

## Temperature-Dependent Photoluminescence from Er-implanted undoped and Mg-doped GaN

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Selectively excited photoluminescence (PL) spectroscopy has been carried out on the ~1540 nm  $^4I_{13/2}$  to  $^4I_{15/2}$  emissions of the multiple  $Er^{3+}$  centers observed in Er-implanted *undoped* and *Mg-doped* GaN at temperatures ranging from 6K to 295K. The temperature dependence of the  $Er^{3+}$  PL spectra selectively excited by below-gap light demonstrates different quenching rates for the distinct  $Er^{3+}$  PL centers, and indicates that the PL spectra with the most rapid thermal quenching rates do not contribute to the room temperature, above-gap-pumped  $Er^{3+}$  PL spectrum. In addition, selective PL spectroscopy has been carried out on the  $Er^{3+}$  emission in Er-implanted undoped and Mg-doped GaN at temperatures ranging from 6K to 295K. The results indicate that the previously reported enhancement of the violet-pumped centers' contribution to the low temperature above-gap excited  $Er^{3+}$  PL in Mg-doped GaN is also evident at room temperature.

**Keywords :** photoluminescence, Er-implanted, Mg-doped, GaN

### 1. INTRODUCTION

In recent years, visible and near infrared emission from the intra-4f shell transitions of ion-implanted and in-situ doped rare earth dopants in GaN has been investigated intensively by photoluminescence (PL) and electroluminescence (EL) techniques.[1-10] Our research has focused on PL studies of Er-implanted GaN as a function of excitation wavelength and annealing temperature which was detected nine different  $Er^{3+}$  emission centers on the basis of characteristic differences in their sharply structured ~1540 nm  $^4I_{13/2}$  to  $^4I_{15/2}$   $Er^{3+}$  PL spectra. Selective optical excitation of these centers is accomplished[4,5,8,9] by choosing a pump wavelength within one of the broad, below-gap, defect-related absorption bands observed in the  $Er^{3+}$  photoluminescence excitation (PLE) spectra, with subsequent nonradiative energy transfer to the 4f electrons of the neighboring  $Er^{3+}$ .

Above-gap excitation at low temperature excites a sub-set of these centers with the relative strengths of their emissions determined by the concentration and capture cross sections of each distinctive center. Only one of these centers can be excited efficiently by direct intra-4f-shell absorption (~809 nm  $^4I_{15/2}$  to  $^4I_{9/2}$ ), indicating that it is the highest concentration center, representing over 99% of the optically active Er atoms.[5,8,9] Remarkably, these "4f-pumped" centers are

not strongly pumped by above-gap light which implies that less than 1% of the Er atoms is involved in the Er-defect complexes which mediate the excitation of the  $Er^{3+}$  emission spectra pumped by above-gap absorption.[8,9] Recently, we demonstrated that the  $Er^{3+}$  PL and PLE spectra of Er-implanted Mg-doped GaN exhibit a significant enhancement of the so-called "violet-pumped"  $Er^{3+}$  emission spectrum that is selectively excited by a characteristic ~2.8-3.4 eV (violet) below-gap PLE band.[8,9] More importantly, the violet-pumped PL center dominates the above-gap excited  $Er^{3+}$  PL spectrum of Er-implanted Mg-doped GaN, whereas it was nearly unobservable under the above-gap excitation in Er-implanted undoped GaN.[8,9]

In the present work, selective PL spectroscopy has been carried out on ~1540 nm  $Er^{3+}$  emission in Er-implanted undoped and Mg-doped GaN at temperatures ranging from 6K to 295K. The results indicate that the previously reported enhancement of the violet-pumped centers' contribution to the low temperature above-gap excited  $Er^{3+}$  PL in Mg-doped GaN is also evident at room temperature.

### 2. EXPERIMENTAL RESULTS AND DISCUSSION

Thin films (~1  $\mu$ m thick) of undoped n-type and Mg-

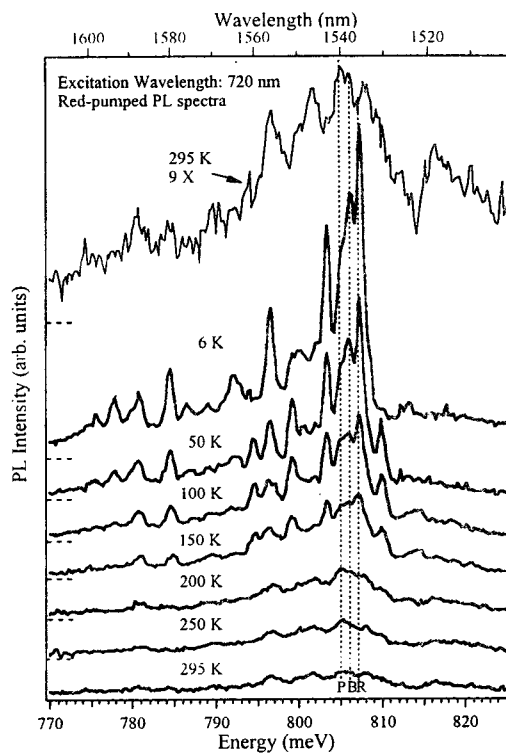
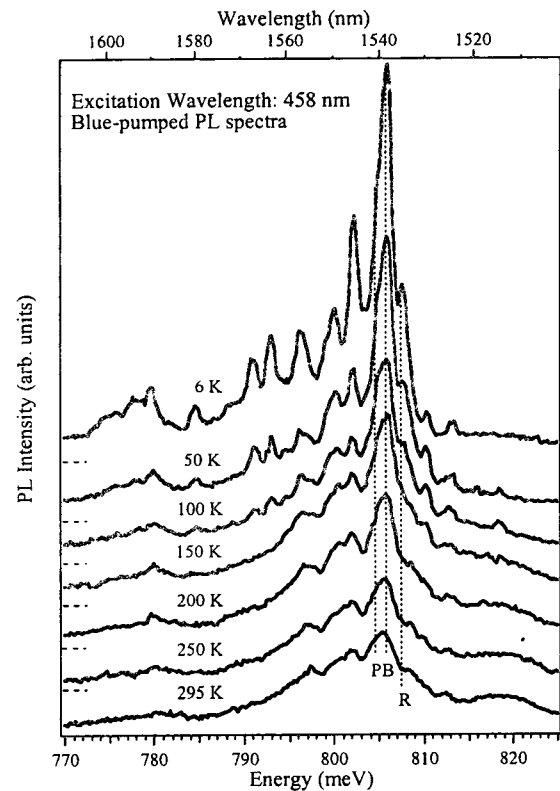


Fig. 1. The red-pumped  $\text{Er}^{3+}$  PL from undoped Er-implanted GaN at selected temperatures from 6 K to 295 K.

doped p-type GaN were grown on (0001) sapphire substrates by atmospheric pressure metal-organic chemical vapor deposition (MOCVD) techniques that have been described previously[8]. The as-grown Mg-doped GaN samples are highly resistive, with typical hole concentrations of  $\sim 5 \times 10^{17} \text{ cm}^{-3}$  after thermal annealing. It was estimated that the concentration of electrically active Mg dopants was  $\sim 5 \times 10^{19} \text{ cm}^{-3}$  before the Er implantation. The Mg-doped GaN film was implanted with a dosage of  $4 \times 10^{13} \text{ Er}^{3+} \text{ ions/cm}^2$  at 280keV, with the projected range for the 280-keV Er ions estimated to be approximately 0.05  $\mu\text{m}$ . The implanted samples were annealed at 900  $^{\circ}\text{C}$  for 90 minutes under a continuous flow of nitrogen gas. The  $\text{Er}^{3+}$  PL spectra were excited by a variety of lasers and lamps with wavelengths chosen to match specific below-gap PLE bands, as described in detail elsewhere.[4,5,8,9] The different  $\text{Er}^{3+}$  PL spectra have been labelled according to the wavelength of the pump light employed, i.e. blue-pumped (458 nm), red-pumped (633 nm), violet-pumped (404 nm), etc.[5,8,9] Luminescence was analyzed by a 1-m single grating monochromator and detected by a cooled Ge PIN detector. The Er-implanted GaN samples were mounted in a Janis Superveritemp liquid helium cryostat, and the PL spectroscopies were carried out at temperatures in the range from 6 K to 295K.

The temperature dependences of the integrated  $\text{Er}^{3+}$



PL intensity measured for each of the violet-, blue-, red-,

Fig. 2. The blue-pumped  $\text{Er}^{3+}$  PL from undoped Er-implanted GaN at selected temperatures from 6 K to 295 K.

4f-, and above-gap-pumped PL spectra in the undoped GaN have been reported previously[5]. The violet- and blue-pumped PL spectra exhibit similar temperature dependences, reaching a reduction of a factor of  $\sim 2$  in integrated intensity as the temperature increases from 6 K to 295 K. The red- and 4f-pumped  $\text{Er}^{3+}$  PL spectra experience a more rapid thermal quenching, reaching a reduction of a factor of  $\sim 10$  at 295 K, and the dominant peaks of the 6 K red-pumped PL spectrum are unobservable at temperatures above 150 K, as is evident in Fig. 1. In contrast, the blue-, violet-, and 4-f pumped  $\text{Er}^{3+}$  PL spectra retain the dominant peaks of their 6 K spectra at room temperature, although they are broadened, as shown for the blue-pumped  $\text{Er}^{3+}$  PL in Fig. 2.

The effects of Mg-doping on the above-gap pumped  $\text{Er}^{3+}$  PL spectrum at 6 K and room temperature are compared in Figs. 3 and 4, respectively. The top spectrum of Fig. 3 is the so-called violet-pumped  $\text{Er}^{3+}$  emission excited from Er-implanted undoped GaN by 404 nm light at the peak of the  $\sim 2.8\text{-}3.4 \text{ eV}$  violet PLE band. The sharp peak at  $\sim 798 \text{ meV}$  (1554.8 nm) in the violet pumped  $\text{Er}^{3+}$  PL spectrum is noticeably absent from the above-gap pumped  $\text{Er}^{3+}$  PL spectrum observed in Er-implanted undoped GaN (bottom of Fig. 3). Careful

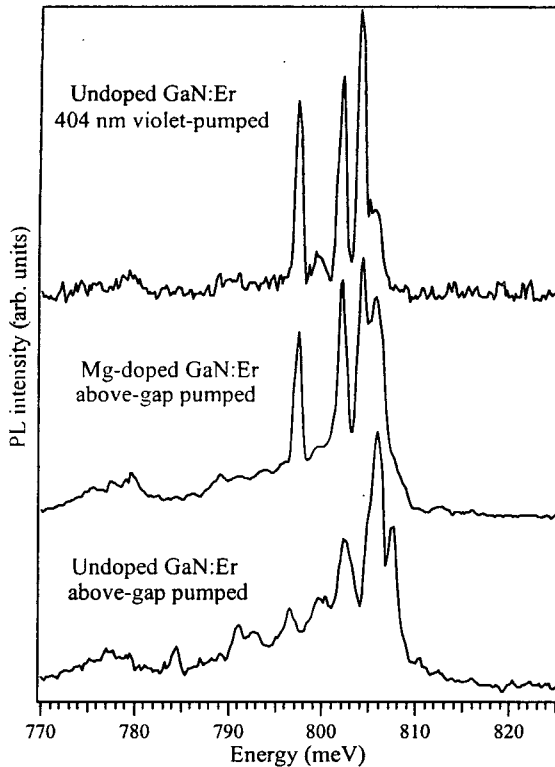


Fig. 3. Violet-pumped  $\text{Er}^{3+}$  PL spectrum excited from Er-implanted undoped GaN, compared with above-gap-pumped  $\text{Er}^{3+}$  PL spectra from Mg-doped and undoped Er-implanted GaN. Fig.2(b).

analysis of this spectrum clearly demonstrated that the distinctive violet-pumped  $\text{Er}^{3+}$  PL spectrum shown in Fig. 3 was not strongly excited by above gap light in the Er-implanted, undoped GaN. In striking contrast, the above-gap pumped  $\text{Er}^{3+}$  PL spectrum from the Er-implanted, Mg-doped GaN sample shown in Fig. 3 is dominated by the strong, distinctive PL peaks that characterize the violet-pumped  $\text{Er}^{3+}$  PL spectrum. Thus the Mg-doping has had the effect of making the violet-pumped  $\text{Er}^{3+}$  emission centers the dominant contributors to the low temperature, above-gap pumped  $\text{Er}^{3+}$  PL spectrum of Mg-doped GaN, whereas they are virtually undetectable in the 6 K above-gap pumped  $\text{Er}^{3+}$  PL spectrum of undoped GaN (Fig. 3(c)).[5,8,9]

Spectra (a) and (b) in Fig. 4 are the above gap-pumped  $\text{Er}^{3+}$  PL spectra obtained at 295 K from the Mg-doped and undoped GaN hosts, respectively. It is evident that the major peak of the Mg-doped spectrum is red shifted by ( $\sim 3$  nm) relative to the 1538 nm peak of the undoped spectrum, which is known to be dominated by contributions from the blue-pumped  $\text{Er}^{3+}$  PL centers.[5,8,9] Weighted subtraction of the undoped spectrum (b) from the Mg-doped spectrum (a) results in the difference spectrum (d). Comparison of the

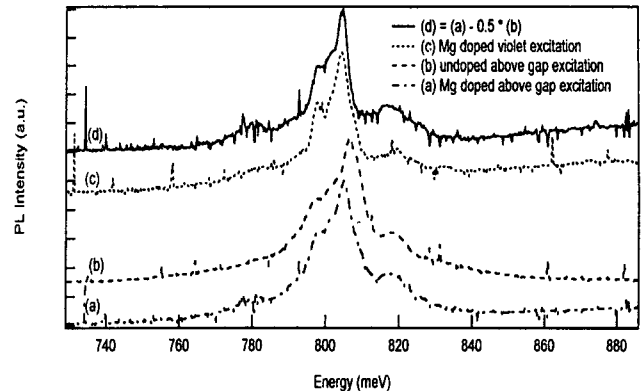


Fig. 4. Spectra (a) and (b) are the above gap-pumped  $\text{Er}^{3+}$  PL spectra obtained at 295 K from the Mg-doped and undoped GaN hosts, respectively. Spectrum (d) is the subtraction of spectrum (b) from spectrum (a). Spectrum (c) is the 295 K violet-pumped  $\text{Er}^{3+}$  PL from the Mg-doped GaN.

difference spectrum (d) with the violet-pumped  $\text{Er}^{3+}$  PL spectrum (c) from the Mg-doped GaN sample reveals that the difference spectrum corresponds to an enhanced contribution of the violet-pumped centers to the room temperature above gap-pumped  $\text{Er}^{3+}$  PL spectrum (a) of the Mg-doped sample. Thus the Mg-doping has greatly enhanced the contribution of the violet-pumped  $\text{Er}^{3+}$  emission centers to the room temperature above-gap pumped  $\text{Er}^{3+}$  PL spectrum of Er-implanted Mg-doped GaN relative to the corresponding spectrum from the undoped GaN. This observation indicates its potential relevance to the performance of room temperature EL devices.

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