

## Electron microscopy of a high strength Mg-Zn-Y alloy containing a quasicrystalline phase

*S.H. Kim<sup>1</sup>, B.J. Inkson, G. Möbus<sup>2</sup>, D.H. Bae, D.H. Kim<sup>3</sup>, W.T. Kim<sup>4</sup>*

<sup>1</sup>*School of Advanced Materials and Systems Eng., Kumoh National Inst. of Tech.*

<sup>2</sup>*Department of Materials, University of Oxford, Parks Road, Oxford OX1 3PH, UK*

<sup>3</sup>*Center for Non-crystalline Materials, Dept. of Metallurgical Engineering, Yonsei University*

<sup>4</sup>*Center for Non-crystalline Materials, Department of Physics, Chongju University*

### Abstract

A new magnesium alloy containing quasicrystalline particles was developed in magnesium rich Mg-Zn-Y system. This new alloy exhibits high strength and large elongation at room and elevated temperature. The Mg<sub>95</sub>Zn<sub>4.3</sub>Y<sub>0.7</sub> specimen was fabricated by conventional casting followed by thermomechanical processing, i.e. hot-rolling around 473 K and then annealing for 30 min at 673K.

The as-cast alloy consisted of primary  $\alpha$ Mg dendrites and eutectic pockets (Mg and icosahedral(i-) phase) in the interdendritic region. The hot-rolled alloy showed equiaxed  $\alpha$ Mg grains with size of 14  $\mu$ m and a distribution of small i-phase particles (0.5-2.0  $\mu$ m) near the grain boundaries of the  $\alpha$ Mg, which were destroyed dendrites during hot-rolling.

To understand the strengthening mechanism of this alloy, TEM investigations were carried out using Philips CM20 and JEOL 3000F transmission electron microscopes. Bright field images show the existence of i-phase particles distributed in  $\alpha$ Mg. The existence of i-phase in this deformed specimen clearly indicates that the i-phase thermally equilibrates with  $\alpha$ Mg. Furthermore, any debonding or nanoscale defects at the particle/matrix interface cannot be observed in the tested specimen.

Besides the i-phase, two more nano-sized phases were found throughout the  $\alpha$ Mg grains. One has a very fine acicular morphology (50 nm in width, 0.5-3.0  $\mu$ m in length) and the other has a bulky particle shape (a few hundred nanometer) as shown in Fig. 1. The acicular phases seem to be orthorhombic structure and show specific orientation relationships with the  $\alpha$ Mg matrix. The Mg [0001] direction is parallel with longitudinal direction of needle of acicular phases. HREM imaging shows a coherent interface between acicular nano-phases and Mg matrices (Fig. 2). The bulky nano-phases seem to have hexagonal

structure and also have specific orientation relationships with the Mg matrix. A high density of dislocations was also observed in this specimen. It is, therefore, thought that in addition to the strengthening effect of i-phase particles, strengthening of this alloy was also achieved by these fine particles, which can act as obstacles for moving dislocations.

In general, particle/matrix debonding has been considered as an initial stage of the failure mechanism during the high temperature deformation of alloys with intermetallic compound particles at grain boundaries, and the coalescence of many cavities at large strains can induce the failure of such an alloy. However, careful examinations of the i-phase/matrix interfaces in this alloy do not reveal any defects. Therefore, the improved mechanical properties might be due to stable i-phase/matrix interfaces. HREM imaging shows that the i-phase/matrix interfaces sometimes have the orientation relationships of  $I2//Mg[0001]$ , even though i-phase particles are highly phason-strained.

#### **Acknowledgements**

A part of this work was supported by the Creative Research Initiatives of the Korean Ministry of Science and Technology.

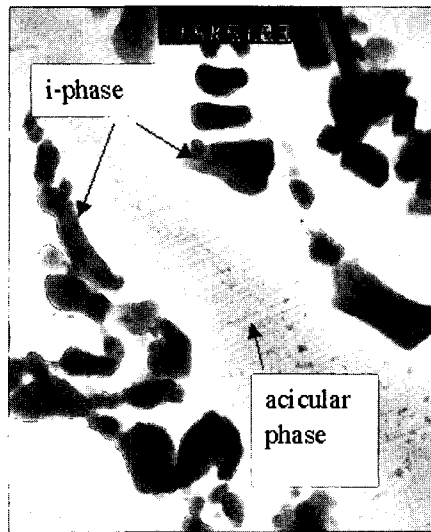


Fig. 1. Bright field image of a Mg<sub>95</sub>Zn<sub>4.3</sub>Y<sub>0.7</sub> alloy showing the fine precipitates between icosahedral quasicrystalline particles

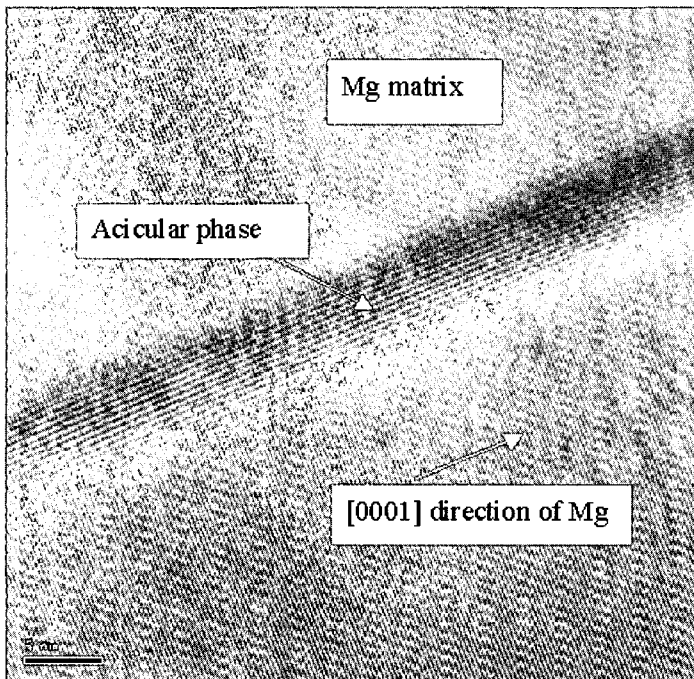


Fig. 2. High resolution electron microscope image showing an acicular phase in Mg<sub>95</sub>Zn<sub>4.3</sub>Y<sub>0.7</sub> alloy