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## FABRICATION OF HIGH QUALITY $\text{YBa}_2\text{Cu}_3\text{O}_x$ THIN FILMS USING PULSED LASER DEPOSITION

Eun-Hong Lee, Sang-Jin Park, I-Hun Song,  
Insang Song, Junho Gohng, Junghyun Sok and Jo-Won Lee

*Electronic Materials Lab., Materials Sector,  
Samsung Advanced Institute of Technology, Suwon, Korea*

### ABSTRACT

High quality  $\text{YBa}_2\text{Cu}_3\text{O}_x$  (YBCO) thin films for directly coupled dc superconducting quantum interference device (SQUID) were fabricated by pulsed laser deposition. Several critical parameters have been optimized through systematic studies. Thus, the films showing the  $T_c$  of above 91K and  $J_c$  of above  $2 \times 10^6 \text{A/cm}^2$  at 77K were routinely obtained. Extensive AFM and X-ray diffraction studies have been conducted for morphological and structural analyses. The directly coupled DC-SQUIDS were fabricated from the YBCO thin films deposited on  $\text{SrTiO}_3$  bicrystals under the optimized conditions. The measurement on  $2I_c$  and swing voltage give  $200 \mu\text{A}$  and  $17 \mu\text{V}$  at 77K, respectively.

### INTRODUCTION

Pulsed laser deposition (PLD) has become a popular method for fabricating many types of thin films. Especially in the field of high- $T_c$  superconducting thin films technologies, PLD is one of the most preferred methods because high quality thin films of complicated compositions could be obtained easily at relatively low cost. Furthermore, the PLD technique is advantageous over other existing techniques due to its high deposition rate and the ease of multi-layer stacking in-situ. When fabricating a device, the quality of thin film such as surface morphology and uniformity is as important as electrical properties. For the past few years, there have been

many efforts to improve film properties such as crystal orientation as in-plane alignment in addition to other electrical properties.

In this article, we systematically studied the effects of several processing parameters on electrical properties, crystal orientation and surface morphology of YBCO thin films using PLD. To confirm the film quality we have measured the device characteristics of directly coupled dc SQUID magnetometer using the above films.

### EXPERIMENTAL

A 248-nm KrF excimer laser (pulse duration = 25ns, Lambda Physik, LPX 305i) operating at 5 Hz was used for the film deposition.

tion. The laser beam was focused through a lens to about  $1 \times 4.5 \text{ mm}^2$  size onto a stoichiometric YBCO target with an energy density of about  $1 \text{ J/cm}^2$ . Substrates, mounted on the heater (US Thin Film Pro. Inc.) by silver paste, were placed in front of the target at the distance of  $40 \sim 80 \text{ mm}$ . Substrate temperature ( $T_s$ ) during deposition was varied from  $700$  to  $840^\circ\text{C}$  and the ambient  $\text{O}_2$  pressure ( $\text{PO}_2$ ) was applied in the range of  $100$  to  $700 \text{ mTorr}$ . YBCO films of  $100 \sim 200 \text{ nm}$  thickness were deposited onto  $10 \times 10 \text{ mm}^2$  MgO and  $\text{LaAlO}_3$  substrates of (100) orientation. For the bicrystal grain boundary junction SQUID magnetometer, we used (100)  $\text{SrTiO}_3$  bicrystal with a  $24^\circ$  misorientation angle. The target pellet was rotated during the irradiation to minimize needle-like morphology on its surface<sup>[1]</sup>. Immediately after the deposition, pure oxygen was introduced into the deposition chamber and  $T_s$  was decreased to  $550^\circ\text{C}$  at a rate of  $15^\circ\text{C}/\text{min}$ . After annealing at this temperature for  $45 \text{ min}$ ,  $T_s$  was cooled down to the room temperature. The films were characterized by x-ray diffraction (both  $\theta$ - $2\theta$  and in-plane scans), AFM (Atomic Force Microscopy) and micro-Raman spectroscopy. The superconducting transition temperatures ( $T_c$ ) and critical current densities ( $J_c$ ) of YBCO films were resistively measured by a standard four-probe geometry on  $30\text{-}\mu\text{m}$ -long and  $5\text{-}\mu\text{m}$ -wide bridges.

## RESULTS AND DISCUSSION

Figure 1 shows the effect of the  $T_s$  during deposition on the  $T_c$ s of the YBCO thin films grown on MgO and  $\text{LaAlO}_3$  substrates. All of the films represented here were deposited

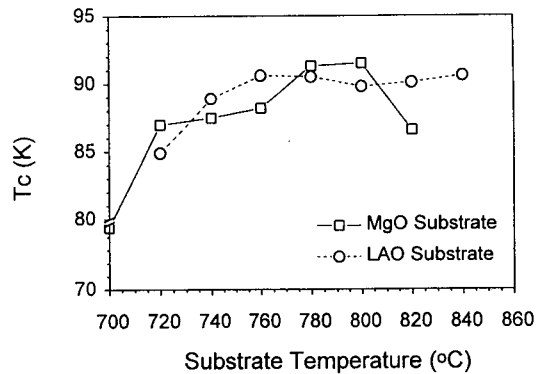


Fig. 1. The effect of the  $T_s$  during deposition on the  $T_c$ s of the YBCO thin films made by PLD

under  $500 \text{ mTorr}$  oxygen and the target-substrate distance was  $60 \text{ mm}$ . When MgO substrates were used,  $T_c$  was increased gradually with  $T_s$  of above  $720^\circ\text{C}$  showing maximum  $T_c$  of  $91.5 \text{ K}$  at  $800^\circ\text{C}$ . Beyond  $800^\circ\text{C}$ ,  $T_c$  decreased abruptly. In case of  $\text{LaAlO}_3$  substrates, similar increase of  $T_c$  was observed, but temperature range, showing  $T_c$ s of about  $90 \text{ K}$ , was much wider than MgO substrates. Over a range of  $T_s$  of  $100^\circ\text{C}$  ( $740$  to  $840^\circ\text{C}$ ) the films on  $\text{LaAlO}_3$  showed  $T_c$ s of above  $89 \text{ K}$ . Depositing films above  $840^\circ\text{C}$  was not available due to heating system limitations.

From the  $\theta$ - $2\theta$  scan data, it was observed that when  $T_s$  was above  $760^\circ\text{C}$ , all the films represented here were oriented with its c-axis perpendicular to the substrate surface regardless of the substrates used.

Figure 2 shows  $\text{PO}_2$  dependences of the  $T_c$ s on MgO and  $\text{LaAlO}_3$  substrates. As shown in Fig. 2, the  $\text{PO}_2$  during the deposition also had a substantial effect on the  $T_c$ s of films.  $T_s$  for the films, represented here, on MgO and  $\text{LaAlO}_3$  were at  $800$  and  $780^\circ\text{C}$ , respectively. The other parameters were the same as those

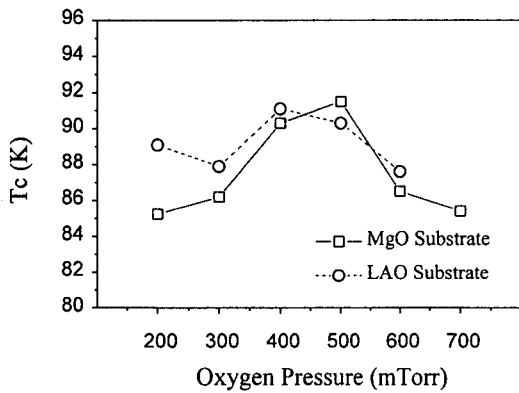


Fig. 2. The effect of the  $\text{PO}_2$  during deposition on the  $T_c$ s of the YBCO thin films made by PLD

of Fig. 1. The films grown under the  $\text{PO}_2$  of 400~500mTorr showed the best  $T_c$  of above 90K, irrespective of the substrates used. Both higher and lower  $\text{PO}_2$  resulted in lower  $T_c$ .

For the growth of YBCO thin films, it is highly recommended to maintain the deposition chamber under an oxidizing environment to compensate for some loss of a constituent oxygen. This is why the deposited films tend to be deficient in oxygen when the ablation is done in vacuum. Other roles of background oxygen is to provide help for the formation and stabilization of the desired crystal phase at the deposition temperature<sup>[2]</sup>. In-situ processing of YBCO films typically requires 100~300mTorr of the  $\text{PO}_2$  in the chamber. 400~500mTorr of the  $\text{PO}_2$ , as represented here, is somewhat higher. Higher in  $\text{PO}_2$  is thought to be due to relatively higher Ts when compared with others (~750°C). This correlation could be obtained from the stability phase diagram of YBCO as a function of temperature and oxygen pressure<sup>[3]</sup>.

While the surface morphology of the film was also influenced by  $\text{PO}_2$ , it was confirmed,

from AFM analysis, that the films grown above 500mTorr contain many small-sized particulates, so called boulders, ranging 0.1 to 0.5 $\mu\text{m}$ , on the surface. It is noted that the optimum  $\text{PO}_2$  value range is 400~500mTorr (Fig. 2).

Figure 3 shows the effect of target to substrate distance on the  $T_c$ s of YBCO films on MgO. All the processing parameters are the same as that of Fig. 2, except keeping  $\text{PO}_2$  of 400mTorr fixed. The target to substrate distance, showing  $T_c$  above 89K, was in the range of 50~60mm. As reported earlier by several groups, the velocities of the species in the laser plume are more than 10<sup>6</sup>cm/s near the target, and are decelerated as a function of ejected distance due to the collision with  $\text{O}_2$ <sup>[4]</sup>. These energetic species may cause some damage to the film if the target-substrate distance is too close. On the other hand, if the substrate is too far away, the deposition rate will be too low and the kinetic energy of the ablated species will be significantly lowered by the collision with surrounding gases as well as each other. This leads to lowering the surface diffusivity of deposited species and results in the poor film quality.

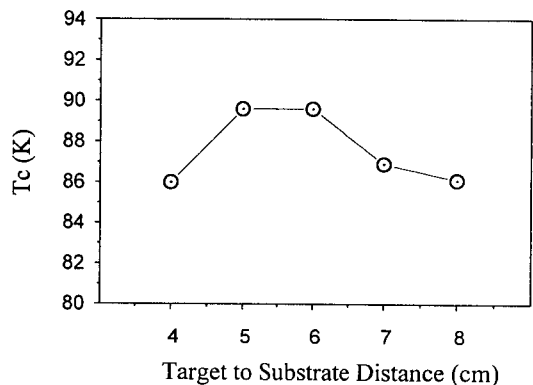


Fig. 3. The effect of target-substrate distance on the  $T_c$ s of the YBCO thin films made by PLD

Therefore, the substrate should be placed in the optimum distance range from the target to produce a good quality film. It is worthwhile to note that the optimum distance, under any processing parameters, is comparable to plume size. The optimum target-substrate distance of 50~60mm, as shown Fig. 3, is exactly the plume length.

Figure 4 shows the resistance as a function of temperature for YBCO films deposited on  $\text{LaAlO}_3$  under the optimized processing parameters. The detailed parameters are shown in inset in Fig. 4. For  $T > T_c$ , the resistance behavior is metallic and can be extrapolated to the origin, as shown in Fig. 4. The zero resistance temperature ( $T_{c,0}$ ) as high as 90~92K was obtained routinely. Similar results were obtained in the case of MgO substrates ( $T_{c,0} = \sim 91\text{K}$ ).

Figure 5 shows a  $\theta$ - $2\theta$  x-ray diffraction (XRD) pattern for a YBCO film deposited on MgO at 800°C at the  $\text{O}_2$  pressure of 500 mTorr. One can see only (00l) peaks, indicating that the film is c-axis textured.

Figure 6 shows the x-ray  $\varphi$ -scan taken at (104) reflection of a YBCO film on MgO to

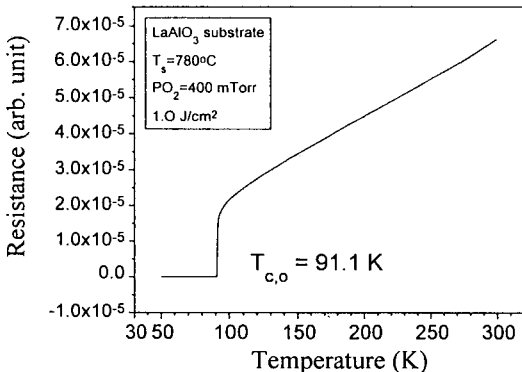


Fig. 4. Resistance vs temperature characteristic of a YBCO thin film grown by PLD on  $\text{LaAlO}_3$  substrate

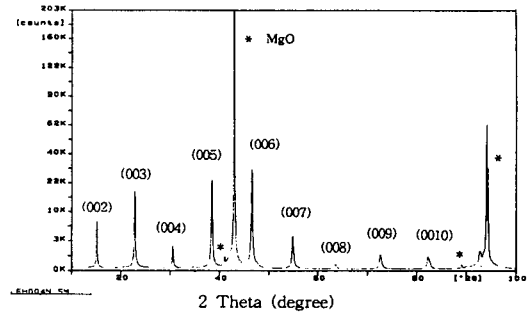


Fig. 5.  $\theta$ - $2\theta$  x-ray diffraction (XRD) pattern for a YBCO film deposited on MgO at 800°C under the  $\text{O}_2$  pressure of 500mTorr.

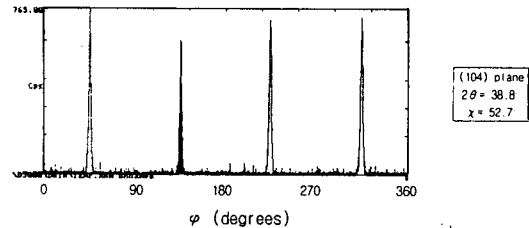


Fig. 6. x-ray  $\varphi$ -scan for a YBCO thin film grown by PLD on MgO substrate.

determine in-plane epitaxy. The  $0^\circ \sim 360^\circ$  scan shows only four peaks, meaning that the film made with the optimized processing parameters has an excellent in-plane epitaxy. This in-plane epitaxy was also confirmed by a polar plot analysis<sup>[5]</sup>.

Figure 7 shows a plot of  $J_c$  for a YBCO film on MgO as a function of temperature. At 77 K and zero field, the  $J_c$  was found to be  $6 \times 10^6$  A/cm<sup>2</sup>. This high value of  $J_c$  could be obtained partly due to the in-plane epitaxy in the YBCO layer on the substrates, as confirmed in Fig. 6.

Figure 8 shows a typical AFM picture of a YBCO film on MgO deposited at 500mTorr  $\text{PO}_2$  and  $T_s$  of 800°C. Several particulates with sizes of 0.2~1.0 $\mu\text{m}$  could be seen over the whole  $20 \times 20 \mu\text{m}^2$  scan area. While the ori-

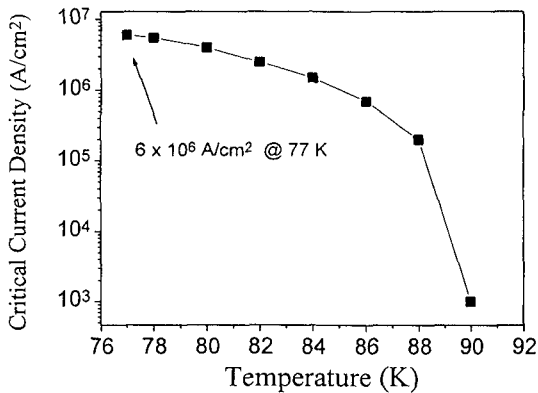


Fig. 7. Critical current density ( $J_c$ ) measurement for a YBCO thin film deposited on MgO substrate.

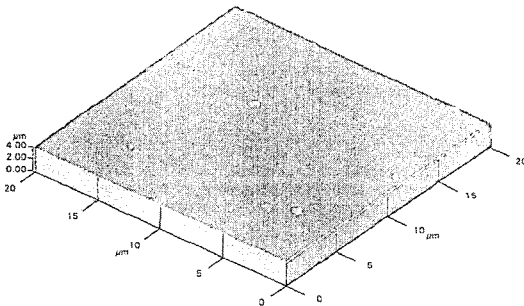


Fig. 8. AFM image of a YBCO thin film on MgO deposited at 500mTorr  $\text{PO}_2$  and  $T_s$  of 800 °C.

gin of the particulates is still under discussion, it is generally known that the irregular-shaped particles larger than about submicron are directly transferred from the target to the substrate. On the other hand, the sizes of the particles formed from the vapour phase tend to be in the nanometer range<sup>[6]</sup>. Considering the size and shape of the particulates formed on the surface, these are thought to be introduced directly from the target.

One of the applications of our high-quality thin films using PLD is the fabrication of the SQUID magnetometers<sup>[7]</sup>. We have made directly coupled dc SQUID magnetometers from single layers of YBCO on  $\text{SrTiO}_3$  bicr-

ystal substrates<sup>[8]</sup>. At 77K, the SQUIDs showed RSJ (Resistively Shunted Junction) like current-voltage ( $I$ - $V$ ) characteristics and voltage modulation in response to external fields.

Figure 9 shows the current-voltage ( $I$ - $V$ ) characteristic of a dc SQUID magnetometer at 77K. The junctions have critical current,  $2I_C$ , of  $200\mu\text{A}$  and normal state resistance  $R_N$  of  $0.5\Omega$  at 77K. Voltage-flux ( $V$ - $\Phi$ ) measurement showed the maximum peak to peak modulation voltage of  $17\mu\text{V}$  (Fig. 10).

Figure 11 shows a rms flux noise of the SQUID at 77K in zero field, obtained using flux-locked loop electronics. The noise characteristics were measured in magnetically shielded box. The white noise of the magnetometer is  $13\mu\Phi_0/\sqrt{\text{Hz}}$  at 400Hz, rising to  $60\mu\Phi_0/\sqrt{\text{Hz}}$  at 10Hz. These flux noise values correspond to noise energy  $7.24 \times 10^{-30}$  J-sec and  $1.54 \times 10^{-28}$  J-sec, respectively.

These obtained values are low enough to measure the biomagnetic signal from the heart of body.

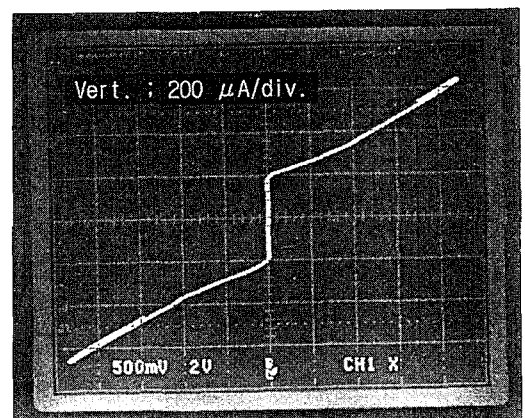


Fig. 9.  $I$ - $V$  characteristic of a YBCO thin film grown by PLD on  $\text{SrTiO}_3$  bicrystal substrate.

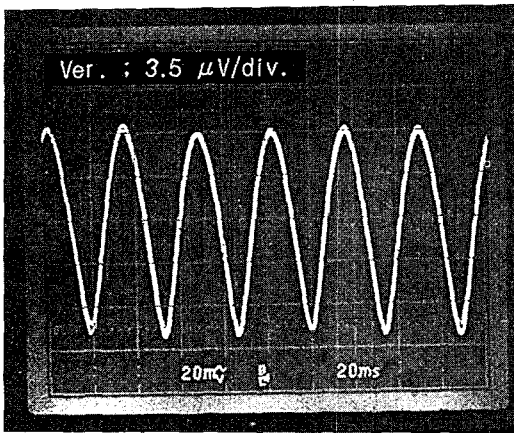


Fig. 10.  $V-\Phi$  characteristic of a YBCO thin film grown on  $\text{SrTiO}_3$  bicrystal substrate.

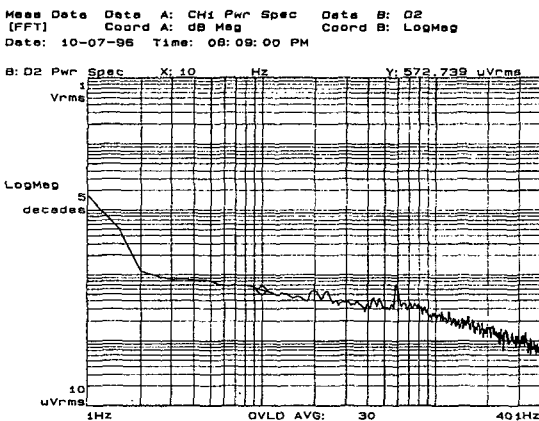


Fig. 11. Measured rms flux noise,  $\sqrt{S\Phi(f)}$ , of the dc SQUID made by the YBCO thin film grown on  $\text{SrTiO}_3$  bicrystal substrate.

## SUMMARY

Critical processing parameters of PLD for high-quality YBCO thin films on MgO and  $\text{LaAlO}_3$  substrates have been optimized through systematic studies. YBCO thin films on MgO and  $\text{LaAlO}_3$  showing the best characteristics could be made under 500 mTorr oxygen at  $800^\circ\text{C}$  and 400 mTorr at  $780^\circ\text{C}$ , respectively. Considering the plume lengths formed

under respective parameters, target-substrate distance of 50~60mm is found to be adequate for obtaining high-quality films. Using obtained parameters to date, the films having  $T_c$  of above 91K and  $J_c$  of above  $2 \times 10^6$  A/cm<sup>2</sup> at 77K have been obtained reproducibly. Extensive X-ray diffraction studies suggest that the YBCO film shows excellent in-plane epitaxy as well as its *c*-axis orientation perpendicular to the substrate surface. The directly coupled DC-SQUIDs with bicrystal grain boundary junction on  $\text{SrTiO}_3$  substrates using above thin films show 2IC and swing voltage of 200  $\mu\text{A}$  and 17  $\mu\text{V}$  at 77K, respectively. The flux noise of the magnetometer is 13  $\mu\Phi_0/\sqrt{\text{Hz}}$  at 400Hz and 60  $\mu\Phi_0/\sqrt{\text{Hz}}$  at 10Hz.

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