High Tc Superconductor, YBa₂Cu₃O₇₋₈;
(I) Its Preparation and Physicochemical Characterization

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고온 초전도체 YBa₂ Cu₃ O_{7- δ}; (1) 합성 및 물리화학적 특성 연구

최진호 • 변송호 • 홍승태 • 정덕영 • 최석용 • 김배환 • 김진태 노동윤 • 유한일 마 • 이동녕 마 • 승도영 마 • 박태석 마 서울대학교 화학과, 무기재료공학과 마, 금속공학과 마 삼성종합기술원 소재연구소 마 (1988년 1월 7일 접수)

요 약

고온초전도체인 YBa₂Cu₃O_x를 고온 고상반응에 의해 합성하였다. 산소분안 1기암하에서 940℃, 40시간 열처리후 450℃까지 서냉하여 약 10시간 유지한 후 상온으로 냉각시켜 X - 선 분석한 결과 사방결정계를 갖는 단일상 구조임을 밝혔으며 초전도전이온도(T_c)는 90 K 이었다. 요드적정법으로 산화확원 적정을 행하여 YBa₂Cu₃O_x의 평균 산소합량(x)을 6.85 ± 0.02 로 결정하였고 X - 선 광전자분광 분석법으로 YBa₂Cu₃O_x내 Cu 전자의 바닥상태가 d³(Cu³⁺) 및 d³(Cu³⁺)의 혼합상태이며 시료 표면에 국부적으로 Cu³⁺의 농도가 높음 왕 알 수 있었다.

ABSTRACT

Almost single phase of $Y_1Ba_2Cu_3O_x$, which exhibits the onset of superconducting transition at 90K, has been prepared via a conventional ceramic processing route followed by an anneal for a period of 10 hours at 450° C in an atmosphere of 100% O_2 . The mean value for the oxygen content has been determined as $x = 6.85 \pm 0.02$ by an iodometric titration technique. X-ray photoemission spectra are found to be consistent with a mixture of the $d^{6}(Cu^{3+})$ and $d^{9}(Cu^{2+})$ as the ground state and reveals that the local concentration of Cu^{3+} decreases with depth from the surface where x corresponds to 7.0. This is explained as being associated with the intercalation of oxygen during cooling from the annealing temperature.

INTRODUCTION

Since Bednorz and Müller(1) first reported a possible high Tc superconductivity in the La-Ba-Cu-O system, many efforts have been made around the globe to raise the superconducting transition temperature. Above all, discovery of superconductivity in the Y-Ba-Cu oxide system by Wu et al. and Chu et al.(2) opened an era of superconductors which may be usable above the liquid nitrogen temperature.

It has been established later that the superconductive phase is $Y_1Ba_2Cu_3O_{7-\delta}$ containing a large amount of oxygen vancancy which stabilizes the peculiar orthorhombic perovskite-derivative structure(3). According to the details of the structure of $YBa_2Cu_3O_{7-\delta}$ established by various techniques including X-ray diffraction(4), transmission electron microscopy, convergent beam electron diffraction(5) and neutron diffraction (6), Ba and Y ions are ordered in a fashion of Ba-Y-Ba sequence along the c-axis, and oxygen atoms at a z=1/2 level are completely deficient. Furthermore half of the oxygen atoms on a z=0 plane are lacking (7).

What is crucial for the realization of superconductivity in this system is likely to control the amount of oxygen vacancy as well as its ordering and/or the Cu3+ concentration, which is governed mostly by the thermodynamic condition of annealing, i.e., oxygen partial pressure and temperature(8). The oxygen content and Cu3+/Cu2+ ratio in the superconductor YBa2-Cu3O7-8 have been determined by iodometric titration and TG analysis(9). It has also been reported that the Cu-O chains along the b-axis of the orthorhombic YBa2Cu3O7-δ play a critical role in superconducting transition either through the existence of mixed Cu3+-Cu2+ ions and/or dimensional effects due to the presence of the one-dimensional Cu-O chains (10).

In this preliminary report we present results of the electrical resistivity measurement, iodometric titration, and X-ray photoelectron spectroscopic measurements made on an almost single phase $Y_1Ba_2Cu_3O_{7-\delta}$ synthesized via a conventional ceramic processing route.

EXPERIMENTAL

The YBa₂Cu₃O_{7- δ} powders prepared by a conventional ceramic processing technique; the intimate ground mixture of BaCO₃, Y₂O₃ and 3CuO was pressed into a pellet, calcined at 900°C in air for 16 hours, reground and pressed into a pellet, and sintered at 940°C in flowing O₂ for 40 hours. The sintered pellet was cooled to a temperature of 450°C at a rate of 100°C/hr, annealed in an atmosphere of 100% O₂ for 10 hours and then removed out of the furnace.

The structure of the prepared samples were examined by X-ray powder diffractometry with Ni filtered Cu-Kα radiation. The conventional four-probe technique was employed to measure the electrical resistivity over a range of temperature, 20 to 300K. Iodometric titration was carried out in an O2 free-nitrogen gas to avoid the oxidation of iodide ion. Sodium thiosulfate titrant was standardized against KIO3. Starch indicator was added immediately before the end point. X-ray photoelectron spectra were recorded as a function of sputtering time on a PHI 5100 Perkin-Elmer spectrometer. Unmonochromatized $Mg-K_{\alpha}$ radiation of 1253.6 eV was used and the base pressure in the spectrometer was no greater than 2.0x10⁻⁹ Torr. Argon ion sputtering was made under the condition of 1kV, 25mA, and 2x10⁻⁹ Torr. The binding energies reported here were corrected from the known reference (BE of $C_{ls} = 284.6 \text{ eV}$).

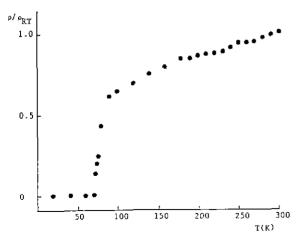


Fig. 1. Temperature dependence of the normalized electrical resistivity for YBa₂-Cu₃O_{6.85}

RESULTS AND DISCUSSIONS

The X-ray powder diffraction patterns for the YBa₂Cu₃O_{7- δ} could be successfully indexed on the basis of an orthorhombic unit cell (Tab. 1) as reported by other investigators (3, 8, 11, and 12). The temperature dependence of the electrical resistivity for the YBa₂Cu₃O_{7- δ} is shown in Fig. 1. Superconducting transition with an onset temperature of 90K has been observed under ambient pressure.

We have determined the $\mathrm{Cu}^{3+}/\mathrm{Cu}^{2+}$ ratio of the superconducting phase and therefrom the oxygen nonstoichiometry(x) of a lattice molecule $\mathrm{YBa_2}(\mathrm{Cu}_{3-2}^{2+}\mathrm{Cu}_{2x}^{3+})\mathrm{O}_{6.5+x}$ quantitatively by iodometric titration;

Cu³⁺·aq. + 3Γ·aq. → CuI(s) + I₂·aq.
Cu²⁺·aq. + 2Γ·aq. → CuI(s) +
$$\frac{1}{2}$$
·aq.
 $\frac{3}{2}$ I₂ + 3S₂O₃²⁻ → 3Γ + $\frac{3}{2}$ S₄O₆²⁻

The x value has turned out to be 0.35 ± 0.02 , which yields the formula of $YBa_2Cu_{2.3}^{2+}Cu_{0.7}^{3+}$. O_{6.85}. The present result is in excellent agree-

Table 1. Powder X-ray data for YBa₂Cu₃O_{6.85}.

			-	-	5 0.05
h	k	1	d(obs.)/A	d(cal.)/A	I/I _O
0	0	1	11.72	11.69	4
0	0	2	5.85	5.84	4
0	0	3		3.896	
0	1	0	3.879	3.889	12
1	0	0	3.827	3.824	4
0	1	2	3.236	3.237	5
1	0	2	3.203	3.200	5
0	1	3	2.753	2.752	58
1	1	0			
1	0	3	2.729	2.729	100
1	1	1	2.655	2.655	1
1	1	2	2.471	2.471	3
0	1	4		2.336	
0	0	5	2.339	2.337	14
1	1	3	2.235	2.234	19
0	1	5	2.005	2.003	1
1	1	4	1.9943	1.9934	1
0	0	6	1.9497	1.9473	13
0	2	0	1.9434	1.9443	18
2	0	0	1.9117	1.9121	12
1	1	5	1.7758	1.7746	4
0	1	6	1.7416		
0	2	3	1.7387	1.7396	2
1	2	1		1.7144	
2	1	0	1.7146	1.7159	1
2	0	3		1.7165	
1	2	2	1.6612	1.6616	1
1	2	3		1.5835	
1	1	6	1.5845	1.5850	25
2	1	3	1.5705	1.5703	14
0	1	7	1.5333	1.5341	2
0	2	5	1.4947	1.4948	3
2	1	4	1.4000	1.4796	_
2	0	5	1.4805	1.4800	3
1	1	7	1.4252	1.4239	2
0	2	6	1.3764	1.3761	5
2	0	6	1 2/24	1.3645	4.0
		0	1.3634	1.3633	10

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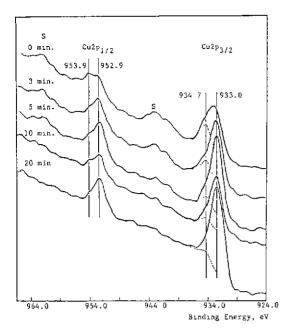


Fig. 2. XPE spectra in the region of the binding energy for Cu2p electrons in the YBa₂Cu₃O_{6.85} against the Ar ion sputtering time.

ment with the equilibrium oxygen content predicted by Jorgensen et al.(10) at 450°C in 1 atm Po₂.

We performed also X-ray photoelectron spectroscopic measurements on YB2Cu3O6.85 to confirm the coexistence of Cu3+ and Cu2+ Figure 2 shows the X-ray photoelectron spectra against the Ar ion sputtering time in the region of the binding energy for the Cu2p electrons. The intensity of the minor shoulder peak $(2p_{3/2} =$ 934.7 eV) decreases gradually and the major one $(2p_{3/2} = 933.0 \text{ eV})$ increases sharply with the increased sputtering time. The peak separation was based on the reference spectra for the La₂Cu²⁺O₄ and the La₄LiCu³⁺O₈ (prepared under the high oxygen pressure(13)). The binding energies for the Cu2p3/2 electrons in the La₂CuO₄ having only Cu²⁺ ions and the La₄-LiCuO₈ in which only the Cu³⁺ ions exist were 933.2 and 934.6 eV, respectively. Compared

with these results, the present spectra for YBa2-Cu₃O_{6.85} are consistent with a mixture of the $d^{8}(Cu^{3+})$ 2p_{3/2}:934.7 eV and the $d^{9}(Cu^{2+})$ 2p3/2'933.0 eV configurations as the ground state. The band in the region of the binding energy for the Cu2p3/2 electrons was separated by using a reiterative curve fitting computer program provided by Perkin-Elmer. The content of Cu3+ ion calculated from the area of the separated bands before the Ar ion sputtering is approximately 1.0 which is almost identical with a chemical formula of YBa2Cu3O7 and this is slightly higher than that (0.7) from the redox titration This indicates that Cu3+ ions present in the sample are more or less concentrated on the surface of the YBa₂Cu₃O_{7-δ} phase. It may be ascribed to the preferential oxidation at the surface of the specimen during cooling from the annealing temperature of 450°C. The possibility of being associated with the local hydration due to the atmospheric humidity is ruled out because hydration proceeds consuming Cu3+ ions of the superconducting phase (14),

According to the crystallographic structure determined (15-17), the copper coordination number should change from 6 in an ideal stoichiometric perovskite-type compound to 4 and 5 in YBa₂Cu₃O_{7-δ} whose local symmetries are D_{4h} and C_{4v}, respectively. It has also been proposed that most of the Cu³⁺ ions present in the sample are located in the one-dimensional chains. That is, as the d9(Cu2+) electronic configuration is more stable in C4v symmetry as shown in Figure 3, the Cu³⁺(d⁸) ions may be enriched in the site of D4h symmetry which is located in one-dimensional chains. According to the simple calculation of crystal field stabilization energy(CFSE) in the reduced symmetry of $C_{4y}(-18.28 \text{ Dq})$ and $D_{4h}(-24.56 \text{ Dq})$ for Cu^{3+} (3d8), it is quite clear the all Cu3+ ions prefer the square-planar site (D_{4h}) than square pyramidal

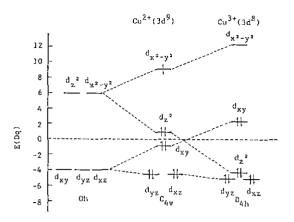


Fig. 3. Relationship of crystal-field energies for d-orbitals in octahedral (O_h) -, square-pyramidal (C_{4v}) - and square planar (D_{4h}) -symmetry.

one (C_{4v}) due to high D_{4h} site preferencial energy. And presumably the Cu2+ ion with a CFSE of -12.28 Dq in D4h site will fail in the competition with the Cu3+ ion in the same site and, therefore, be forced to occupy the Can site in YBa2Cu3O7.8 system. From this consideration of CFSE, it should be apparant why square planar complexes are much more common for d⁸ electron configurations than for d⁹ and actually, most square planar complexes known involve Ni2+, Pd2+. Pt2+, and Au3+, all of which are d⁸ configuration(18). A study on the quantitative depth profile, which would give the exact information for the distribution of Cu3+ ion present in the superconductor YBa, Cu, O,, has already been under way.

To date, there is no X-ray photoelectron spectroscopic evidence for the coexistence of Cu^{3+} and Cu^{2+} ions. Fujimori et al. (19) suggested that the Cu^{3+} is screened by a ligand-to-metal atom charge transfer and becomes nearly d^9 -like in its ground state. This may appear inconsistent with our observation. But taking into consideration that the Y-Ba-Cu-O system is very sensitive to the ambient $H_2O(14)$, the con-

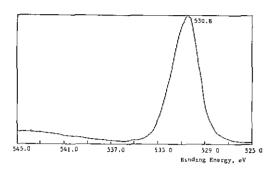


Fig. 4. Oxygen O_{ls} singlet spectrum for YBa₂-Cu₃O_{6.85}.

dition in which the sample was handled might be a considerable factor changing the XPE spectra. Besides above observation, it is very interesting that the satellite peak at about 943 eV diminishes gradually with increasing the Ar ion sputtering time as shown in Figure 2 Van den Laan et al.(20) and Larsson(21) pointed out that, in the Cu2p region of the photoelectron spectra of copper dihalides, the main peak corresponds to transitions to a 2p3d10 L-like final states (2p: core hole in the 2p state, L:hole in the ligand state) and the satellites to a 2p3d9-like multiplet of final states. The satellite structure can appear from a lignad-to-metal atom charge transfer(22) as well as the coexistence of the d8 and d9 configuration in the ground state(23). Thus we can propose that, since the temperature of the sample is increased during the Ar ion sputtering at extremely low oxygen pressure, the intensity of the satellite peak is decreased remarkably because of decreased charge transfer from the ligand which results from the increased oxygen vacancy, the Cu3+ and Cu2+ ions are reduced to Cu⁺ ions (3d¹⁰) which show no measurable Cu2p3d9 satellite peaks at all. However, the fact that there is little shift of the main peak to lower energy region does not confirm the possibility of the reduction of the Cu³⁺ and Cu²⁺ to the Cut. As shown in Figure 4, the oxygen

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 O_{1s} line position of 530.8 eV with FWHM = 1.3 eV in YBa₂Cu₃O_{6.85} phase is more or less consistent with the binding energies observed in the CuO (BE of O_{1s} ; 529.6 ± 0.15 eV with FWHM = 1.3 eV) and the Cu(OH)₂ (530.9 ± 0.2 eV with FWHM = 1.8 eV) (24).

In summary, the prepared sample can be formulated as $YBa_2Cu_{0.69}^{3+}Cu_{2.31}^{2+}O_{6.85}$ as a spatial average from Cu^{3+}/Cu^{2+} ratio determined by the iodometric titrations. XPE spectra has revealed that the content of oxygen(x) or Cu^{3+} ion decreases with depth over a narrow extent beneath the surface exposed to atmosphere. It is interpreted as being due to the preferential oxidation at the surface during cooling. The strong Jahn-Teller distortion due to Cu^{2+} ions stabilized in C_{4v} -sites and the lattice contraction because of the existence of Cu^{3+} in D_{4h} -ones might be responsible for the high temperature superconducting transition temperature in Y-Ba-Cu-O system.

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REFERENCES

- J. G. Bednorz and K. A. Müller, Z. Phys., B64, 189 (1986).
- M. K. Wu, J. R. Ashburn, C. J. Torng,
 P. H. Hor, R. L. Meng, L. Gao, Z. J. Huang,
 Y. Q. Wang and C. W. Chu, Phys. Rev. Lett., 58, 908 (1987).
- R. J. Cava, B. Batlogg, R. B. van Dover, D. W. Murphy, S. Sunshine, T. Siegrist, J. P. Remeika, E. A. Rietman, S. Zahurak and G. P. Espinosa, *Phys. Rev. Lett.*, 58, 1676 (1987).

- E. Takayama-Muromachi, Y. Uchida, Y. Matsui and K. Kato, Jpn. J. Appl. Phys.,
 26, L619 (1987)
- Y. Matsui, E. Takayama-Muromachi, A. Ono, S. Horiuchi and K. Kato, *Jpn. J. Appl. Phys.*, 26, L777 (1987).
- F. Izumi, H. Asano, T. Ishigaki, E. Taka-yama-Muromachi, Y. Uchida. N. Watanabe and T. Nishikawa, *Jpn. J. Appl. Phys.*, 26, L649 (1987).
- F. Izumi et al., Jpn. J. Appl. Phys., 26, L649 (1987).
- E. Takayama-Muromachi, R. Uchida, Y. Matsui and K. Kato, Jpn. J. Appl. Phys., 26, L619 (1987).
- P. K. Gallagher, H. M. O'Bryan, S. A. Sunshine and D. W. Murphy, *Mat. Res. Bull.*, 22, 995 (1987).
- J. D. Jorgensen, M. A. Beno, D. G. Hinks,
 K. J. Volin, R. L. Hitterman, J. D. Grace
 and I. K. Schuller, *Phys. Rev.*, in press (1987).
- E. Takayama-Muromachi, Y. Uchida, Y. Matsui and K. Kata, Jpn. J. Appl. Phys., 26, L476 (1987).
- P. M. Grant, R. B. Beyers, G. Lim, S. S. P. Parkin, M. L. Ramırez, V. Y. Lee, A. Nazzal, J. E. Vasquez and R. J. Savoy, *Phys. Rev.*, **B35**, 7242 (1987).
- J. H. Choy, S. H. Byeon, D. Y. Jung, S. T. Hong, and G. Demazeau, in preparation (1988).
- M. F. Yan et al., Appl Phys. Lett., Preprint (1987).
- J. J. Capponi et al., Europhys. Lett, 3, 1301 (1987).
- M. A. Beno et al , Appl. Phys. Lett., Preprint (1987).
- 17. W.I.F. Davit et al., Nature, Preprint (1987).
- J. C. Bailer, et al., "Comprehensive Inorganic Chemistry", Vol. 3, Pergamon Press Ltd, Oxford (1976).

- 19. A. Fujimori, E. Takayama-Muromachi, Y. Uchida and B. Okai, Preprint (1987).
- G. van den Laan, C Westra, C. Haas and
 G. A. Sawatzky, *Phys. Rev.*, **B23**, 4369 (1981).
- S. Larsson and M. Braga, Chem. Phys. Lett., 48, 596 (1977).
- 22. K. S. Kim, J. Electron Spectrosc. *Relat. Phenom.*, 3, 217 (1974).
- A. Kotani and Y. Toyozawa, J. Phys. Soc. Jpn., 35, 1073; 1082 (1973).
- N. S. McIntyre and M. G. Cook, *Anal. Chem.* 47, 13, 2208 (1975).

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