive index was measured 5 times at wavelengths of 480 nm, 546 nm and 589 nm with samples having dimensions of  $(15-40) \times (6-8) \times (1-10)$  mm. Also, the refractive index of glasses was predicted by the Demkina-76 (Si-I) equation (1).<sup>7-8)</sup>

$$K = \frac{\frac{a_1}{S_1}k_1 + \frac{a_2}{S_2}k_2 + \dots + \frac{a_i}{S_i}k_i}{\frac{a_1}{S_1} + \frac{a_2}{S_2} + \dots + \frac{a_i}{S_i}} \quad (1)$$

Where  $K$  is the calculated value of the refractive index.  $a_1, a_2, a_3, \dots, a_i$ , the wt. percentages of oxides in the glass,

$S_1, S_2, S_3, \dots, S_i$ , the structural factors for the individual oxides, viz the coefficient giving the conversion of the mass units to units of volume, for which linearity remains valid in a wide range of the composition,

$k_1, k_2, k_3, \dots, k_i$ , the factors for the index of refraction value of the individual oxides.

The  $^{11}B$  MAS NMR spectra were obtained on an FT-NMR spectrometer (Varian unitynova 400, Varian, Palo Alto, CA) operating at 400MHz.

## 3. Results and Discussion

The calculated and measured refractive indexes were shown in Fig. 1. When  $R_2O$  and  $RO$  were added to the  $B_2O_3$ - $SiO_2$ - $Al_2O_3$  glass, the calculated and experimentally measured refractive indices were found to be similar up to 17 mol% of  $R_2O$  and  $RO$ , and then to increasingly deviate from each other ( $\pm 0.02$ - $0.04$ ). It is expected that change of the composition have influence on the refractive index in glasses.

The obtained spectra by the  $^{11}B$  MAS NMR are presented in Fig. 2. The [BO3] and [BO4] peaks are observed at about 25 and 75ppm. It shows a large, relatively narrow peak from [BO4] and a small broader peak from [BO3] being representative of the  $^{11}B$  MAS NMR spectra for all

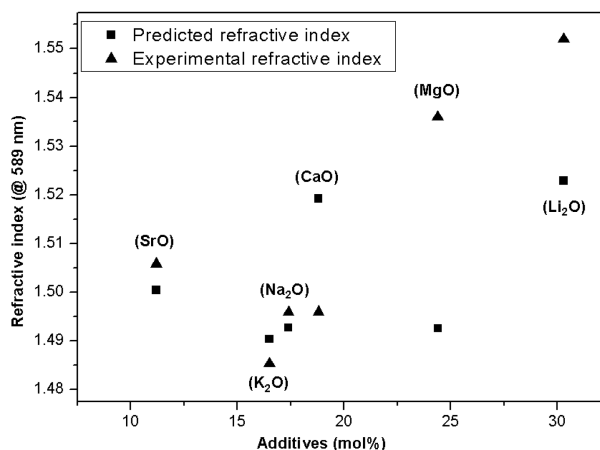
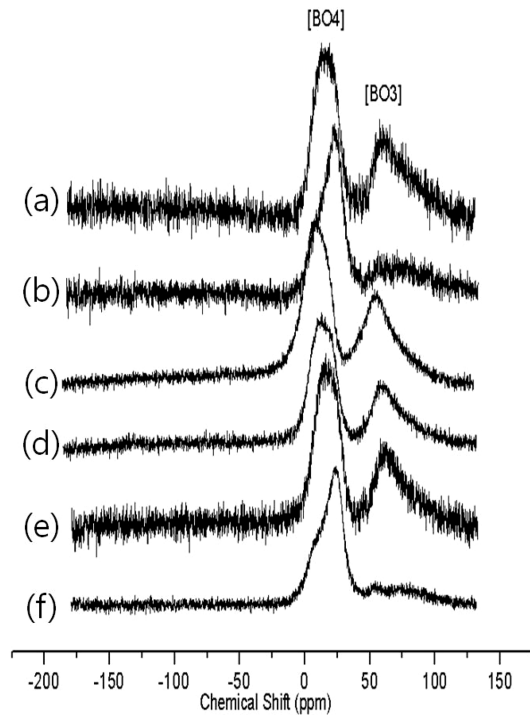


Fig. 1. Calculated and experimentally measured refractive indices for the different kinds of additives.

Table 1. Experimental compositions (in mol%).

	$B_2O_3$	$SiO_2$	$Al_2O_3$	$SrO$	$K_2O$	$Na_2O$	$CaO$	$MgO$	$Li_2O$
BS	44.3	25.7	18.9	11.2					
BK	41.6	24.1	17.8		16.5				
BN	41.2	23.9	17.6			17.4			
BC	40.5	23.4	17.3				18.8		
BM	37.7	21.8	16.1					24.4	
BL	34.7	20.1	14.8						30.3



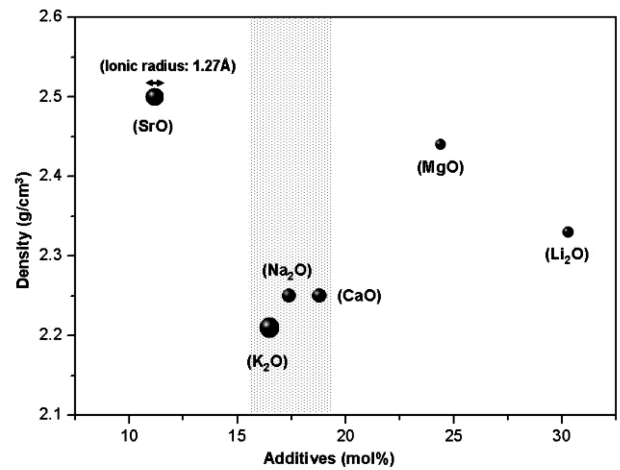
**Fig. 2.**  $^{11}\text{B}$  MAS NMR spectra for  $\text{B}_2\text{O}_3\text{-SiO}_2\text{-Al}_2\text{O}_3\text{-(RO or R}_2\text{O)}$  with various content of additives: (a) 30.3 of  $\text{Li}_2\text{O}$ , (b) 24.4 of  $\text{MgO}$ , (c) 18.8 of  $\text{CaO}$ , (d) 17.4 of  $\text{Na}_2\text{O}$ , (e) 16.5 of  $\text{K}_2\text{O}$  and (f) 11.2 of  $\text{SrO}$  in mol%, respectively.

**Table 2.** Composition and relative concentration of [BO4] and [BO3] determined by calculating the coordination fraction of the NMR spectra for all of the glass systems.

Sample	$\text{B}_2\text{O}_3$ (mol%)	Additive (mol%)	[BO4] (%)	[BO3] (%)
BS	44.3	11.2	94	6
BK	41.6	16.5	72	28
BN	41.2	17.4	66	34
BC	40.5	18.8	63	37
BM	37.7	24.4	93	7
BL	34.7	30.3	67	33

glasses system.<sup>9-12)</sup>

In order to characterize the relationship between the refractive index and the polarizability in glasses, we calculated the fraction of four-coordinated boron atoms,  $N_4$ , by quantitative analysis using the NMR technique. Table 2 shows the calculated coordination fractions of boron atoms determined by NMR. It is known that the refractive index increases with increasing polarizability of the glass, because the polarizability increased by increasing the amount of non-bridging oxides with adding the alkali and alkali earth oxides.<sup>13)</sup> Therefore, the NMR spectra of this glass system enabled us to identify the relationship between the refractive index and polarizability.



**Fig. 3.** Density of the  $\text{B}_2\text{O}_3\text{-SiO}_2\text{-Al}_2\text{O}_3$  glasses with  $\text{R}_2\text{O}$  and  $\text{RO}$ .

With increasing content of alkali and alkali earth, the shift of the coordination occurs from [BO3] to [BO4] and non-bridging oxides are not formed by the [BO4] tetrahedron. The structure is strengthened, because the number of points of linkage of the polyhedrons increases from three to four. When the fraction of [BO4] to the total number of boron exceeds the maximum value, the opposite coordination shift of [BO4]  $\rightarrow$  [BO3] occurs at higher alkali and alkali earth contents and the structure of the glasses is weakened by the formation of [BO3] groups with non-bridging oxygens.<sup>9,14-15)</sup>

In Table 2, the [BO3] fraction shows a similar tendency to the refractive index of the  $\text{B}_2\text{O}_3\text{-SiO}_2\text{-Al}_2\text{O}_3$  glass system to which  $\text{K}_2\text{O}$ ,  $\text{Na}_2\text{O}$  and  $\text{CaO}$  are added. However, the variation of the refractive index of the glass system to which  $\text{SrO}$ ,  $\text{MgO}$  and  $\text{Li}_2\text{O}$  are added is not correlated with the coordination fraction. This is attributed to the difference in the amount of  $\text{B}_2\text{O}_3$ .

The density is not a suitable parameter to use to determine the correlation between the properties of the glass system and its composition, as shown in Fig. 3. Therefore, the molar volume ( $V_M$ ) was used, which was calculated from the experimental density ( $\rho$ ) using the equation (2)

$$V_M = \frac{M_T}{\rho} \quad (2)$$

where  $M_T$  is the total molecular weight of the components in the glass and it is given by equation (3)

$$M_T = x_1Z_1 + x_2Z_2 + \dots + x_iZ_i \quad (3)$$

where  $x_i$  is the mole fraction of the constituent oxide and  $Z_i$  is the molecular weight of the constituent oxide.<sup>15)</sup>

The refractive index is not dependent on the polarizability, but is related to the molar volume. As shown in Fig. 1 and Fig. 4, the refractive index and molar volume are

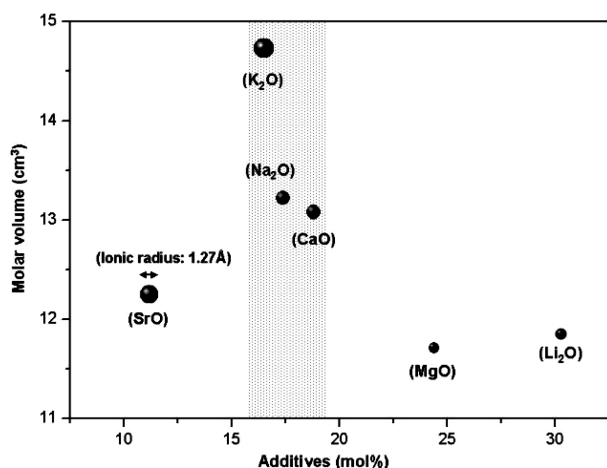


Fig. 4. Molar volume of the  $B_2O_3$ - $SiO_2$ - $Al_2O_3$  glasses with  $R_2O$  and  $RO$ .

correlated with each other. The molar volume determines the refractive index behavior of the glass. The refractive index increases with decreasing molar volume due to the increase in the number of non-bridging oxygen. In this case, the effect of the molar volume is relatively strong. The refractive index was observed to be in inverse proportion to the molar volume. The  $B_2O_3$ - $SiO_2$ - $Al_2O_3$ - $K_2O$  glass system was observed to have a higher molar volume than the  $B_2O_3$ - $SiO_2$ - $Al_2O_3$ -( $K_2O$ ,  $Na_2O$  or  $CaO$ ) system with a similar alkali ( $K_2O$ ,  $Na_2O$ ,  $CaO$ ) content. It is thought that the network is expanded by the larger ionic radius of  $K^+$ .<sup>13)</sup> Therefore, the glass system with  $K_2O$  has the lowest refractive index.

#### 4. Conclusion

We investigated the density, structure and refractive index of glasses based on the  $B_2O_3$ - $Al_2O_3$ - $SiO_2$  system with the addition of  $R_2O$  and  $RO$ . The refractive index was found to depend on the molar volume and four-coordination fraction of boron atoms. It was examined determined by NMR analysis that the refractive index is related to the molar volume of the glass as well as the polarizability of the ions. When the glasses have a similar

content of composition, the refractive index depends on the polarizability which is determined by the amount of non-bridging oxygen's. However, the molar volume is correlated with the refractive index irrespective of the composition.

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