세라믹 선재의 전기 구조적 특성

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Abstract

Fabrication of c-axis oriented ($Hg_{0.75}Re_{0.25}$) $Ba_2Ca_2Cu_3O_y$ thick fabricated has been attempted using Ni substrates with the buffer layer of Cr or NiO. Coexistence of ($Hg_{0.75}Re_{0.25}$) $Ba_2Ca_3Cu_4O_y$ pellets wad found to stabilize ($Hg_{0.75}Re_{0.25}$) $Ba_2Ca_2Cu_3O_y$ phase of the tape. The c-axis oriented tapes were reproducibly obtained on the NiO/Ni substerate and they recorded high B_{irr} at $77K.(Hg_{0.75}Re_{0.25})Ba_2Ca_2Cu_3O_y$ 1223.

Keywords: (Hg_{0.75}Re_{0.25})Ba₂Ca₂Cu₃O_y, NiO/Ni, buffer layer

1. Introduction

We have reported that (Hg_{0.75}Re_{0.25})Ba₂Ca₂Cu₃O_y superconductor was chemically stable with high Tc above 130K[1]. In ddition, its intragrain Jc under magnetic fields was dramatically enhanced with increasing Re doping level up to x=0.25, suggesting that (Hg_{0.75}Re_{0.25})Ba₂Ca₂Cu₃O_y can be considered as a candidate material for practical high field applications at high temperatures. We have attempted to increase Re doping level in c-axis oriented (Hg_{0.75}Re_{0.25})Ba₂Ca₂Cu₃O_y tapes for further improvement of critical current properties.

2. Experimental

The calcined powder of nominal composition Re_{0.25}Ba₂Ca₂Cu₃O_y percursor, which was prepared from mixed powders of ReO₃, BaCO₃, CaCO₃, and CuO, was mixed with HgO, then sintered at 90

0°C for 20h in a sealed quartz ampoule. The obtained (Hg_{0.75}Re_{0.25})Ba₂Ca₂Cu₃O_v 1223 pellets were ground and used as starting powder of the thick film. The Hg-containing pellets as Hg vapor source were prepared with nominal compositions of (Hg_{0.75}Re_{0.25})Ba₂Ca₂Cu₄O_y in the similar way. In the case of the Cr/Ni substerate 1223 thick film was formed on the substrate by painting method using a slurry containing calcined powder of 1223. On the other hand, the thick film on the NiO/Ni substrate was formed by the doctor-blade technique [2], using the mixture of the precursor and 1223. Here the NiO layer was prepared by oxidizing the surface of the Ni foil at 900°C in air. The tapes were put into Al2O3 tube, then sintered at various temperatures in a scaled quartz ampoule together with Hg-containing pellets. The microstructure of the tape surface was observed by a scanning electron microscope. The electrical transport properties of the tapes were measured under magnetic fields up to 9T by the conventional four probe method Tc and B_{irr} were determined by the resistivity of $1\mu\Omega cm$.

3. Results and Discussions

 $(Hg_{0.75}Re_{0.25})Ba_2Ca_2Cu_3O_y$ 1223 tapes were obtained with the Cr/Ni and NiO/Ni when substrates, pellets of Hg_{0.75}Re_{0.25}Ba₂Ca₃Cu₄O_y and Hg_{0.6}Re_{0.4}Ba₂CaCu₂O_y were used as Hg vapor source. addition. platelet crystals of (Hg_{0.75}Re_{0.25})Ba₂Ca₂Cu₃O_v 1223 were grown on the substrate by sintering at 830-890°C. In the case of the Cr/Ni substrate, strongly c-axis oriented (Hg_{0.75}Re_{0.25})1223 tape was fabricated by sintering at 890°C with an intermediate uniaxial pressing. The oriented tape exhibited Tc=127K under zero magnetic field and Birr=5T at 77K. This Birr at 77K is much higher than that of pure Hg1223 single crystals[3], suggesting that Re doping is effective for enhancement of Birr under B//c.

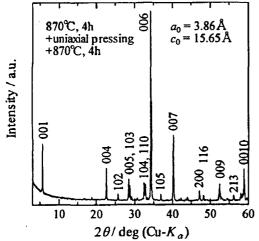
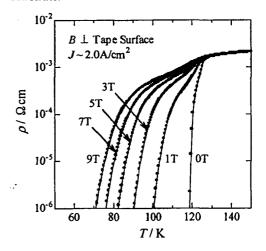


Fig. 1. The X-ray diffraction pattern for (Hg_{0.75}Re_{0.25})1223/NiO/Ni tape.

However, it has been difficult to reproduce (Hg_{0.75}Re_{0.25})1223/Cr/NiO/Ni tapes possibly due to the incompleteness of Cr coating on the Ni substrate. On the other hand. (Hg_{0.75}Re_{0.25})1223/NiO/Ni could tapes be the surface synthesized reproducibly and morphology was found to be improved. Figure 1 shows a typical X-ray diffraction pattern of the tape. The best tape exhiibited Tc=119K and Tirr=82K under the magnetic field of 5T applied perpendicular to the thapes surface as shown in Fig.2. Though Tc of the tape was lower than that of the oriented Cr/Ni tape because of partial substitution of Ni for Cu, the Tc still remained high. These results indicate that the NiO/Ni substrate is more preferable for (Hg_{0.75}Re_{0.25})1223/NiO/Ni tapes than the Cr/Ni substrate.



. Fig. 2. Temperature dependence of the resistivity ${\rm of} \ ({\rm Hg_{0.75}Re_{0.25}})1223/NiO/Ni \ tape.$

4. Conclusions

 $(Hg_{0.75}Re_{0.25})1223/NiO/Ni$ tapes were fabricated on the Cr/Ni and NiO/Ni substrates. The c-axis oriented $(Hg_{0.75}Re_{0.25})1223/NiO/Ni$ tapes were reproducibly fabricated. The observed Birr was quite high, > 5T at 77K.

References

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