The Characteristics of Degradation in Epoxy Composites for Molding Materials of PT

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ABSTRACT

The electrical degradation phenomena of epoxy composites to be used as a molding material for transformers were studied. The electrets were first manufactured by applying high voltages to five kinds of specimens given a mixing rate, and then TSC(Thermally Stimulated Current) values at the temperature range of -160~200[°C] were measured from a series of experiments. The behaviour of carrier and its origin in epoxy composites were examined, respectively. And various effects of electrical degradation on epoxy composites were also discussed in this study.

Keyword: degradation phenomena, Epoxy Composites, TSC(Thermally Stimulated Current)

I. INTRODUCTION

As the scale of a power delivery system is increased, the insulating design techniques such as a materialization of high electric field and a security of reliability in insulating composition are required accordingly.

The main purpose of this research is to study the TSC values in electrical stressed epoxy composites and interrelationship between electrical stress and material degradation. The TSC values associated with the determination of the currents which were released from poled specimens were measured as a function of temperature. Poling was accomplished by placing the specimen under a high DC stress and followed by cooling with liquid nitrogen to freeze the molecular motion of the polymer, resulting in immobilizing the charges.

When the specimens are warmed slowly, the trapped charges are released. By measuring the amount of charge released and their release characteristic temperature, inferences on the space charges which form in the polymer can be made.

It has been noticed that peak temperatures for a given heating rate depend on a variety of factors such as age of the specimen, thermal history, and charge density. To investigate the effect of the electrical degradation on properties of epoxy composites, the TSC spectra were measured as a function of temperature and the various aspects of electrical degradation effects were analyzed.

II. EXPERIMENTAL

1. Specimens

The specimens used in this study were designed to have a constant mixing rate in the epoxy resin of Bisphenol-A type and the MeTHPA. Bisphenol-A type is in liquid phase at room temperature, and the MeTHPA is a hardener of acid hydro-anhydride system.

Also, the DY-040 and the SiO2 with amount of 5wt% are then added to improve both the impact strength and the machinery strength, and the silane coupling agents, which belong to an amino silane system have been used for surface treatment of fillers.

2. Design of Mixing Ratio

Since the electrical and physical properties of epoxy composites depend greatly on the mixing ratio among resin, hardener and filler, as well as the hardening conditions^{[1]~[2]}, these aspects need to be considered in manufacturing specimens. The design parameters and their condition used for this study are summarized in Table 1.

Table 1. Mixing Ratio [wt%]

Samples	Ероху	Hardener	DY -040	Filler	Curing Condition
H 70FN	100	70	10	0	▶1st Curing; 100['c]×4[hr] ▶2nd Curing; 140['c]×10[hr] ▶Deg. Curing; 140['c]×240[hr]
H100FN	100	100	10	0	
H130FN	100	130	10	0	
H100F60	100	100	10	60	
SH100F60	100	100	10	60	

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3. Experimental Method

After the samples were degraded by appling high voltages, 22.9[κ V/cm], during 5[hr], the electrets^{[3]-[4]} were formed by appling a electric field, $10[\kappa$ V/cm], to epoxy composites and then TSC spectra were measured at the temperature range of -160~200[°C] with a step of 5[°C/min]. A Schematic drawing of the experimental apparatus to be used for TSC values is given in Fig. 1.

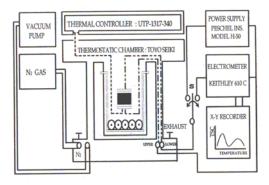


Fig. 1. Experimental Apparatus

III. RESULTS AND DISCUSSION

1. TSC Spectra due to Forming Electric Field

Fig. 2 shows TSC spectra formed at 5, 10, and $30[\kappa V/cm]$, which were observed from electrets for the 1st cured(100 [°c]×4[hr]) H100FN.

Similarly, Fig. 3 includes TSC spectra formed at 5, 10, and 30[kV/cm], using the 2nd cured(140[°C]×1[hr]) H100FN.

From the experimental study, the five peaks of δ , γ , β , α 1 and α 2 were obtained at the temperature of -100[°C], -40[°C], 20[°C], 100[°C] and 130[°C], respectively. The first three peaks which are observed at the region of low temperature below the glass transition temperature(Tg) are relaxation peaks due to the action of side chain or terminal radical. [5]

Also, both the $\alpha 1$ peak near $100[\columnwfe]$ and the $\alpha 2$ peak above Tg are peaks caused by ionic space charge. The causes of these peaks are as follows:

- (1) Since the δ peaks have been suggested by either a methyl radical or a non-acted epoxy radical, the overall size of peaks is not so big. From the molecular structure, it is observed that the methyl radical is independently operated in proportion to the level of hardener added and it also affects curing reaction.
- (2) The γ peak is produced by the mixture of hydroxyether

- radical, methoxy radical, aliphatic ether radical, and aromatic ether radical which is related to hardener. Also, as the amount of hardener is increased, the size of peak is getting bigger likewise the δ peak.
- (3) The β peak appears to be resulted from aromatic ether radical which is thermally oxided through the procedure of maximum reaction.
- (4) The $\alpha 1$ peak is believed to be the effects from the virtue of molecule which is related to glass transition temperature.
- (5) The α2 peak which is observed at the highest temperature is generated by the ionic space charge which is accumulated by impurities or inside defects.

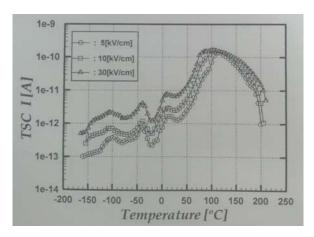


Fig. 2. The Dependence on Forming Electrical Strength in the 1st Cured H100FN Samples

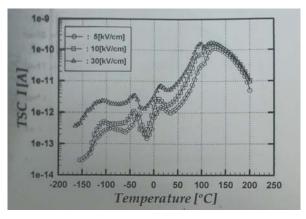


Fig. 3. The Dependence on Forming Electrical Strength in the 2nd Cured H100FN Samples

2. TSC Spectra due to Electrical Stress

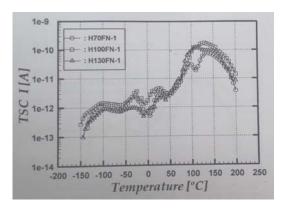
TSC spectra, which are divided according to curing time and observed from electrets degraded by appling electrical stress, are shown in Fig. 4 and 5.

It is noted that in the case of the longest curing time, the

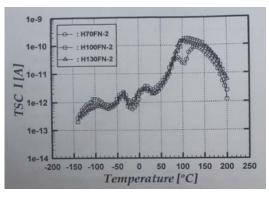
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size of the α peak which is shown above $100[\ensuremath{\,^{\circ}}$

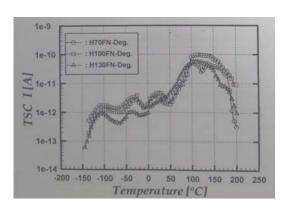
On the other hand, molecular motion become dull because of excessive hardener, and the peak size of H130FN is decreased above Tg. Also, Im of SH100F60 is reduced above Tg but that of H100F60 is not reduced because of improving deformation of interfacial state.



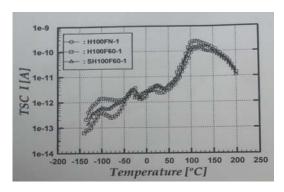
(a) 1st Curing $\langle 100[c] \times 4[hr] \rangle$



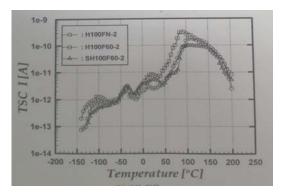
(b) 2nd Curing ⟨140[°c]×10[hr]⟩



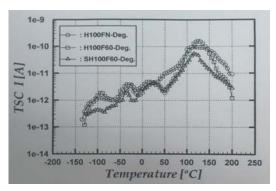
(c) Degradation ⟨140[℃]×240[hr]⟩ Fig. 4. TSC Spectra according to Hardener



(a) 1st Curing ⟨100[°c]×4[hr]⟩



(b) 2nd Curing $\langle 140[^{\circ}\text{c}] \times 10[\text{hr}] \rangle$



(c) Degradation $\langle 140[\color{c}] \times 240[\color{hr}] \rangle$ Fig. 5. TSC Spectra according to Filler

IV. CONCLUSION

In order to understand the degradation characteristics of epoxy composites due to electrical stress, TSC spectra were measured at the temperature range of -160 $^{\sim}$ 200[°]. From a series of experiments, five peaks including δ , γ , β , α 1 and α 2 were obtained at the temperature of -100°C, -40°C, 20°C, 10 0°C, and 130°C, respectively. Results of this study indicates that (1) the δ peak is resulted from the methyl radical; the γ peak is produced due to the mixture by hydroxyether radical, methoxy radical, aliphatic ether radical and aromatic ether radical; the β peak is obtained from aromatic ether

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radical; the $\alpha 1$ peak is possibly due to molecule which is related to glass transition temperature; and the $\alpha 2$ peak seems to turn up by the ionic space charge, and (2) in the case of the longest curing time, the size of the α peak which is shown above $100[\cap{C}]$ is smaller than those of 1st cured specimen and 2nd cured specimens. Particularly, the amplitude among samples is not significant below Tg and this result implies that the radicals shown at low temperature are affected by electrical stress. And, molecular motion become dull because of excessive hardener, and the peak size of H130FN is decreased above Tg. Also, Im of SH100F60 is reduced above Tg but that of H100F60 is not reduced because of improving deformation of interfacial state.

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