Emitting Level Change and Enhancement of Red Emission from SrTiO₃:Pr³⁺ by Y³⁺ Addition

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Abstract

The effect of Y^{3+} addition to $SrTiO_3:Pr^{3+}$ on the photoluminescence and cathodoluminescence was studied. We discovered that light emitting levels of Pr^{3+} vary by addition of Y^{3+} . In $(Sr_{1-x}Y_x)TiO_3:Pr^{3+}$, both the green and red emission are discovered while the red emission prevails in $Sr(Ti_{1-x}Y_x)O_3:Pr^{3+}$. $Sr(Ti_{1-x}Y_x)O_3:Pr^{3+}$ shows enhancement of red emission by two kinds of enhancement process.

INTRODUCTION

Field emission display (FED) is promising flat panel display for next generation. But, one of the most serious problems to be solved is to find proper phosphors. Although cathode ray phosphors have been optimized to work at high voltages and relatively low current densities, the requirements for FED are totally different. So new phosphors are required for FED and various kinds of phosphor are being researched.

SrTiO₃:Pr³⁺ is known as the red phosphor under low-energy electron or uv light excitation and its intensity is enhanced by addition of Al³⁺,Ga^{3+ 1)}. In this work, we discovered the light emitting level change and red emission enhancement by Y³⁺ addition.

EXPERIMENT

Polycrystalline compounds with nominal compositions (Sr_1 , A_x) TiO_3 : Pr^{3+} and $Sr(Ti_{1-x}A_x)O_3$: Pr^{3+} were prepared. The appropriate amounts of $SrCO_3$, TiO_2 , Y_2O_3 , Ga_2O_3 and $PrCl_37H_2O$ were mixed together under ethanol in agate mortar for 4hrs, then fired under air at $1200\,^{\circ}\text{C}$ for 6 hours.

XRD pattern was obtained using Rigaku, D/Mac-RC. The PL excitation and emission spectra were obtained with a Aminco Bowman Luminescence spectrometer. The CL was measured at room temperature using a demountable ultrahigh vacuum chamber equipped with in-house assembled CL spectrophotometer The CL measurements were carried out with an accelerating voltage of 700V. The fluorescence were measured by pulsed radiation PL Perkin Elmer spectrometer.

RESULTS AND DISCUSSIONS

XRD patterns in Figure 1 reveal that a small amount of Y³⁺ can be incorporated in SrTiO₃:Pr³⁺.

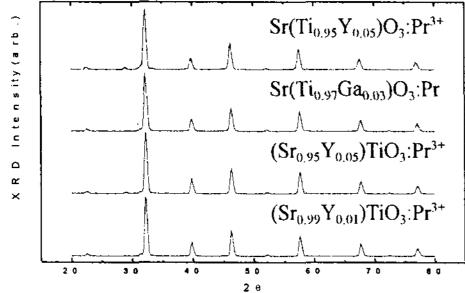


Fig.1. XRD patterns of $Sr(Ti_{1-x}Y_x)O_3:Pr^{3+}$ and $(Sr_{1-x}Y_x)TiO_3:Pr^{3+}$ doped with different amount of Y^{3+} .

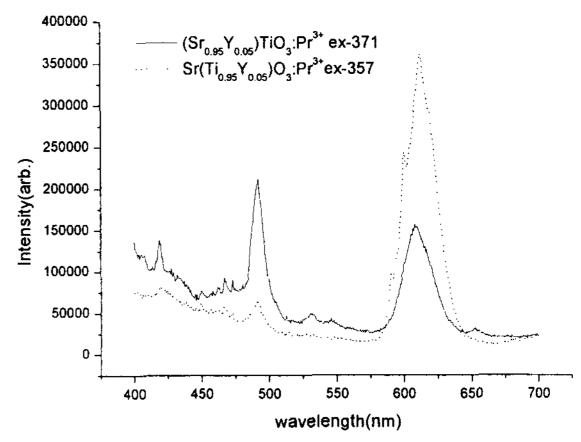


Fig. 2. PL emission spectra of $Sr(Ti_{0.95}Y_{0.05})O_3:Pr^{3+}$ and $(Sr_{0.95}Y_{0.05})TiO_3:Pr^{3+}$ at highest peak of excitation band.

Figure 2 shows the PL emission spectra of 5 mol% of Y³⁺ doped SrTiO₃:Pr³⁺with different molar ratio of Sr and Ti. As shown in the figure, the green emission band is due to the transition from ³P₀ to ³H₄ at 493 nm and red emission band due to the transition from ¹D₂ to ³H₄ at 614 nm. The red emission band is very weak in SrTiO₃:Pr³⁺ without Y³⁺¹. There is severe influence of excitation and emission bands depending on the molar ratio of Sr and Ti for SrTiO₃:Pr³⁺. The addition of Y³⁺ to Ti-deficient SrTiO₃:Pr³⁺ (Sr(Ti_{1-x}Y_x)O₃:Pr³⁺: a-type) reveals that the red emission band increases, but the green emission at 493nm is very weak. Intensity of the red emission decreases while the green emission increases by the addition of Y³⁺ to Sr-deficient SrTiO₃:Pr³⁺ ((Sr_{1-x}Y_x)TiO₃:Pr³⁺: b-type). This green emission of Pr³⁺ is discovered in other host lattices^{2),3)}, but not in SrTiO₃:Pr³⁺ at room temperature.

Figure 3 shows the PL excitation spectra for 5 mol% of Y^{3+} doped $SrTiO_3:Pr^{3+}$ with different molar ratio of Sr and Ti. The excitation spectrum consists of broad band below 380 nm and sharp band in the region $440 \sim 490$ nm. The sharp band consists of three bands peaked at 450, 470 and 486 nm, it is assigned to optical transition of ${}^3H_4 \rightarrow {}^3P_J$ (J=0,1,2) in Pr^{3+} , respectively. The broad band below 380 nm is different between a-type and b-type. In the case of a-type, there is intense broad band below 380nm at emission due to ${}^1D_2 \rightarrow {}^3H_4$ (614nm) but not at emission due to ${}^3P_0 \rightarrow {}^3H_4$ (493 nm).

In the case of b-type, the intensity of excitation band below 380 nm at emission due to ${}^3P_0 \rightarrow {}^3H_4$ (493 nm) is greater than at emission due to ${}^1D_2 \rightarrow {}^3H_4$ (614nm). This process can be explained by ${}^3P_0 \rightarrow {}^1D_2$ non-radiative relaxation. After absorbing the energy in UV region, the population of the 3P_0 state occurs. We confirmed this by measuring fluorescence. After exciting both of them at 360nm by pulsed radiation, we discovered the very intense peak at 493nm in both emission spectra. Through ${}^3P_0 \rightarrow {}^1D_2$ non-radiative relaxation, it results in the population of the 1D_2 state. In a-type, ${}^3P_0 \rightarrow {}^1D_2$ non-radiative relaxation is too fast for to emit 493nm emission originating from 3P_0 , so only 614nm emission originating from 3P_0 is dominant. But, in b-type ${}^3P_0 \rightarrow {}^1D_2$ non-radiative relaxation is not so fast that 493nm emission originating from 3P_0 can occur quite well.

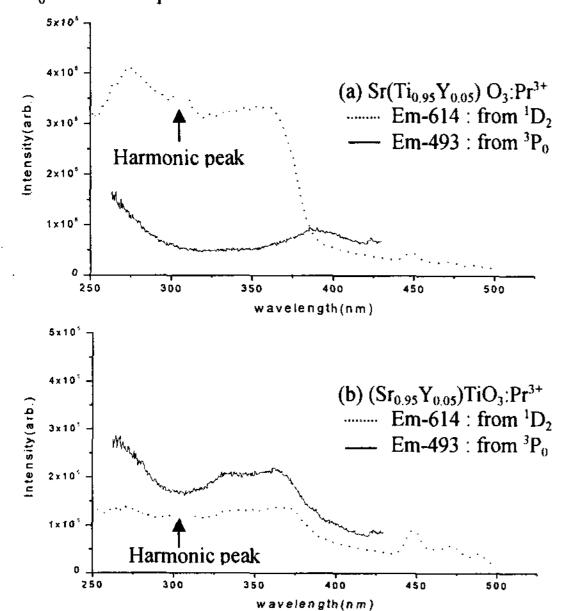


Fig. 3. PL excitation spectra of (a) $Sr(Ti_{0.95}Y_{0.05})O_3:Pr^{3+}$ and (b) $(Sr_{0.95}Y_{0.05})TiO_3:Pr^{3+}$

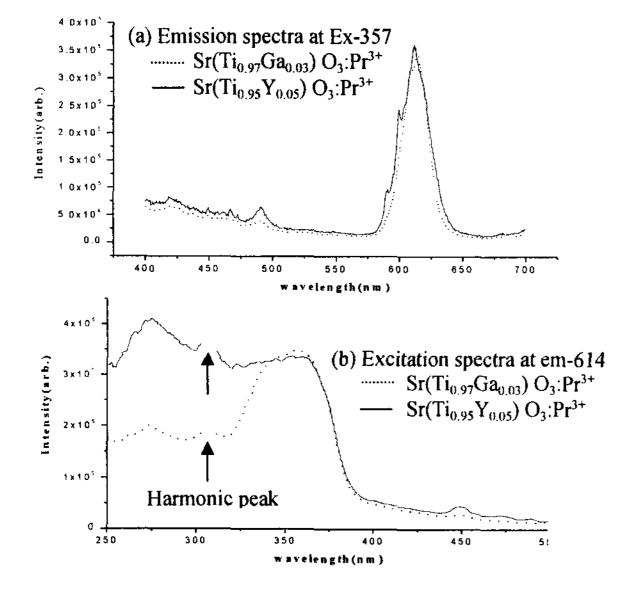


Fig. 4. PL (a)emission and (b) excitation spectra of $Sr(Ti_{0.95}Y_{0.05})O_3:Pr^{3+}$ and $Sr(Ti_{0.97}Ga_{0.03})O_3:Pr^{3+}$

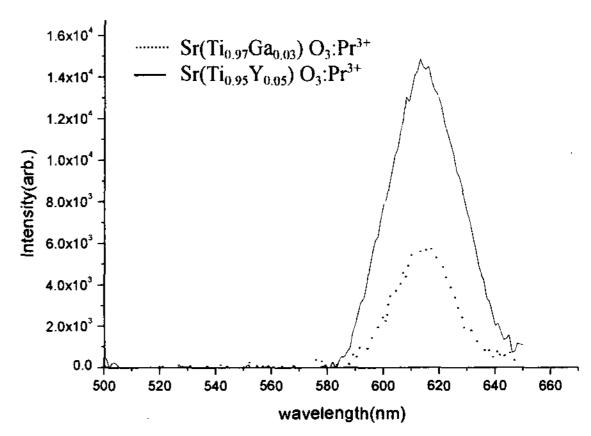


Fig. 5. CL spectra of $Sr(Ti_{0.95}Y_{0.05})O_3:Pr^{3+}$ and $Sr(Ti_{0.97}Ga_{0.03})O_3:Pr^{3+}$ (acceleration voltage: 700 V).

Figure 4 shows PL excitation and emission spectra of Ga³⁺ doped a-type and Y³⁺ doped a-type. Their emission spectra are similar, but excitation spectra is absolutely different. The intensity of excitation band around 360 nm is nearly same, but the intensity of excitation band around 270 nm is different. The intensity of excitation for Y³⁺ doped a-type is two times higher than Ga³⁺ doped a-type. This result coincides with CL result that the CL intensity of Y³⁺ doped a-type will be higher than that of Ga³⁺ doped a-type as shown in Fig. 5. And emission at 493nm is higher in Y³⁺ doped a-type than in Ga³⁺ doped a-type so we think that excitation band around 270nm is related with green emission.

Recently, it was reported that the broad excitation band around 360nm is due to host lattice excitation at emission by self-trapped excitions(500nm) in SrTiO₃. ^{1),4),5)} But, the broad excitation band around 270nm is not explained yet. We think this band is related with absorption of Pr³⁺ itself and affected by the distance between Pr³⁺ and O ligand, which can be varied by the size of co-activator. We also suppose that the location of this band is related with emitting level change discussed before.

CONCLUSION

We have observed the change of emission band of green and red emission for SrTiO₃:Pr³⁺ by Y³⁺ and Ga³⁺ addition. The addition of Y³⁺ to Sr-deficient SrTiO₃:Pr³⁺ reveals the enhancement of green emission at 493 nm. But the addition of Y³⁺ to Ti-deficient SrTiO₃:Pr³⁺ reveals the enhancement of red The mechanism of enhancement¹⁾ was suggested for Al-doped SrTiO₃:Pr³⁺, there will be another mechanism which is not explained yet. Further investigation is necessary to find it out.

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