

Studies on Preparation of Ti₃SiC₂ Particulate Reinforced Cu Matrix Composite by Warm Compaction and its Tribological Behavior

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Abstract

Warm compaction powder metallurgy was used to produce a Ti_3SiC_2 particulate reinforced Cu matrix composite. Fabrication parameters and warm compaction behaviors of Cu powder were studied. Based on the optimized fabrication parameters a Cu-based electrical contact material was prepared. Results showed that in expend of some electrical conductivity, addition of Ti_3SiC_2 particulate increased the hardness, wear resistivity and anti-friction ability of the sintered Cu-base material.

Keywords: powder metallurgy, warm compaction, Cu-based composite, electrical conductivity

1. Introduction

Cu is an excellent electrical conductor with relatively low strength. Addition of alloying elements will severely decrease its conductivity. Ti₃SiC₂ is a good anti-friction material with good electrical and thermal conductivity [1]. Electrical conductivity and mechanical property of powder metallurgy (PM) materials are directly related to their density. Conventional powder metallurgy (PM) process can produce Cu compact with a density of 8.3 g/cm³. Mechanical properties and electrical conductivity of this sintered material are substantially less than that of their full density counterparts. Warm compaction is an effective way to produce high density PM parts [2-3]. In this study, warm compaction was used to produce a Ti₃SiC₂ particulate reinforced copper matrix composite. Fabrication parameters and warm compaction behaviors of Cu powder were studied. Based on the optimized fabrication parameters Cu-based composite for electrical contacts was prepared.

2. Experimental and results

Cu powder with 99.7 mass % purity and particle size of $\leq 74 \,\mu\text{m}$, was mixed with an admixed lubricant for 120 min. Lubricant concentrations of 0, 0.1, 0.15, 0.2, 0.3 and 0.4 mass % was used. The mixed powder was heated to 145°C and then compacted into standard tensile specimens in a steel mold, which maintained at 145°C also. Compaction pressure of 100 to 700 MPa was used. The green compacts were then pre-sintering at 800°C for 0.5 h and sintered at 1000°C for 1 h under cracked ammonia atmosphere.

For the composite samples, mixed powder was prepared by mixing Cu powder with 5 % Ti_3SiC_2 ($\leq 50 \mu m$). Then 0.2 % of lubricant was added and mixed for 2 h. The

powder was warm compacted at 700 MPa and sintered under the condition mentioned above. Density, hardness, ultimate tensile strength, elongation and electrical resistivity of the sintered samples were measured. The Cu powder was melted and cast to form the cast pure Cu samples for comparison.

Friction coefficient and wear volume of the sintered samples were measured using a block on ring tribotester. The counterpart was a GCr15 steel ring with a hardness of HRC 60. No lubricant was used in the test; a rotating speed of 200 r/min and a load of 98 N were used. The total testing time was 7.5 min.

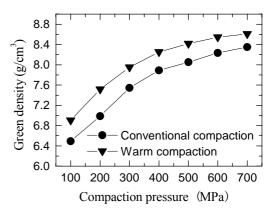


Fig. 1. Density Vs. compaction pressure.

Fig. 1 is a comparison between the green densities of warm compacted and conventionally compacted Cu powder mix with 0.2% lubricant, using different compaction pressure. Fig. 1 shows that green densities of the compacts that obtained from both warm and conventional compaction

increase with increasing pressure, then leveled off when compaction pressure reached 700 MPa.

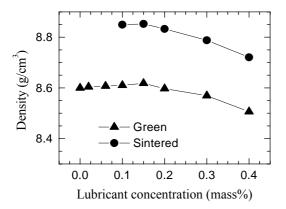


Fig. 2 Density versus lubricant concentration.

Fig. 2 shows the relation between compact density (both green and sintered) and lubricant concentration for samples compacted at 700 MPa and 145 $^{\circ}$ C. From Fig. 2 it can be seen that green density increase slightly as lubricant concentration increase, then drop after lubricant concentration reached at 0.15 %. From Fig. 2 it can be seen that highest green density can be obtained by using a lubricant content of 0.15 %. However, if the green density is too high, there is no way out for the trapped gases such as vaporized or dissociated lubricant and leads to blistering of the sintered samples [4]. Results show that blisters can be found in samples that contain lubricant lubricant concentration in order to avoid blistering.

Based on above results, warm compacted copper material with a green density of 8.57 g/cm^3 and a sintered density of 8.83 g/cm^3 was obtained. Its ultimate tensile strength was 264 MPa compared to 182 MPa of the conventionally compacted and sintered material. Using the same processing parameters, Cu-based composite reinforced by 5 % Ti₃SiC₂ particulate was prepared. Mechanical properties, electrical

resistivity and tribological behaviors of the warm compacted pure Cu and the Ti_3SiC_2 reinforced Cu-based composites were listed in Table 1.

The warm compacted and sintered pure Cu fabricated in this study has a resistivity of 1.92 x $10^{-8} \Omega$ m. While the sintered composite, with 5 % Ti₃SiC₂, has a resistivity of 11.04 x $10^{-8} \Omega$ m.

As shown in Table 1, all pure Cu samples could not endure the complete tribo-test; they failed within couples of minutes in the test. However, the 5% Ti_3SiC_2 reinforced composite show remarkable tribological behaviors. In expend of some electrical conductivity, the addition of Ti_3SiC_2 particulate increased the hardness, wear resistivity and anti-friction ability of Cu-base PM materials.

3. Summary

In this study, warm compaction was employed to develop a Ti₃SiC₂ particulate reinforced copper matrix composite with high density, high electrical conductivity and high strength. Sintered composite with 5 % Ti₃SiC₂, has an ultimate tensile strength of 201 MPa, an elongation of 8% and a hardness of HB 68 and an electrical resistivity of 11.04 x 10⁻⁸ Ω m. The increase in hardness from HB 51 (sintered pure Cu) to HB 68 will greatly enhance the tribological behaviors of the Cu-based PM material without losing much of the electrical conductivity.

4. References

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	Cast Cu	Warm compacted Cu	5% Ti ₃ SiC ₂
Density [g/cm ³]	8.96	8.83	8.014
Resistivity $[x10^{-8}\Omega.m)$	1.73	1.92	11.04
Tensile strength [MPa]	220	264	201
Elongation [%]		29	8
Hardness [HB]	51	51	68
Friction coefficient	Seizure ^a	Seizure ^b	0.123
Wear volume [mm ³]	Seizure ^a	Seizure ^b	1.291
Normalized wear volume [x10 ⁻⁸ mm ³ /N.mm]	Seizure ^a	Seizure ^b	6.288

Table 1. Some properties of the prepared materials compared with Cu

Note: a=Seizure after 3 min; b=Seizure after 7 min